

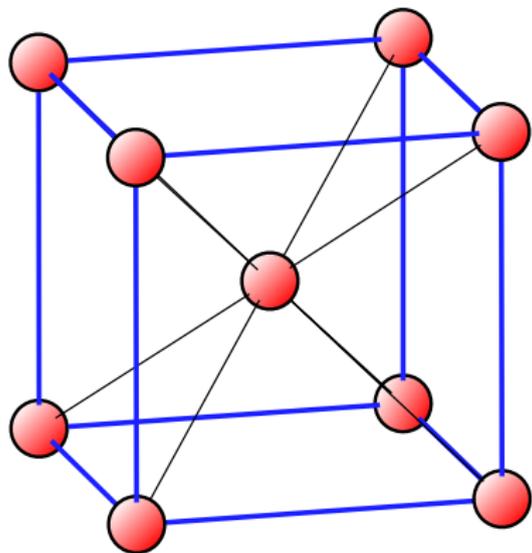
**Multiscale Aspects of Materials Modelling:
A Mathematician's View**

Tom Hudson, Warwick Mathematics Institute

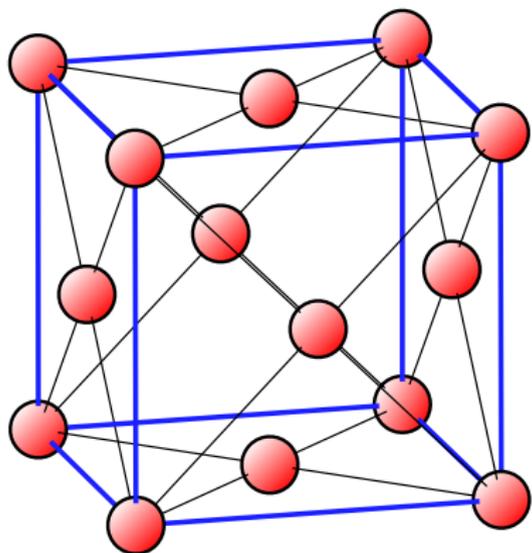
IPAM Tutorials

March 2023

A motivating multiscale system: Crystals

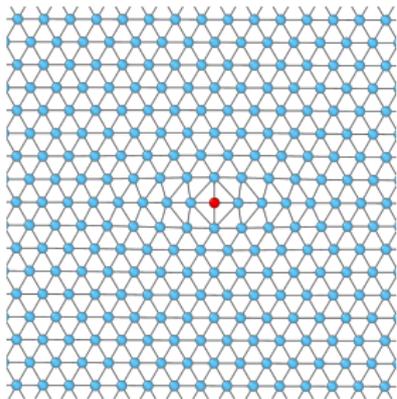


Body-Centred Cubic (BCC)

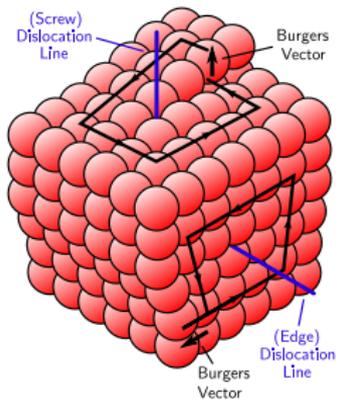


Face-Centred Cubic (FCC)

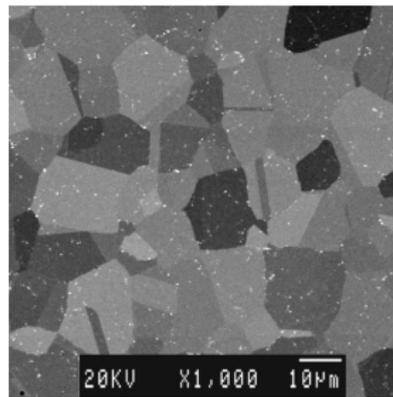
A motivating multiscale system: Crystals



0D: Point Defects

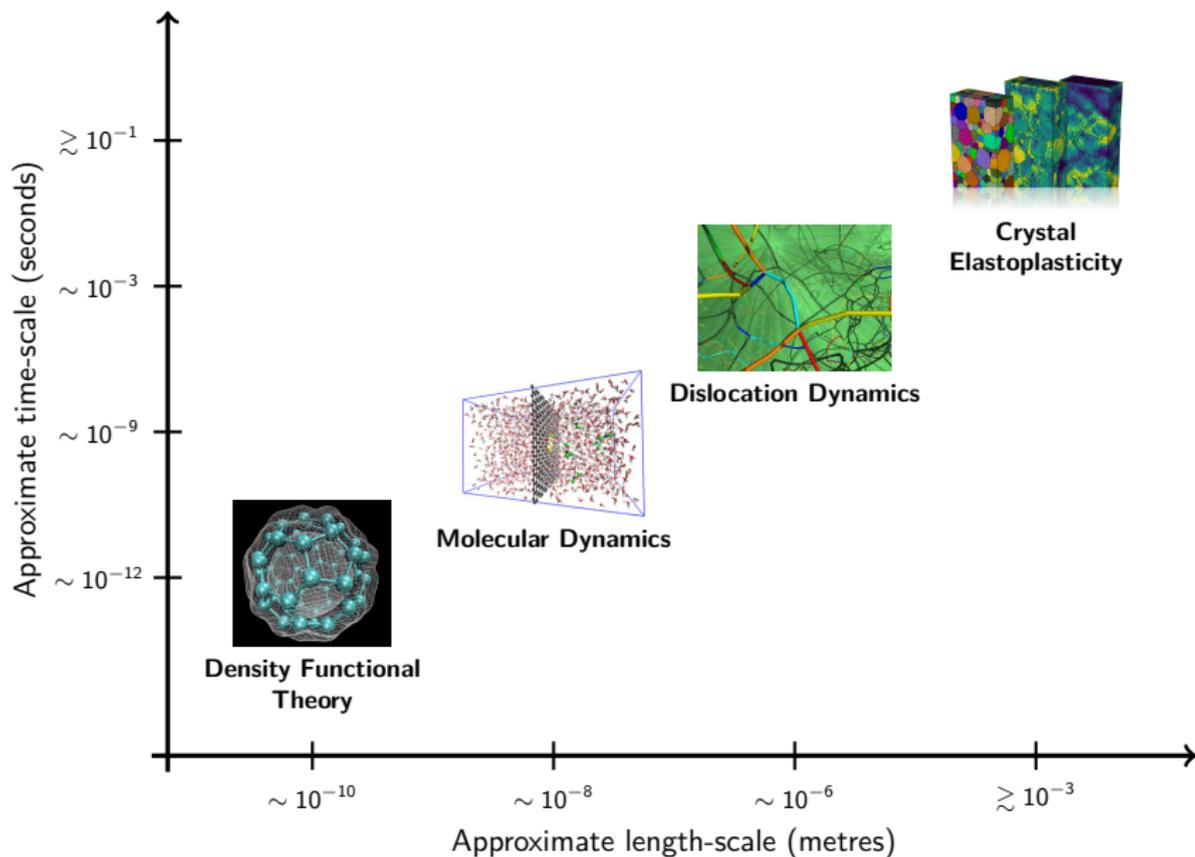


1D: Dislocations

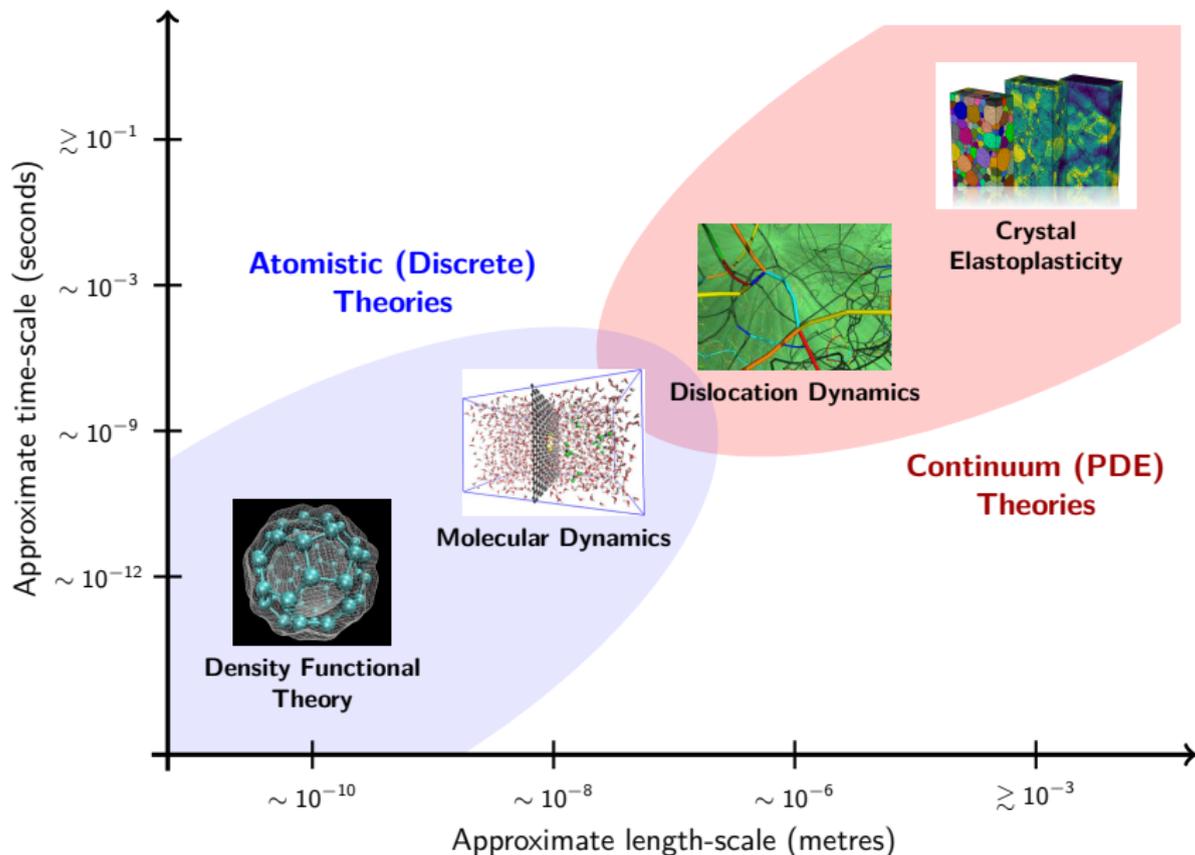


2D: Grain boundaries

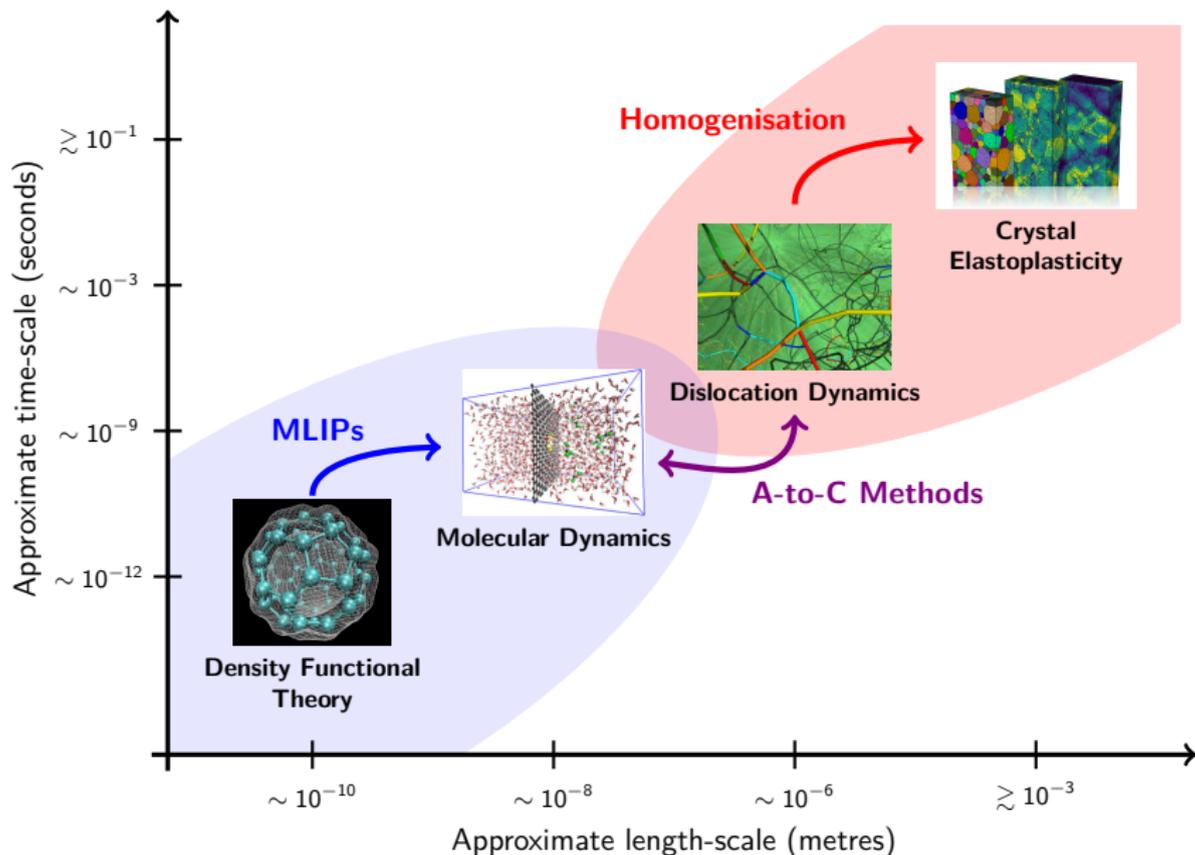
Modelling approaches



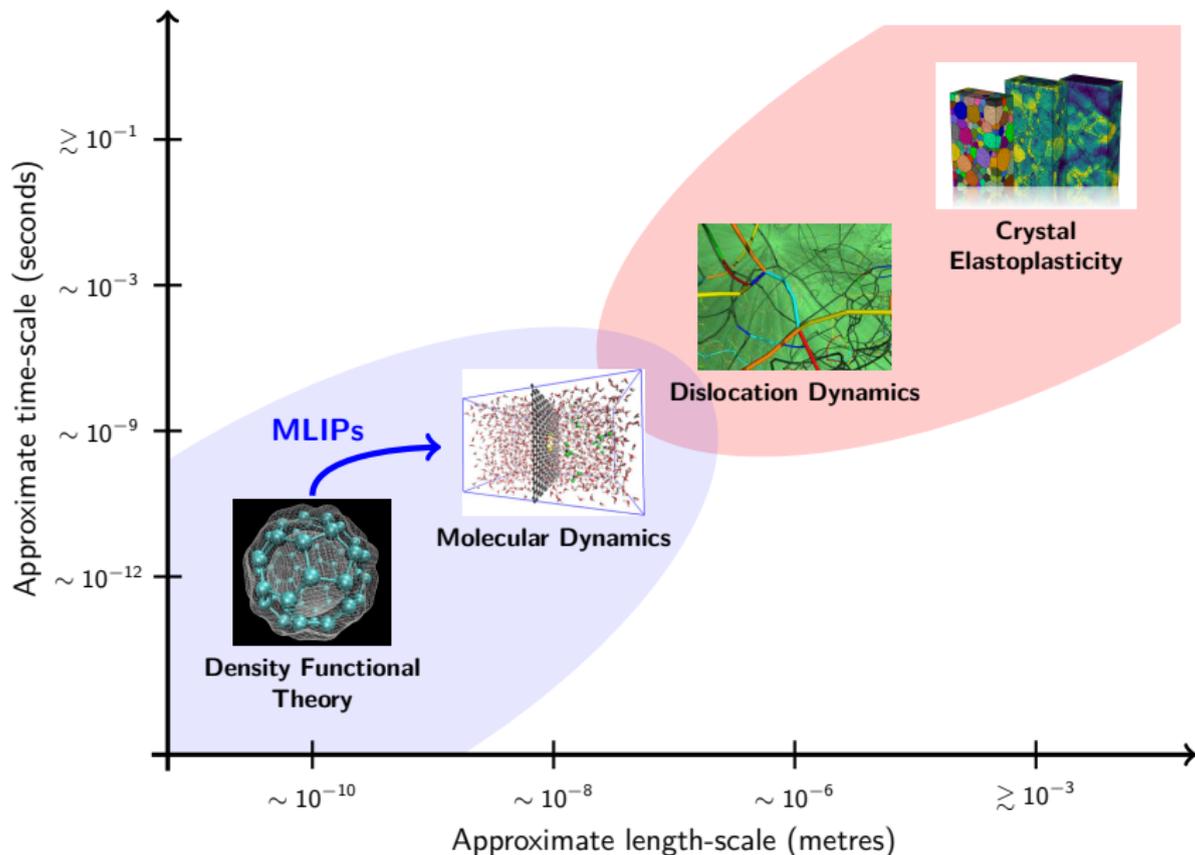
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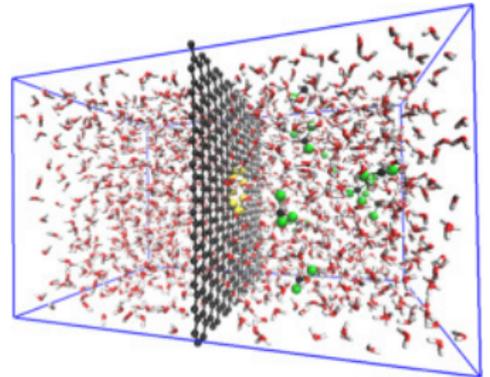
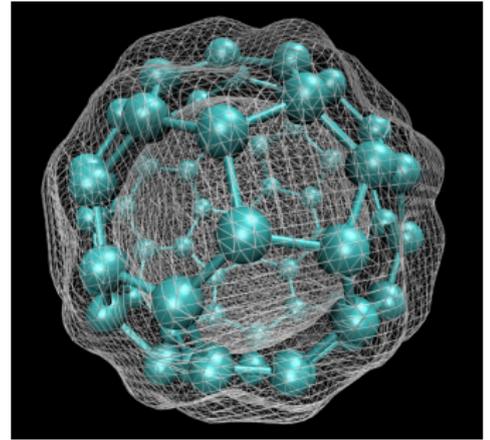


Machine-Learned Interatomic Potentials (MLIPs)



Atomistic theories

- ▶ Variables in these theories:
 - ▶ **Atomic nuclei** $\ell_i \in \mathbb{R}^3$
 - ▶ **Electron density** (in DFT).

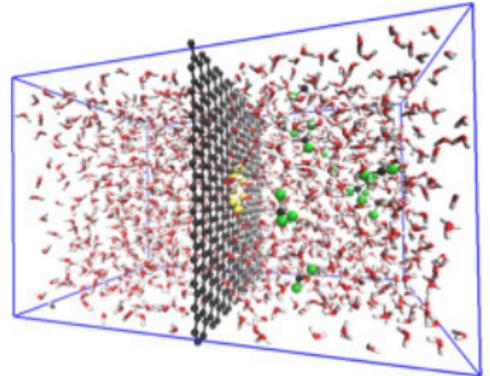
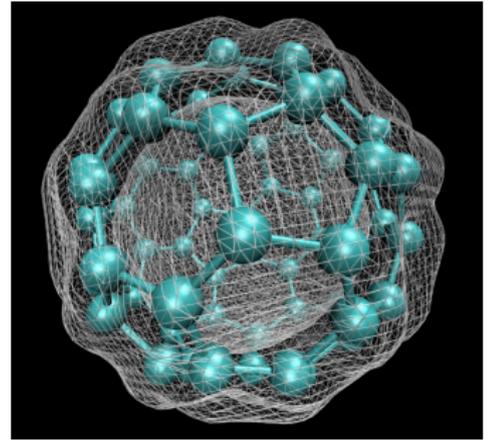


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Forces: $\mathbf{f}_j = -\frac{\partial \mathcal{E}}{\partial \ell_j}(\{\ell_i\}_i)$



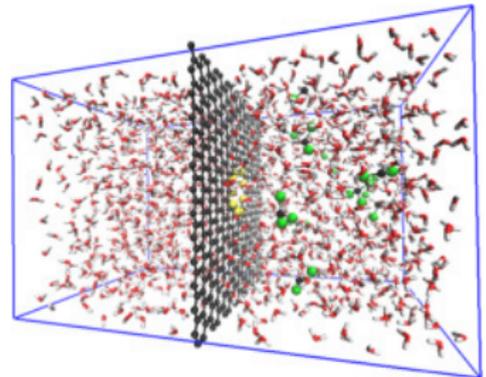
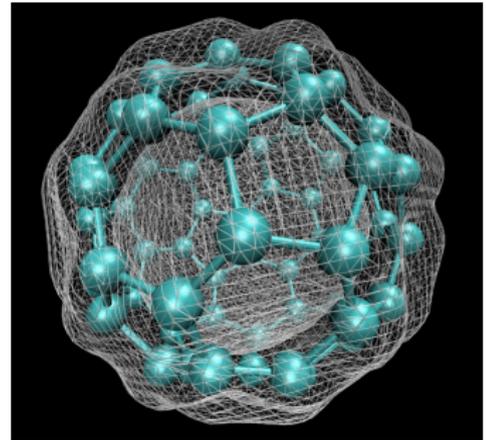
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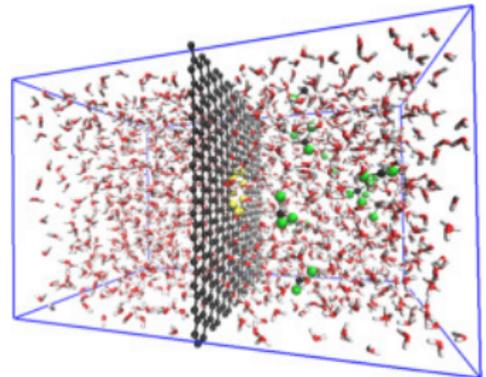
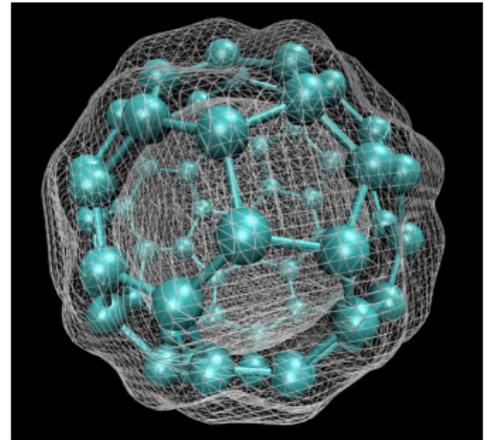
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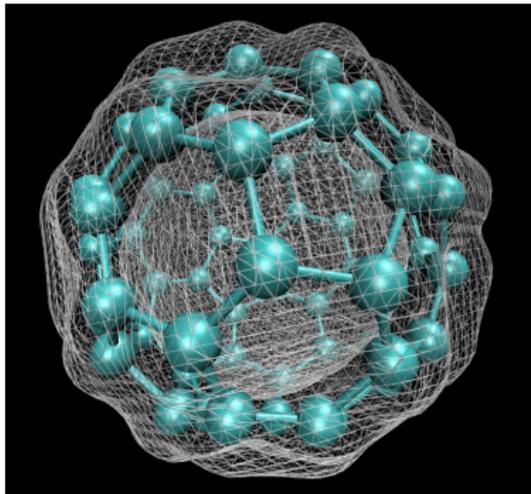
- ▶ **DFT energies** are expensive to evaluate: nonlinear eigenvalue problems requiring iterative solves.
- ▶ **MD energies** are empirical and fitted: cheaper, but not uniformly accurate (c.f. Tim Germann's talk).



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Build-your-own MLIP:

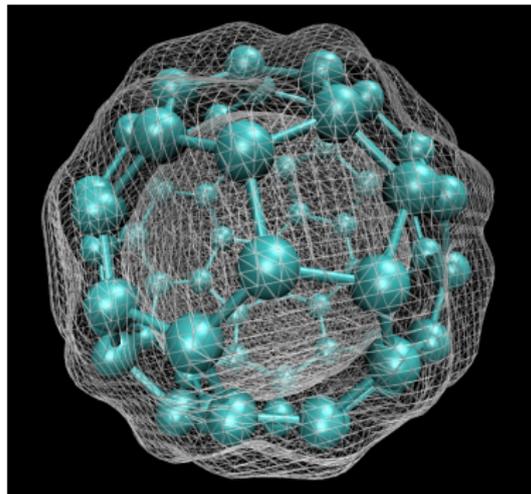
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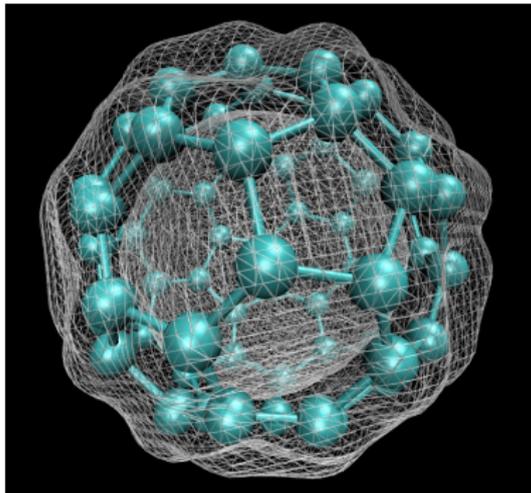
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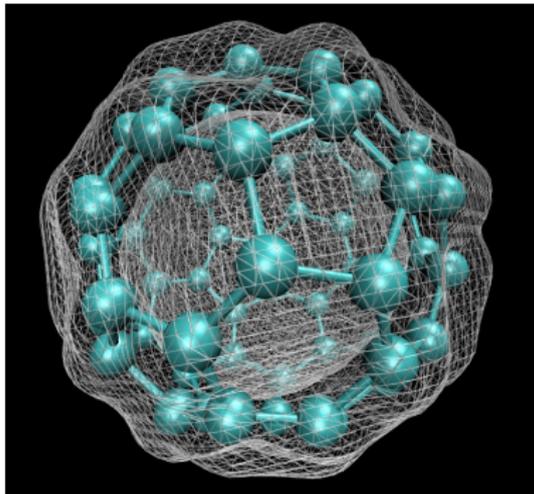
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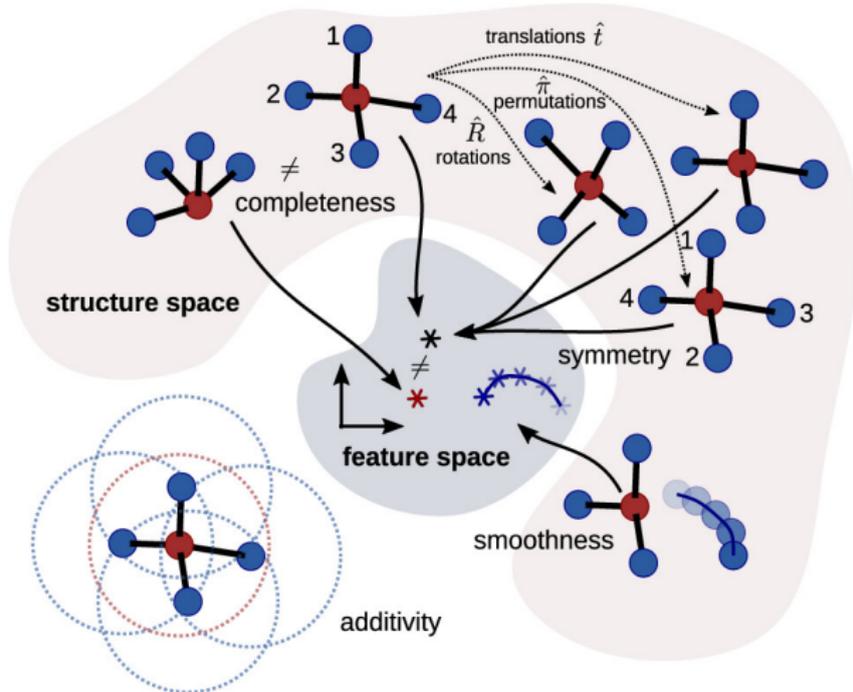
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- ▶ **Parametrise** a class of energy functions based on these descriptors.
- ▶ **Fit** parametric energy (and forces) using a regression methodology.



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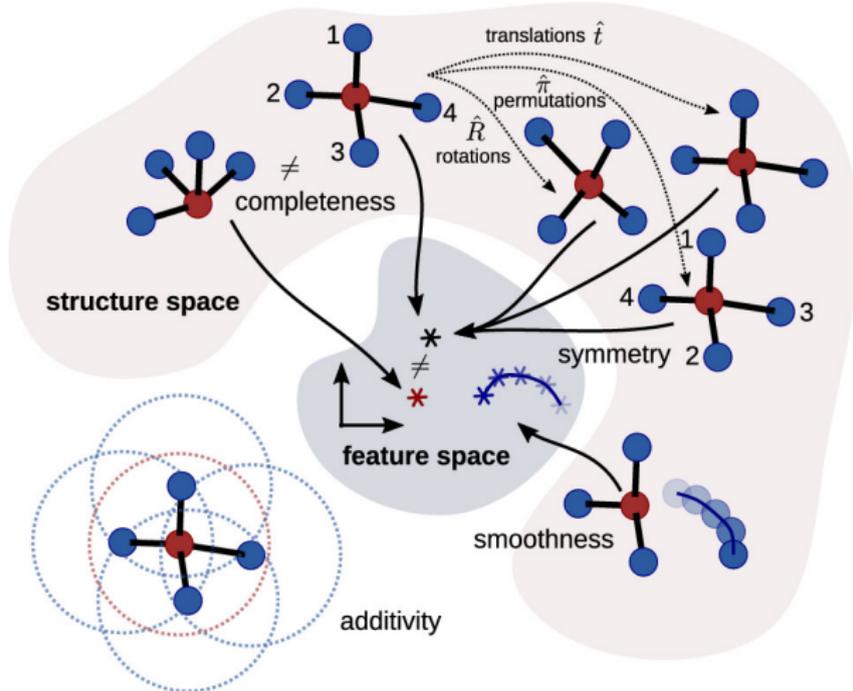
Key ingredient: **Descriptors** which compress structural information.



Musil et al, 2021

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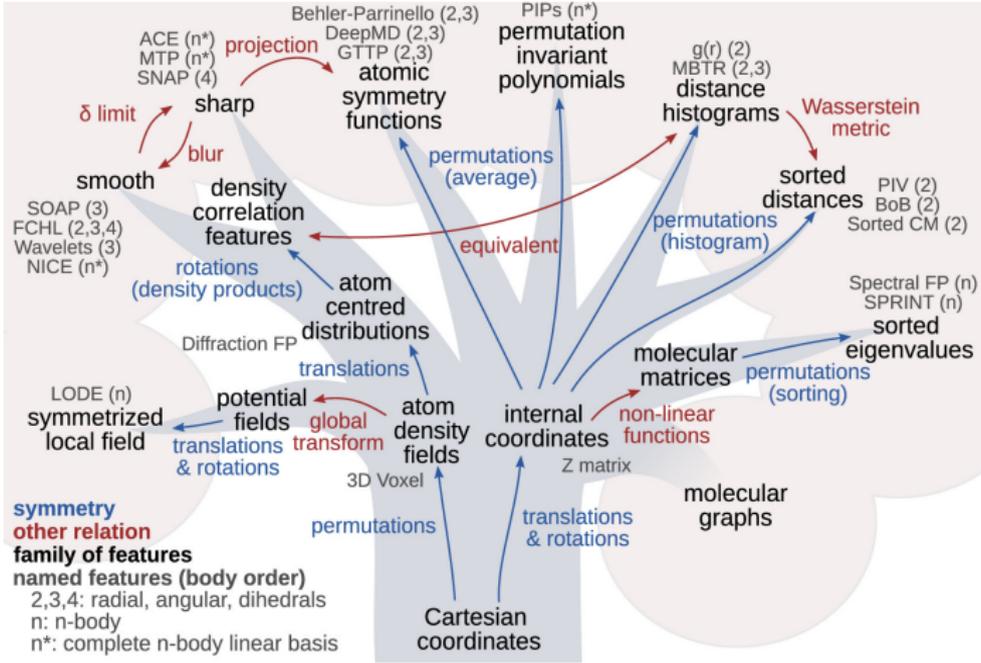
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Efficient representations 'integrate out' symmetries.

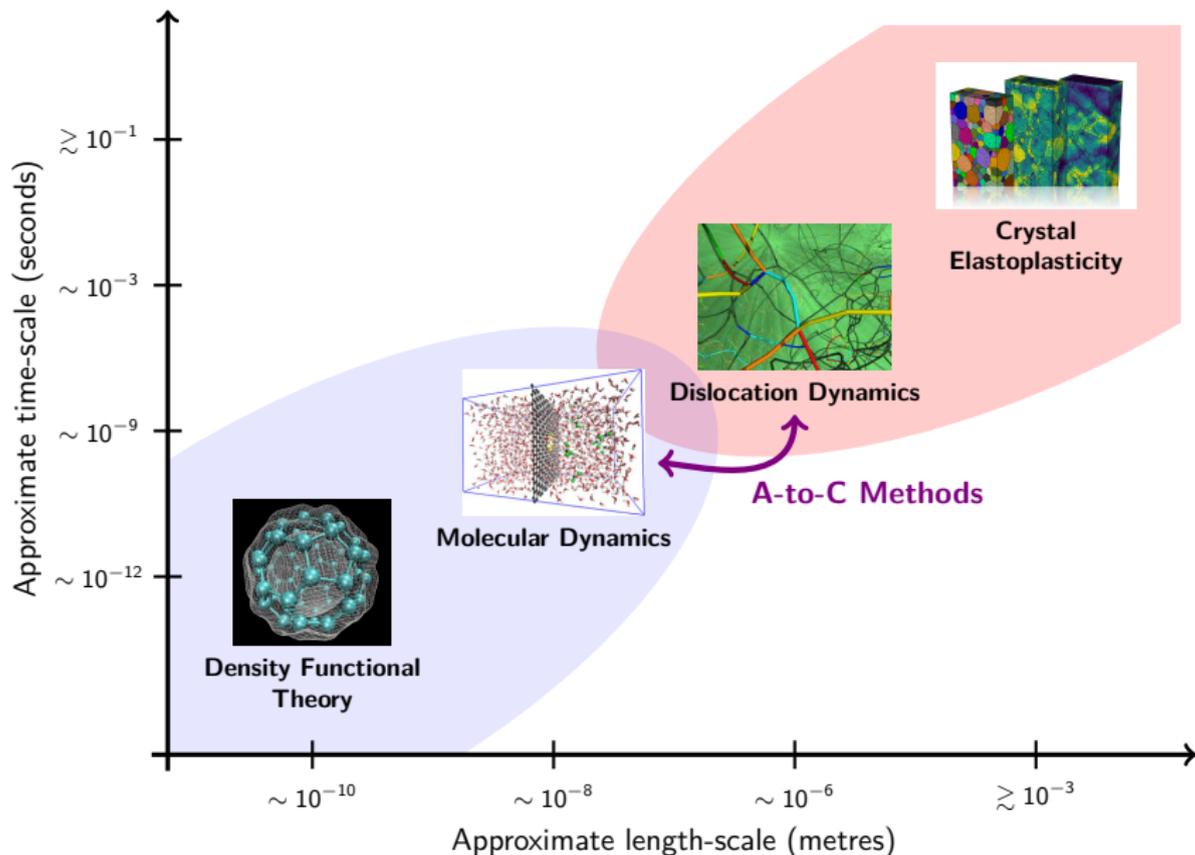
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Open questions:

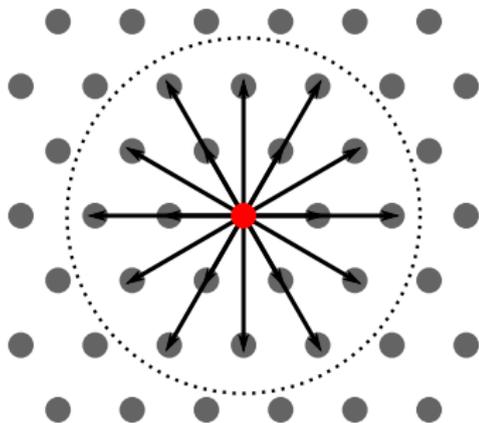
- ▶ Which architectures/descriptors are 'best'? Efficiency, completeness...

Atomistic-to-Continuum Methods



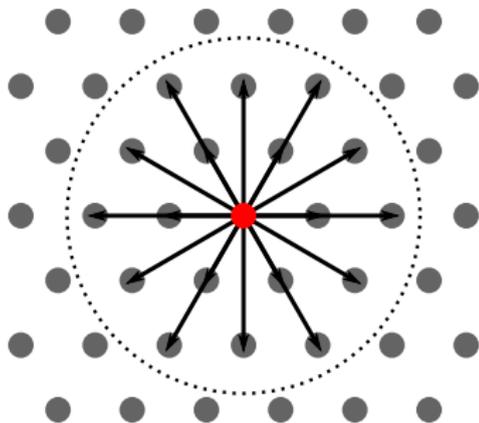
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- ▶ Consider a **deformation** of atoms,
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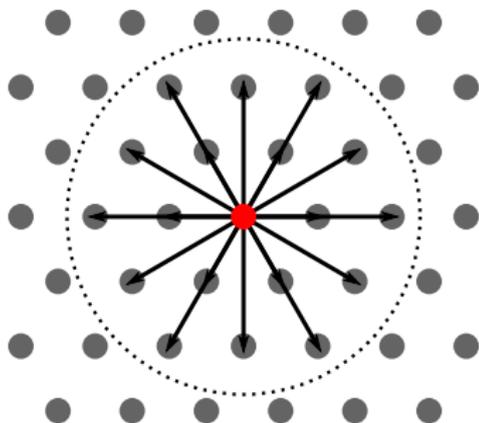
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- ▶ V_ℓ must only depend on **interatomic displacements**:

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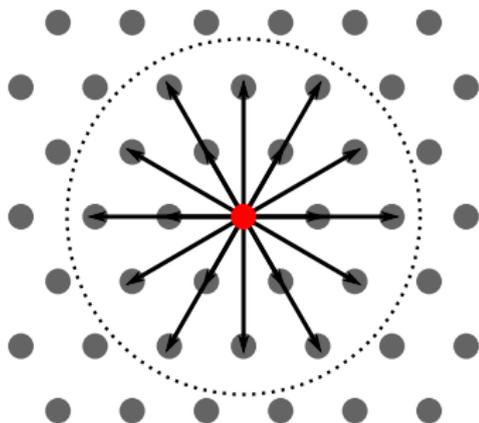
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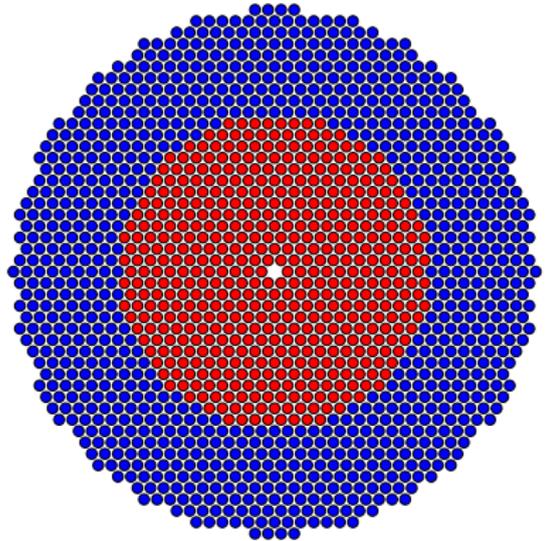
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- ▶ For example, when interactions are pairwise:

$$V_\ell = \sum_{\rho \in \mathcal{R}} \varphi(\mathbf{y}(\ell + \rho) - \mathbf{y}(\ell))$$

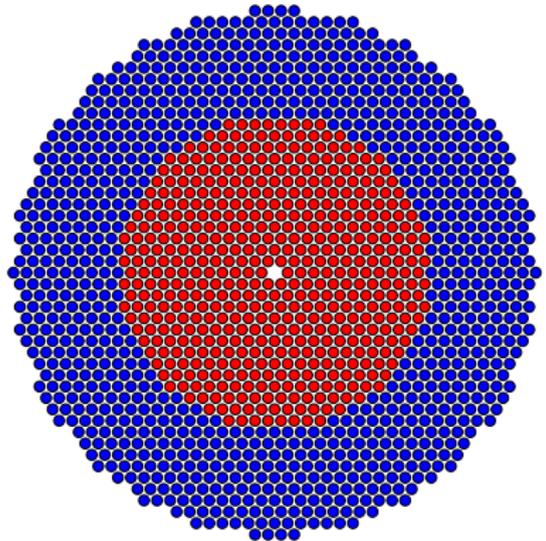
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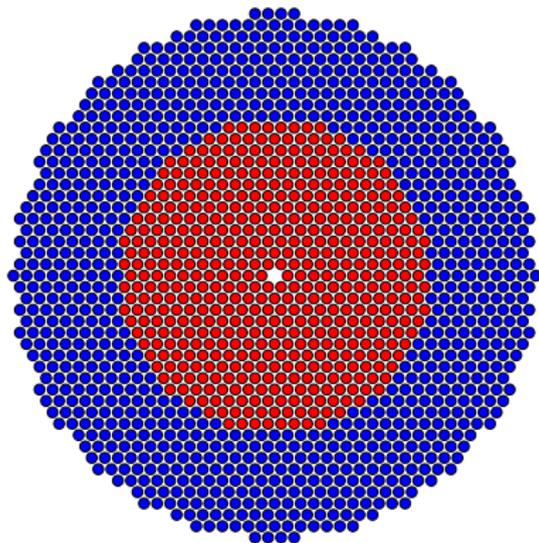
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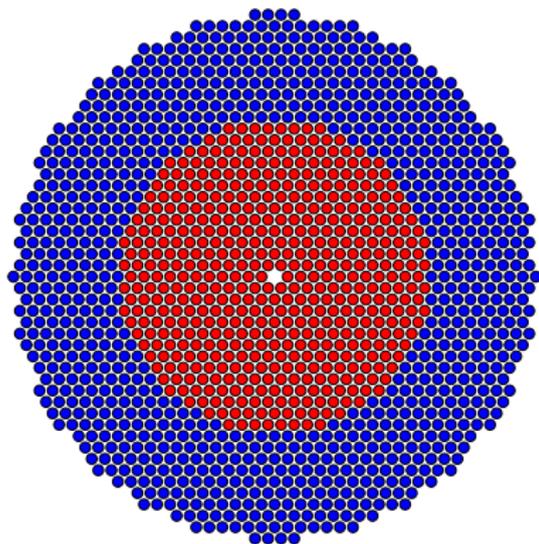
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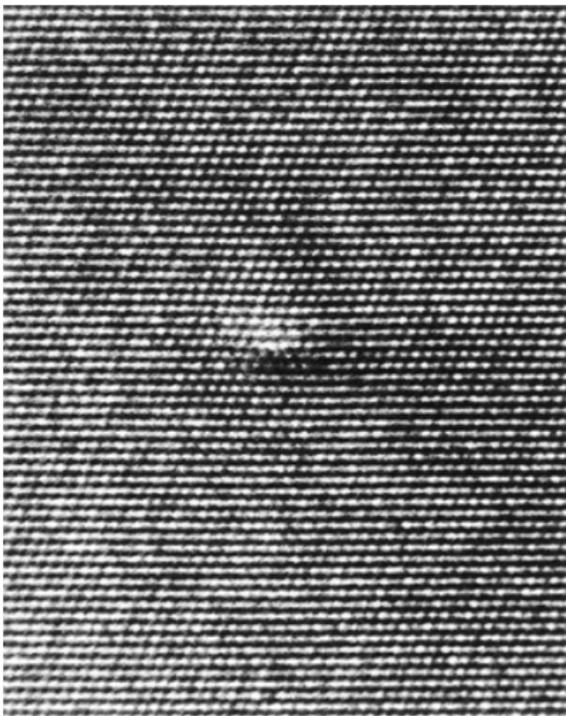
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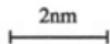
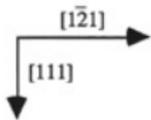


Question: How should the **boundary conditions** be set to achieve an efficient approximation?

An observation

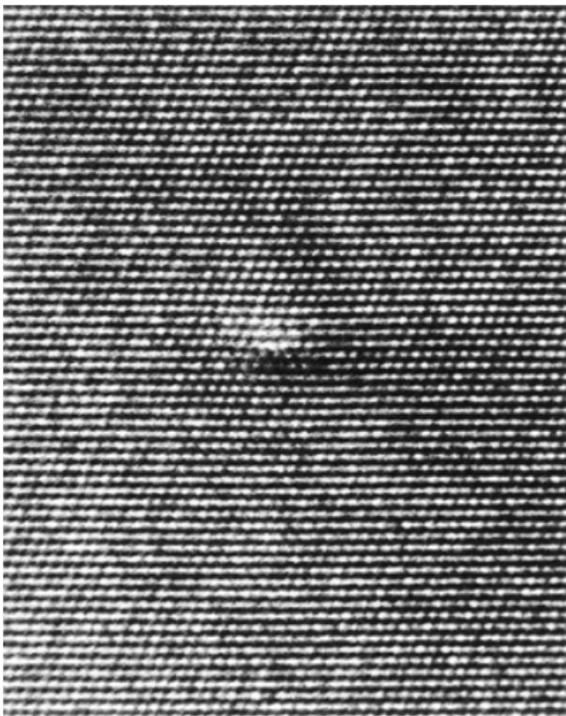


- ▶ Away from a defect core, atomic displacements decay back to lattice positions in a **smooth** way.

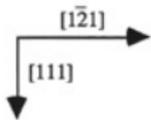


Inkson, 1994

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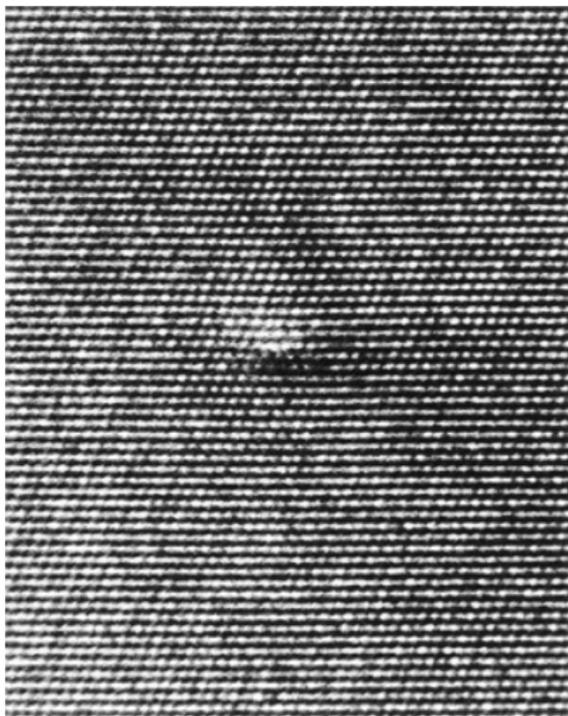
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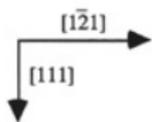
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- ▶ Away from a defect core, atomic displacements decay back to lattice positions in a **smooth** way.
- ▶ **Continuum Linear Elasticity (CLE)** provides singular solutions thought to predict atomic displacements away from defects.
- ▶ Use this insight to inform computational and mathematical approaches.



2nm

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Boundary conditions

In practice: Often use periodic boundary conditions.

Issue: Defect densities end up being very high.

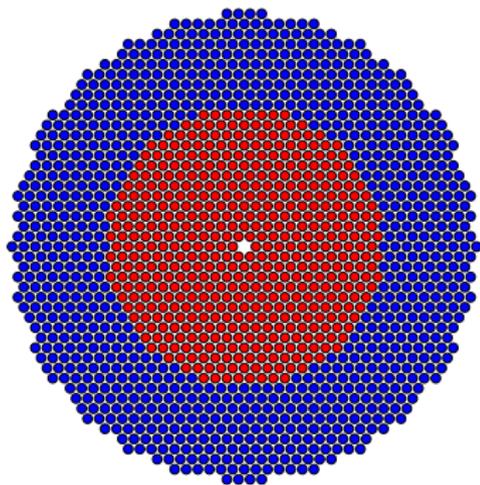
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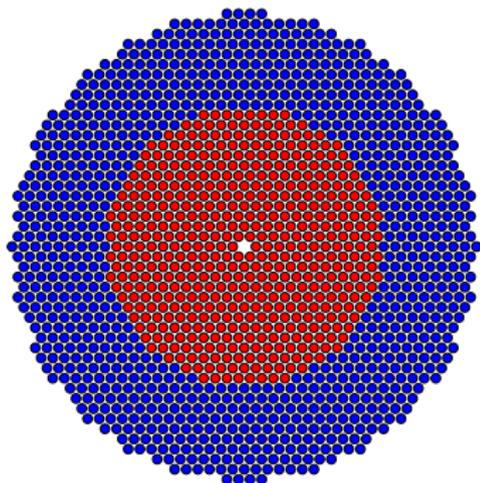
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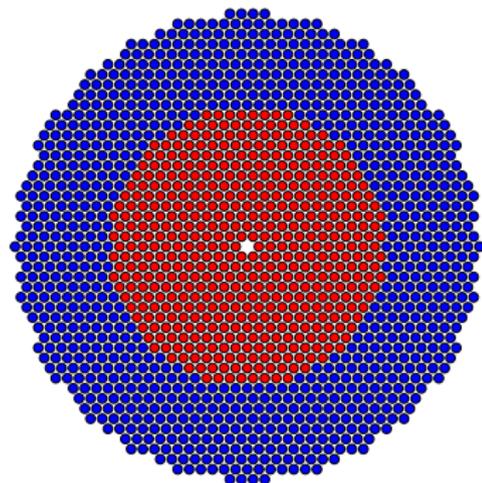
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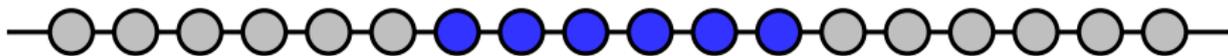


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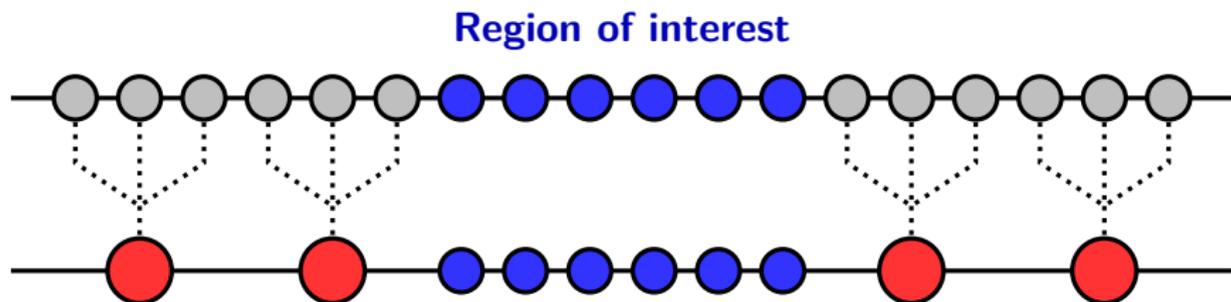
- Luskin and Ortner, 2013. *Acta Numerica*, 22, 397-508.
- Ehlacher et al, 2016. *Arch. Rational Mech. Anal.* 222, 1217-1268.

Coarse-graining

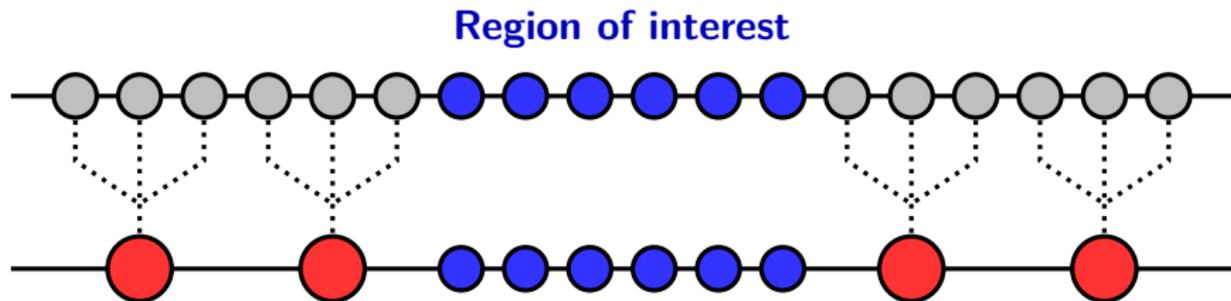
Region of interest



Coarse-graining

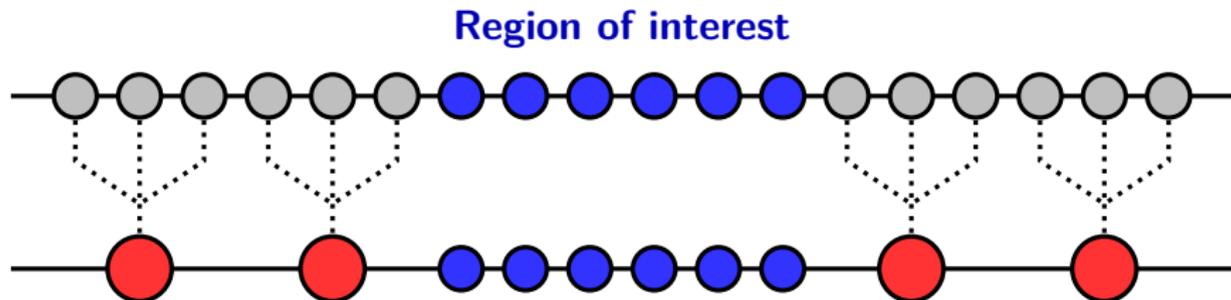


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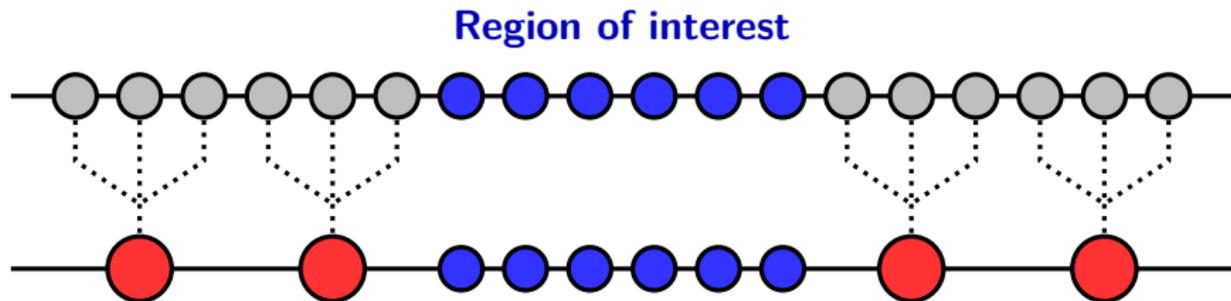
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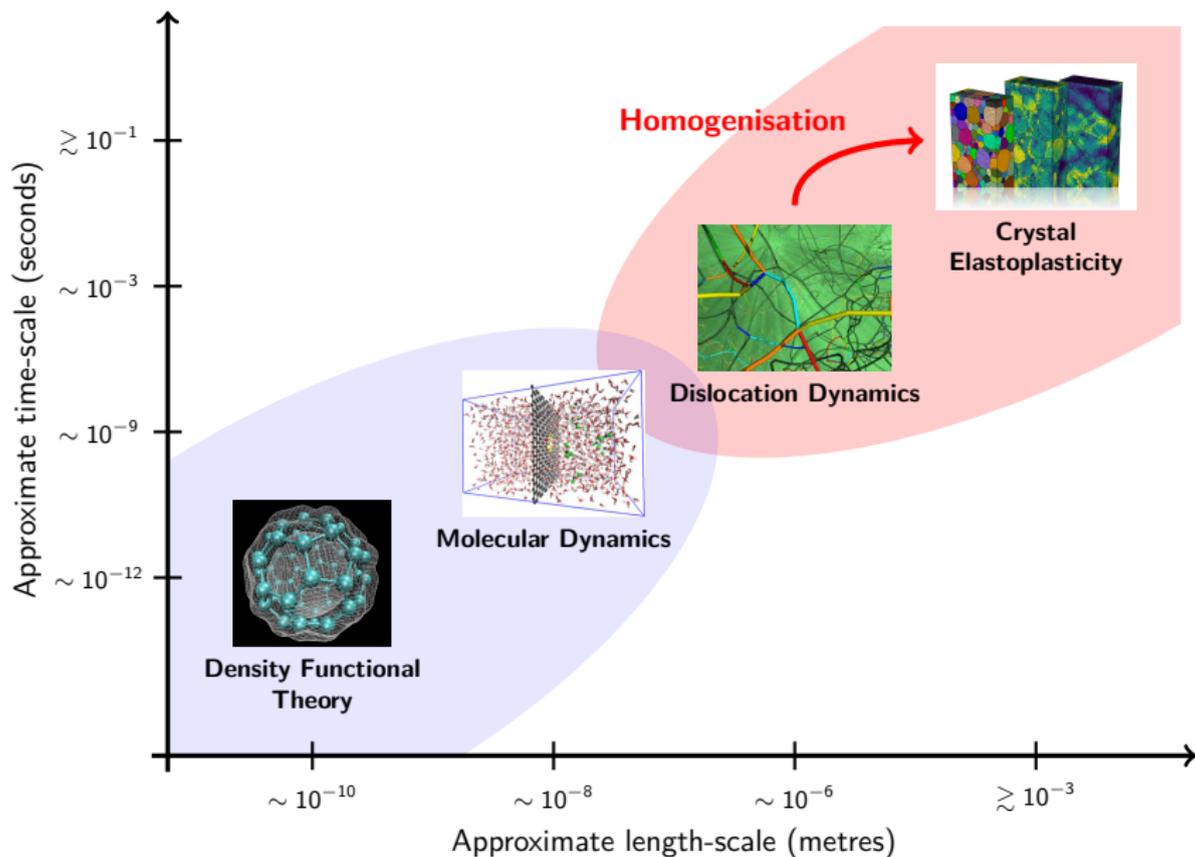
Open problem:

- ▶ Find a robust framework to fit free energy and compatible thermostat.

Reference:

– Lelièvre, Rousset and Stoltz, *Free Energy Computations: A Mathematical Perspective*.

Homogenisation



Homogenisation basics: Periodic case

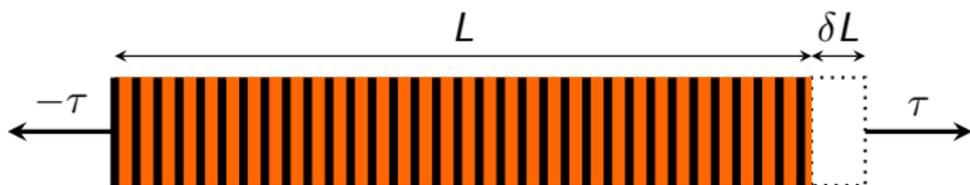
Periodic layered material subject to tensile stress:



1D linear elasticity: $\frac{d\sigma}{dx} = 0, \quad \sigma = k\left(\frac{x}{\varepsilon}\right) \frac{du}{dx}$

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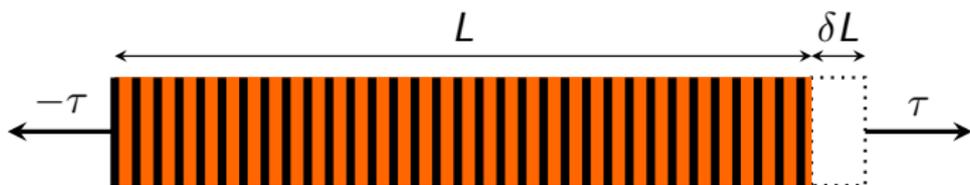


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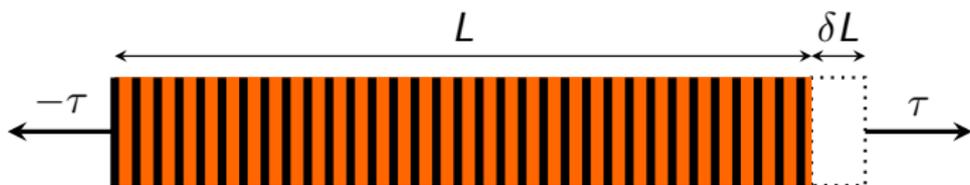
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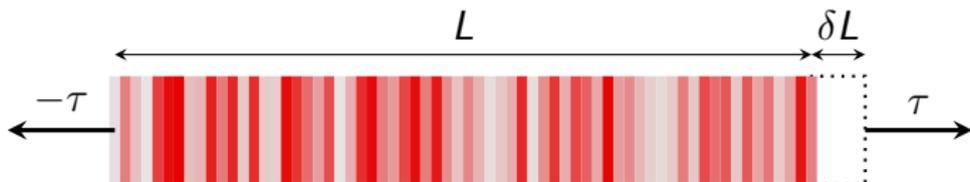
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$$\text{Effective stress-strain: } \tau = K \delta, \quad K = \left(\frac{1}{L} \int_0^L \frac{1}{k\left(\frac{x}{\varepsilon}\right)} dx \right)^{-1}$$
$$\approx \left(\frac{\alpha_1}{k_1} + \frac{\alpha_2}{k_2} \right)^{-1}$$

Homogenisation basics: Stochastic case

Random layered material subject to tensile stress:

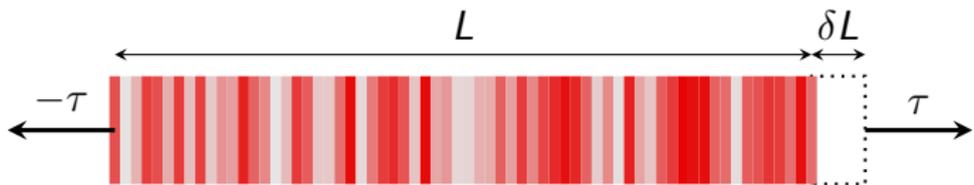


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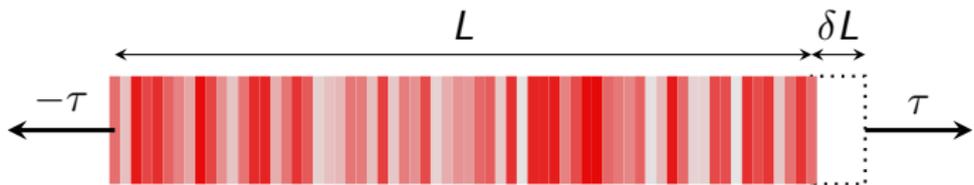
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Tensile loads: $-\sigma(0) = -\tau, \quad \sigma(L) = \tau.$

Effective stress-strain: $\tau = K \delta, \quad K = \left(\frac{1}{L} \int_0^L \frac{1}{k\left(\frac{x}{\varepsilon}, \omega\right)} dx \right)^{-1}.$

Homogenisation basics: Stochastic case

Random layered material subject to tensile stress:



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Assuming **ergodicity**, spatial mean approximates the true mean:

$$\frac{1}{K} \approx \mathbb{E}\left[\frac{1}{k}\right].$$

Homogenisation basics



► More generally, consider the problem

$$-\nabla \cdot \left(\mathbf{C} \left(\frac{\mathbf{x}}{\varepsilon} \right) : \nabla \mathbf{u}^\varepsilon \right) = \mathbf{f}.$$

Homogenisation basics



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- Under the right assumptions: $\mathbf{u}^\varepsilon \rightarrow \mathbf{u}^0$ in L^2 as $\varepsilon \rightarrow 0$, where \mathbf{u}^0 solves

$$-\nabla \cdot \left(\overline{\mathbf{C}} : \nabla \mathbf{u}^0 \right) = \mathbf{f}$$

The effective $\overline{\mathbf{C}}$ may be found numerically by solving a **cell problem**.

Homogenisation basics



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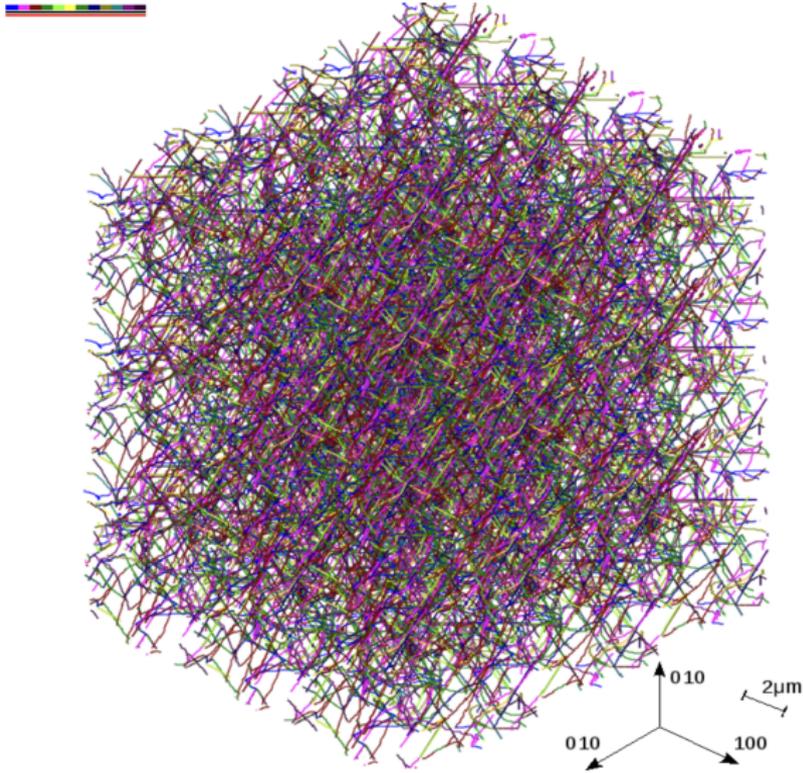
The effective $\overline{\mathbf{C}}$ may be found numerically by solving a **cell problem**.

- ▶ Can treat **random** coefficients, and **nonlinear** problems: **sampling** and **fitting** then become significant challenges.

References:

- Pavliotis and Stuart, *Multiscale Methods: Averaging and Homogenization*.
- Braides, *Gamma-convergence for Beginners*

An open problem: Predicting plasticity



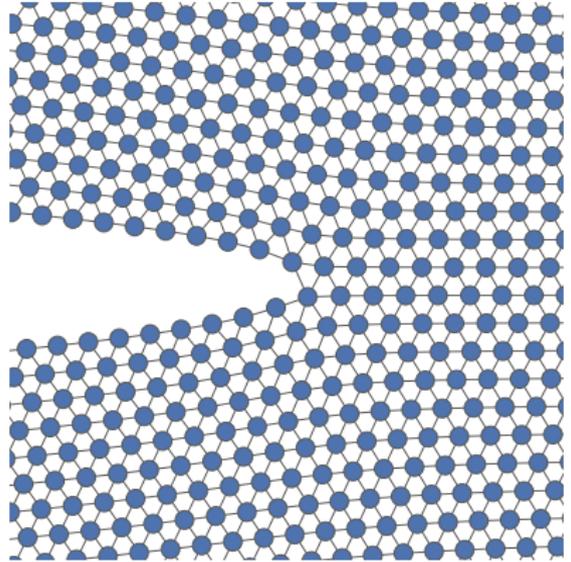
Assistant Professor in Predictive Modelling and Scientific Computing (107274-0323)

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Vacancy Type/Job category	Academic
Department	School of Engineering
Sub Department	Mechanical, Materials and Process
Salary	£44,414 - £52,841 per annum
Location	University of Warwick, Coventry
Vacancy Overview	<p>Full time, permanent position.</p> <p>The School of Engineering is seeking to recruit a talented and enthusiastic Assistant Professor in Predictive Modelling on the traditional research and teaching pathway. We are looking for someone who can help us strengthen and broaden our research activities, while also contributing to teaching of the newly established MSc in Predictive Modelling and Scientific Computing to start in the 2023-2024 academic year.</p> <p>We are open to appointments in any aspect of predictive modelling and scientific computing research consistent with the activities of the Warwick Centre for Predictive Modelling and the facilities available in the School and the University. Those with backgrounds in mathematical and statistical foundations of predictive modelling, or in its application to physics-based uncertainty quantification (for example aligned with existing WCPM strength areas such as continuum solid or fluid mechanics, environmental sustainability, energy, materials science) are particularly encouraged to apply. The appointee will complement existing activity in the School or establish new activity, and it is expected the research will have</p>

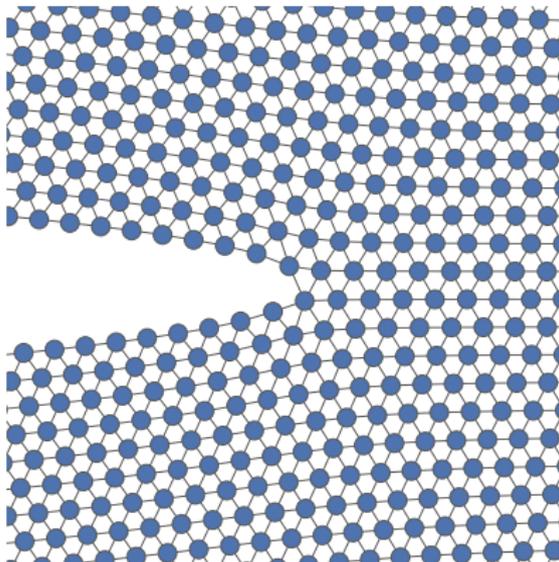
Summary

- ▶ **Multiscale methods** continue to develop in computational Materials Science, connecting and improving models, but mature tools are available.



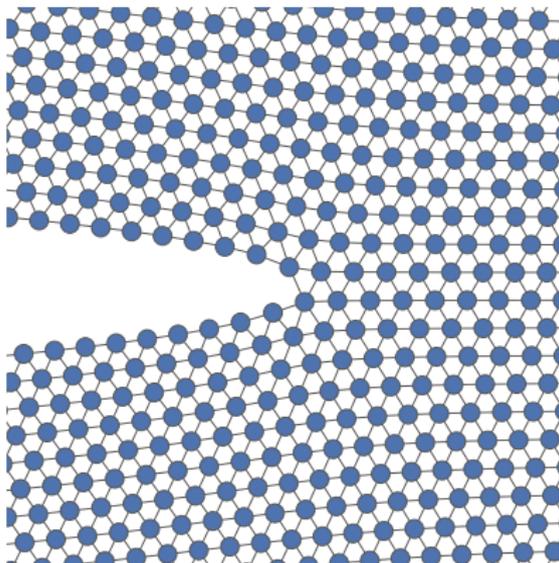
Summary

- ▶ **Multiscale methods** continue to develop in computational Materials Science, connecting and improving models, but mature tools are available.
- ▶ Both **theory development** and **numerical implementation** in robust, scalable code have important roles to play for the future of Materials modelling.



Summary

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References:

- Musil et al, Chem. Rev. 2021, 121, 16, 9759–9815
- Luskin and Ortner, 2013. Acta Numerica, 22, 397-508.
- Ehlacher et al, 2016. Arch. Rational Mech. Anal. 222, 1217–1268.
- Trinkle, 2008. Phys. Rev. B 78, 014110.
- Lelièvre, Rousset and Stoltz, *Free Energy Computations*, Imperial College Press, 2010.
- Pavliotis and Stuart, *Multiscale Methods*, Springer, 2008.
- Braides, *Gamma-convergence for beginners*, Oxford University Press, 2002.