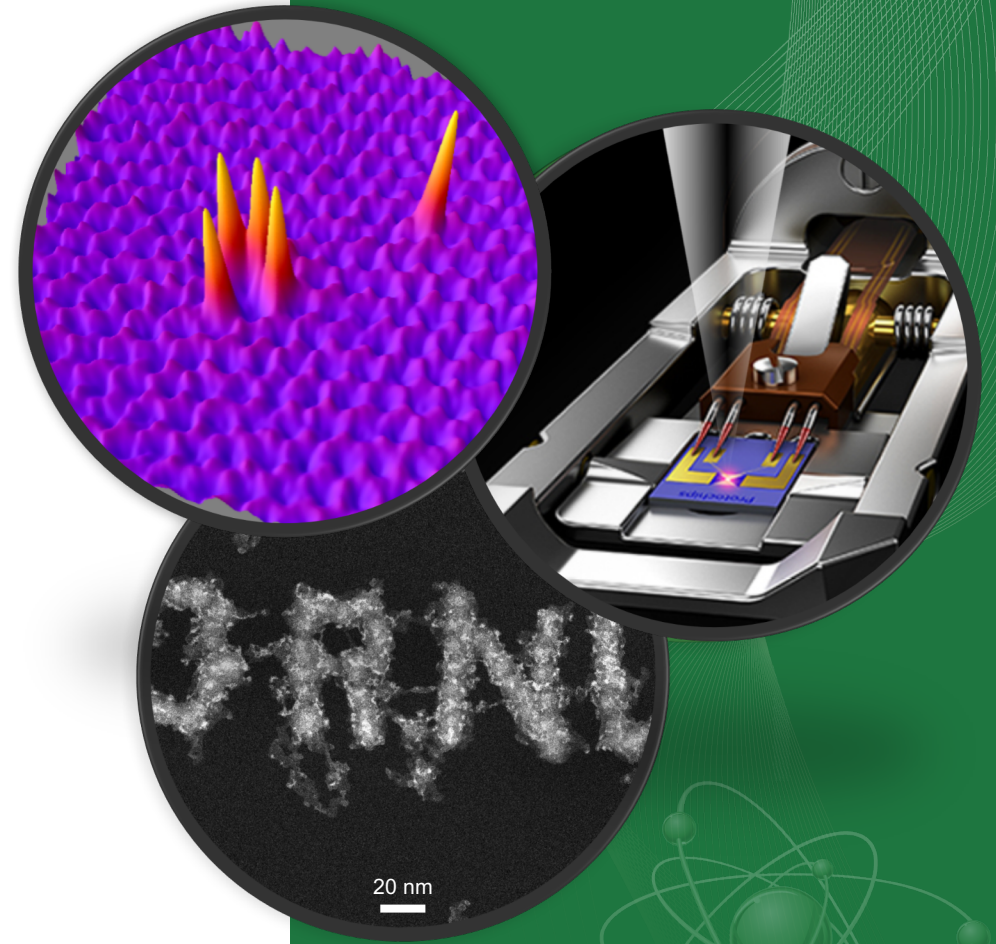


# AI and ML in Microscopy: A Material Opportunity

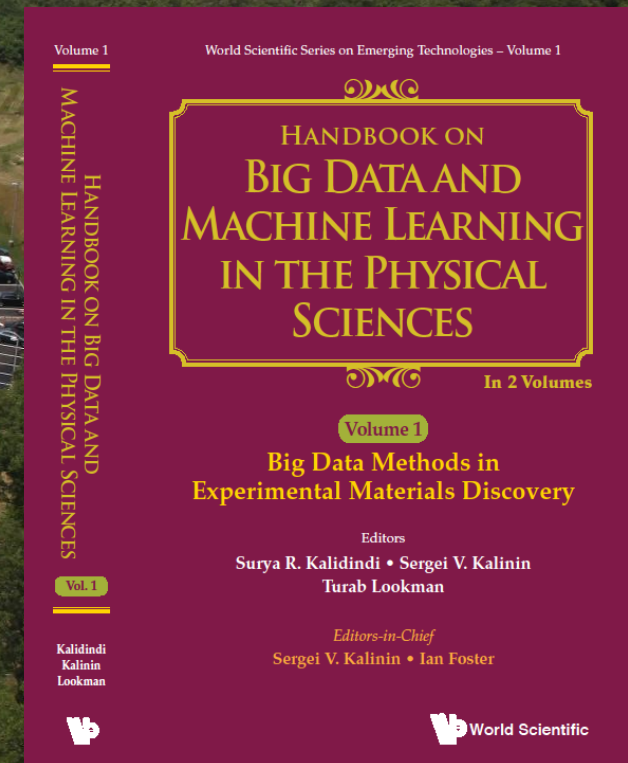
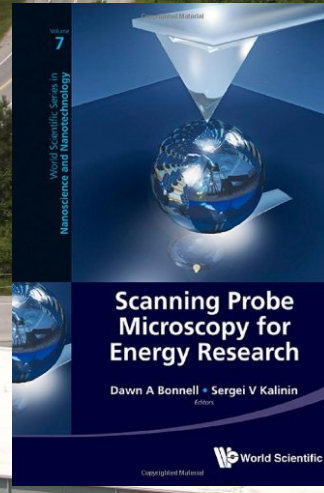
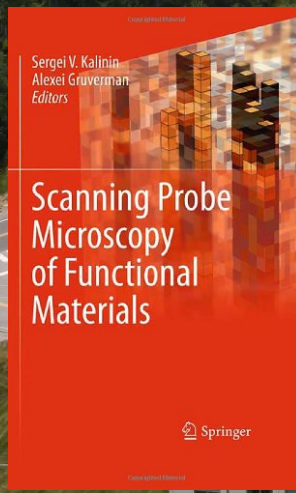
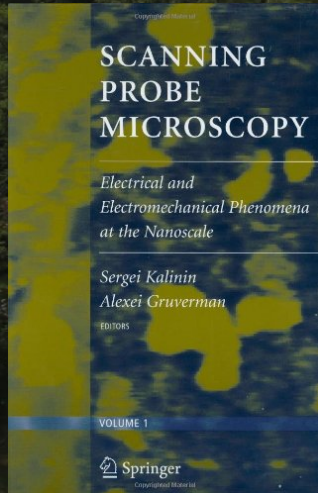
**Sergei V. Kalinin**

The Center for Nanophase Materials Sciences (CNMS),  
Oak Ridge National Laboratory

*October 17, 2019*

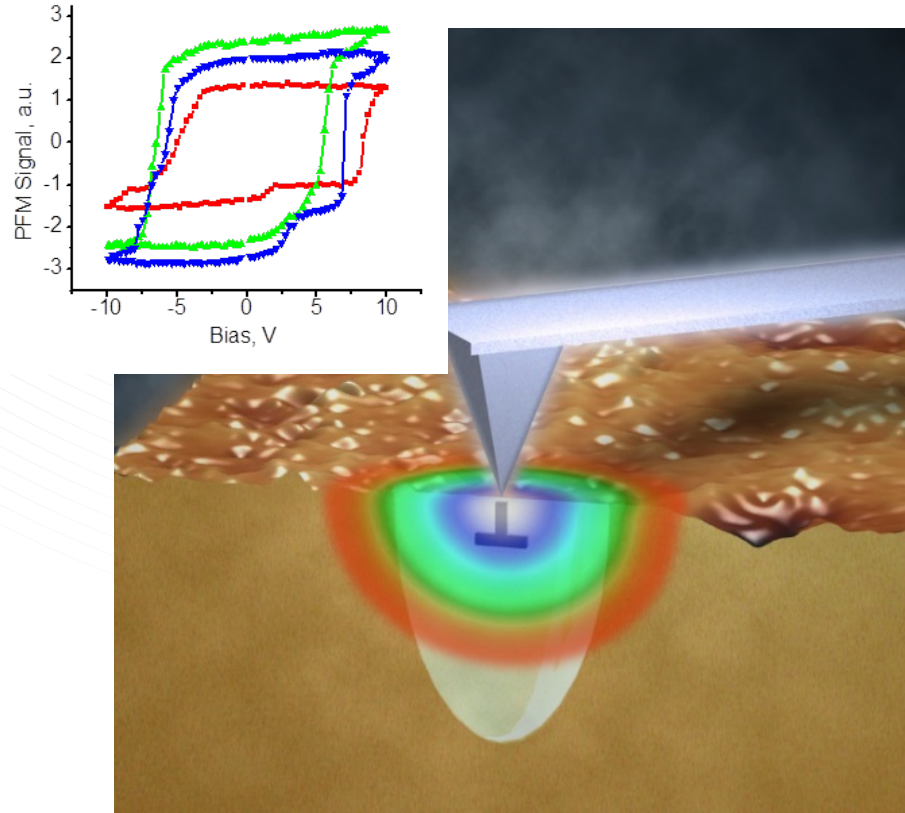








# Multidimensional SPM modes

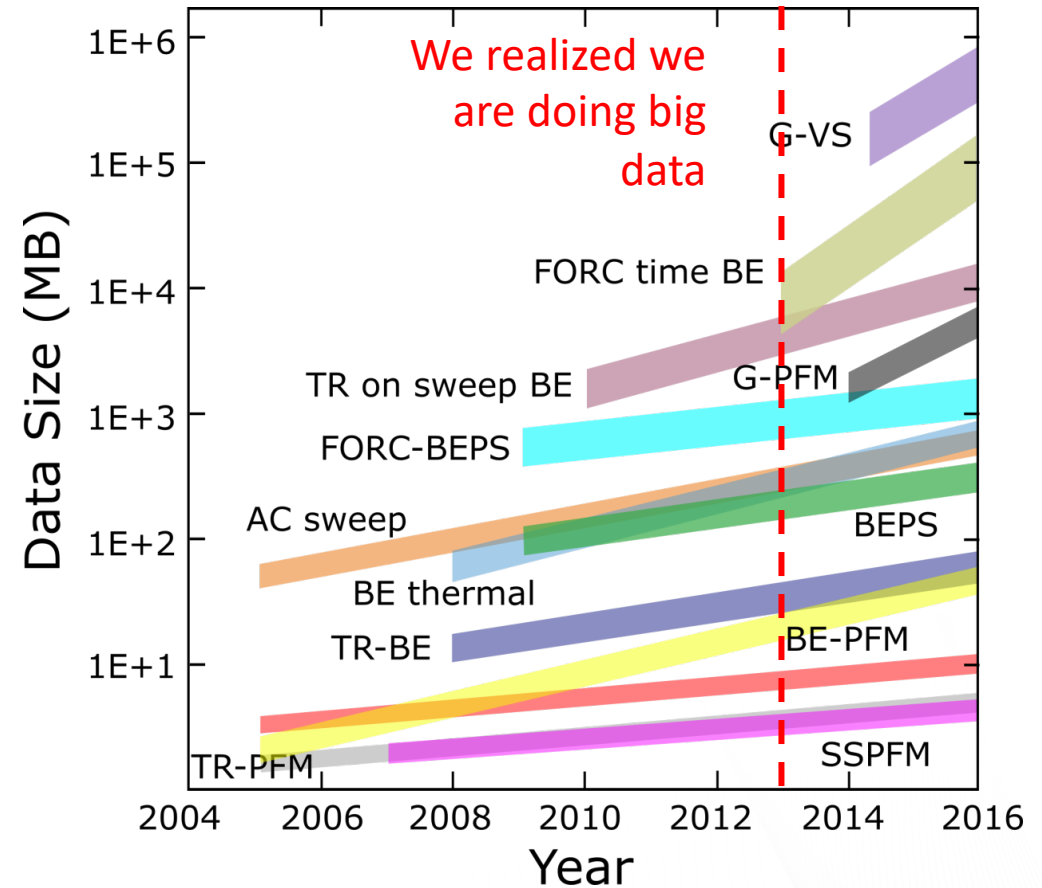


## Physics perspective:

1. Local stimulus: spectroscopic (3D) SPM modes
2. Phase transitions are hysteretic: First Order Reversal Curves
3. Phase transitions can be rate-controlled

## SPM Perspective:

4. SPM requires resonance enhancement (frequency domain)

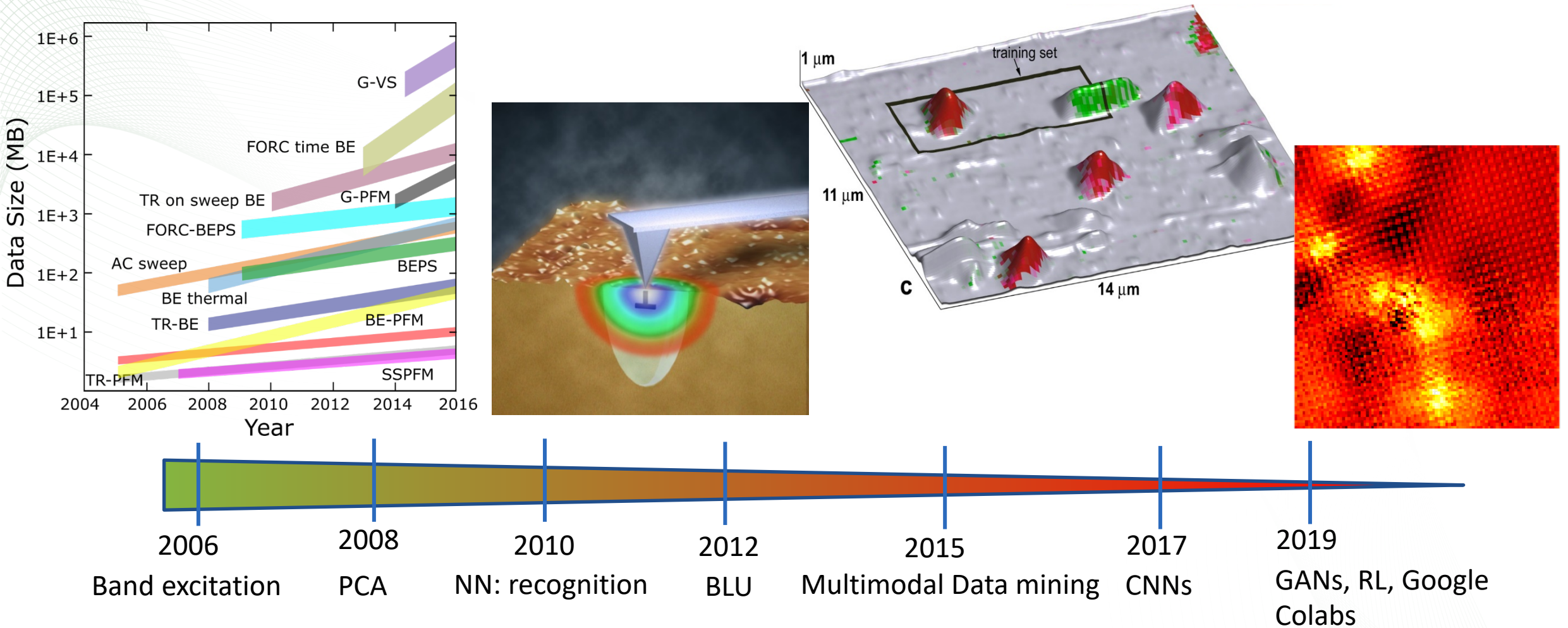


**Instrumental limit:** photodetector bandwidth ( $\sim 10$  MHz) x DAQ performance (32 Bit)

- **Single frequency/heterodyne:** lock-in compression to  $\sim 1$  kHz
- **Band excitation:**  $10^2$  bins at  $\sim 1$  kHz = 100 kHz
- **G-mode:** full streaming at  $\sim 10$  MHz



# We have the data... We need to do something with it!



M.P. NIKIFOROV, A.A. VERTEGEL, V.V. REUKOV, G.L. THOMPSON, S.V. KALININ, and S. JESSE,  
*Functional recognition imaging using artificial neural networks: Applications to rapid cellular  
identification by broadband electromechanical response*, Nanotechnology **20**, 405708 (2009).



# Opportunities in Materials Science

Predicting crystal structure by merging  
data mining with quantum mechanics

CHRISTOPHER C. FISCHER<sup>1</sup>, KEVIN J. TIBBETTS<sup>1</sup>, DANE MORGAN<sup>2</sup> AND GERBRAND CEDER<sup>1\*</sup>

<sup>1</sup>Department of Materials Science and Engineering, Massachusetts Institute of Technology, Cambridge, Massachusetts 02139, USA

<sup>2</sup>Department of Materials Science and Engineering, University of Wisconsin, Madison, Wisconsin 53706, USA

\*e-mail: gceder@mit.edu

- **“Improve”**: Renewable energy, self-driving cars, transparent displays, new memory technologies
- **“Discover”**: Room temperature superconductivity, high mechanical stress materials
- **“Engineer”**: Quantum computing, single-atom catalysts, biomolecules

**Functionality, manufacturability, cost**



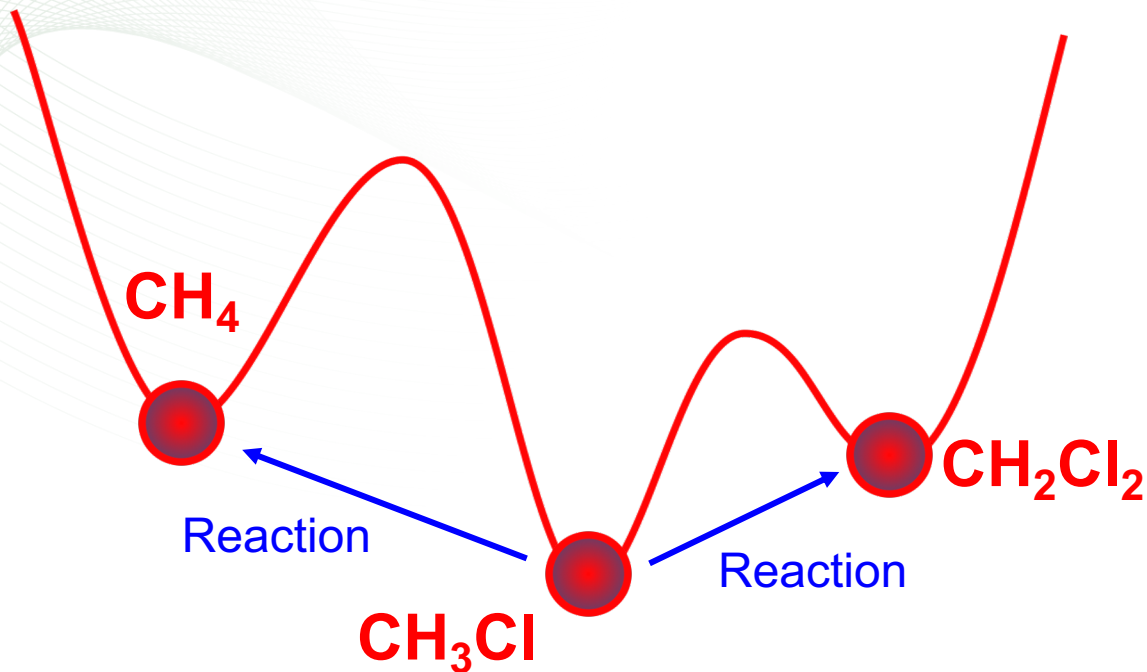
*“New directions in science are launched by new tools much more often than by new concepts. The effect of a concept-driven revolution is to explain old things in new ways. The effect of a tool-driven revolution is to discover new things that have to be explained.”*

*Freeman Dyson*



# Materials in the Chemical Space

- Full atomic coordinate space: un-tractable
- Chemical space: minima corresponding to (meta)stable compounds



- Chemical space is non-differentiable
- The pathways between different regions are non-obvious
- “Useful” functionalities can be very complex and poorly understood
  - We can calculate bandgaps and ideal Young moduli
  - Biological activities and superconductivity, not so much

**We typically need outliers!**

But:

- There are underlying physical laws that determine what is possible

- Materials and molecules are points in chemical space
- Finding “the right” material is then a search/optimization problem.
- Machine learning is great at this. Right?



# Molecular Systems:

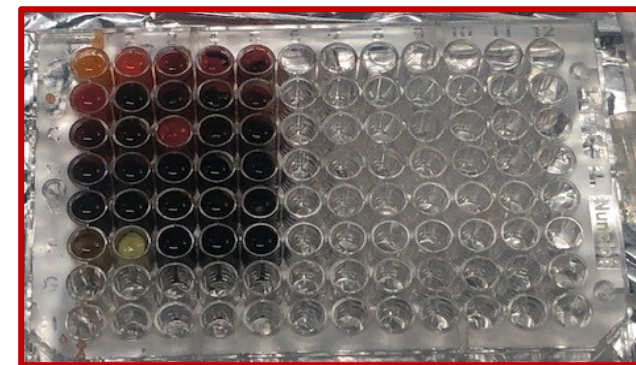
M. Ceriotti

Opentrons

- Chemical space is a graph:  $10^{63}$  (only less than 30 atoms)
- Edges (reactions) can be optimized individually
- Amenable to the literature mining (*Chematica*)
- Area of known compounds can be expanded via retrosynthesis
- Amenable to laboratory robotics

**But: functionalities at the nodes are defined within the context  
(CO<sub>2</sub> adsorption, water desalination, biological functionality)**

Chematica (B. Grzbowski)



M. Ahmadi

Biology took  $10^9$  years for search: MCMC with  $\sim 10^{20}$  steps and  $\sim 10^{35-40}$  chains



S. Curtarolo

Let's think about it as a search problem:

- **Alloying:** need maintain composition  $\sim 1\%$
- **Doping:** need maintain composition  $\sim 10^{-6}$
- Grid search is out for  $D > 3$  (experiment)
- MCMC type problem: how do we make it work?

Discovery of copper HTSC,  $\text{MgB}_2$ , iron pnictides. Serendipitous discovery followed by exploring large families. Could not predict – no theory

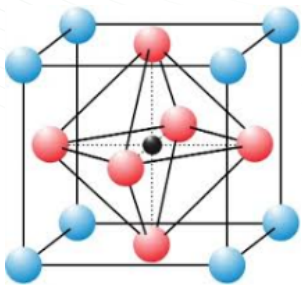


# The Solid Problem

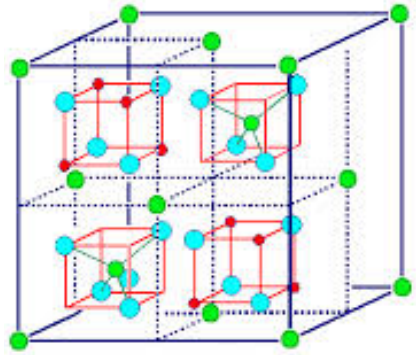
Chemical space is heavily degenerate: can use mean field descriptors

- *Symmetry*,
- *Concentrations*,
- *Order parameters*
- *Thermodynamic potentials*

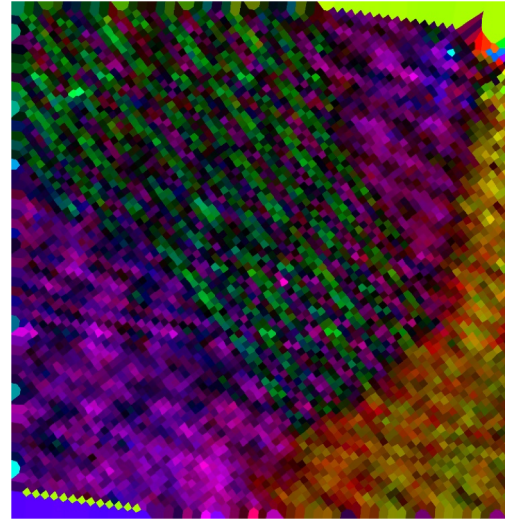
Simple unit cell



Complex unit cell:  
partial occupancies

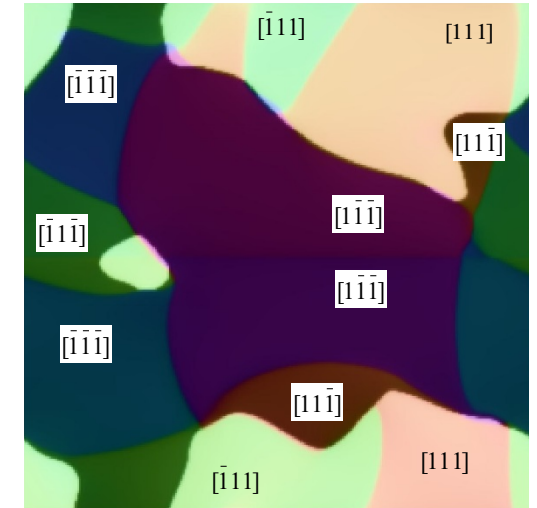


Nanoscale phase  
separation



C. Nelson

Mesoscale:  
microstructures



L.Q. Chen

- Really interesting materials are those when it is not the case (relaxors, Kitaev materials) – some correlations and disorder in ground state.
- Many physical properties cannot be predicted. Bandgaps ok, superconductivity not so much
- Large scale organization: defects, microstructures, etc.

How do we start describing structures of solids beyond symmetry-based methods, establish structure-property relationships, and use this information for prediction of materials and discovery of synthesis pathways?

# More than atoms

## Hasten high resolution

Build precision microscopes to map atoms, say  
**Stephen J. Pennycook and Sergei V. Kalinin.**

Nature 515, 487 (2014)

J.J. Guo et al.,  
Nat. Comm. 5, 5389 (2014)

Atomic positions can be  
determined to <10-pm  
precision

Bond length: Chemical  
reactivity, catalytic  
activity

Bond angles:  
Magnetism  
and transport

Configurations  
and repeating  
elements?



# The Lab on a Beam

Image  
atomic  
columns

Spectra from  
single atoms

Image light atoms  
and sensitive  
materials

Diffraction from  
subatomic  
volumes

Beams with  
orbital  
momentum

Beam  
manipulation

Vortex beams

High-res EELS

In-situ holders

4D STEM

Physics extraction

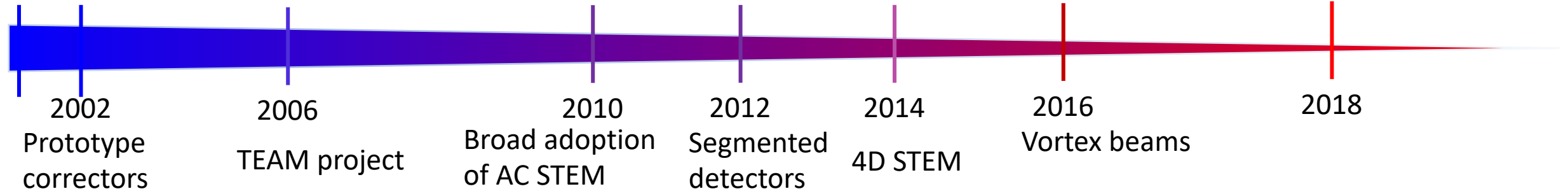
Atomic manipulation

Sample  
Source  
Detector  
Data  
Feedback

0.61 Å resolution

First AC (Nion)

1997



**Challenge 1:**  
**Data Infrastructure**

**Standard image:** ~10 MB

**Dynamic data:** ~ 100 MB/S

**4D STEM:** ~TB data (limited by the availability of data infrastructure)

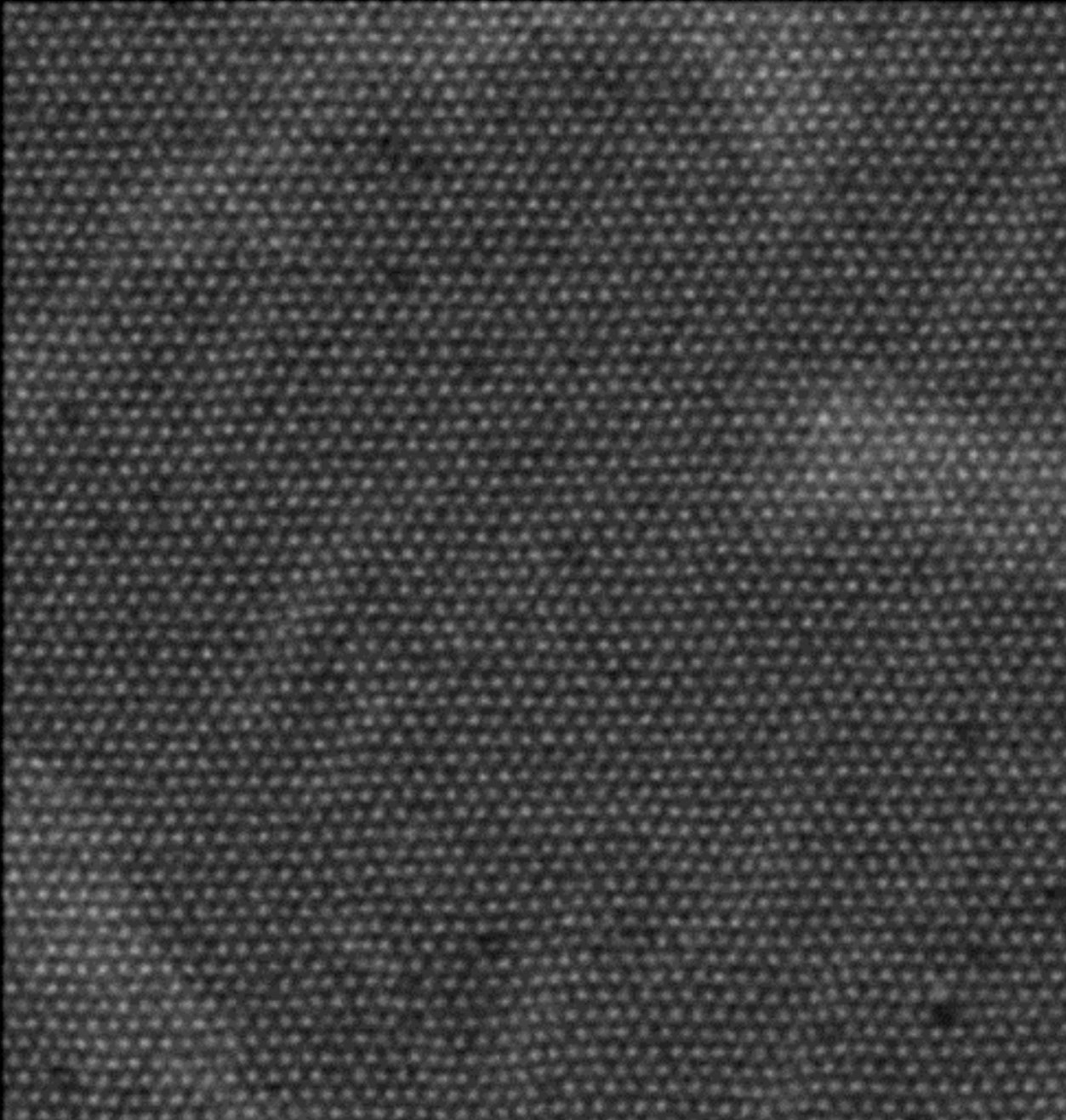
**Potentially:** Large Hadron Collider level data flows (from single microscope)

# Physics from STEM and SPM data

- **Can we get materials specific information** (e.g. atomic coordinates from STEM, scattering potentials from 4D STEM, etc.) **from microscopy data**, at which level of confidence, and **how this knowledge is affected and can be improved from knowledge of imaging system** (e.g. classical beam parameters, resolution function, all the way to full imaging system modelling), and **knowledge of material**.
- **Can we use the materials-specific information** with uncertainties determined by incomplete knowledge of imaging system or intrinsic limitations **to infer physics and chemistry, either via correlative models or recovery of generative physics** (force fields, exchange integrals, etc.)
- **Can we use thus determined materials information**, either correlative or causative, **to reconstruct materials behaviors** (phase diagrams, etc) in the broader parameter space (e.g. for temperatures and concentrations different for specific sample studied), and determine how reliability of such prediction depends on position in parameter space.
- Can we harness the data stream from the microscope to engender real time feedback, e.g. for **autonomous experimentation** and **atomic manipulation**



# Dynamic Atomic Changes



Observations of atomic dynamics induced by beam (or temperature, field, etc.) gives information on multiple atomic configurations as they form and evolve.

Can we learn:

- Effective interaction parameters (e-ph coupling)?
- Force fields?
- Phase transition dynamics

# Deep learning for atomically resolved images

## Top 3 predictions

Staffordshire bullterrier 43 %

American pit bull terrier 23 %

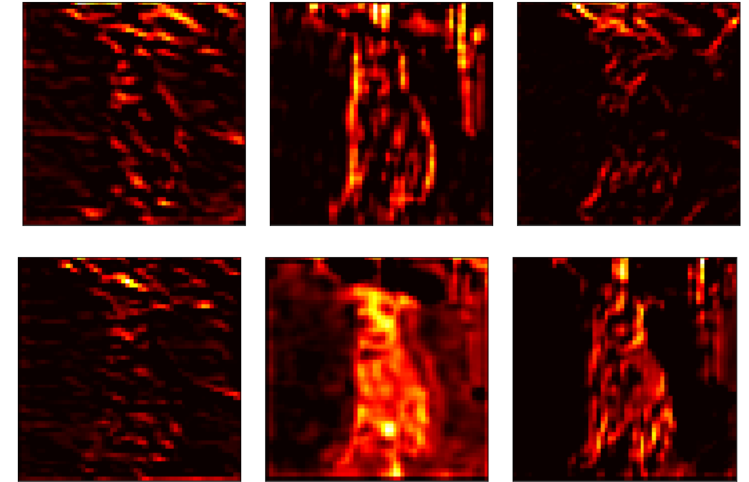
Basenji 11 %

*Very  
close*

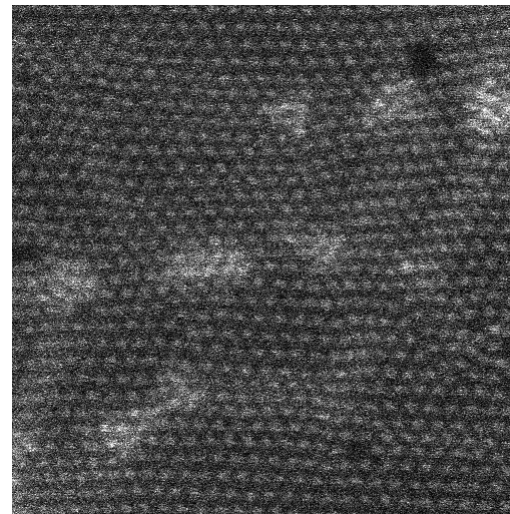
**Meet Duffy:** Pitbull-Shepard-Collie mix



Randomly selected feature maps



**STEM of WS2**

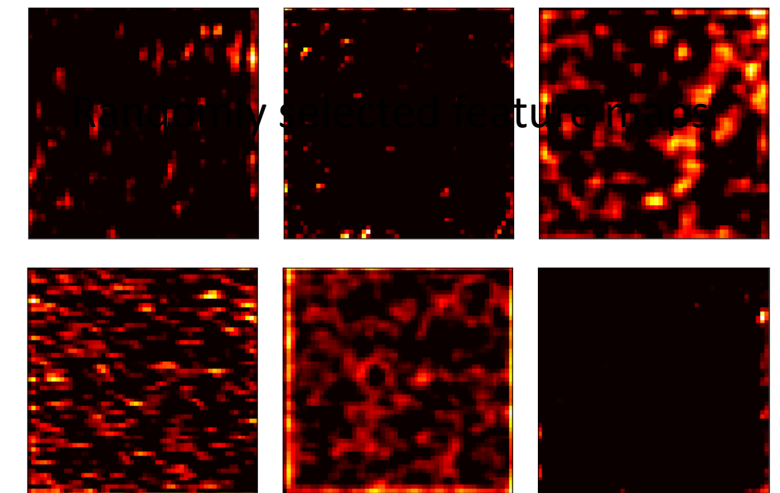


## Top 3 predictions

Wool 6.3 %

Velvet 5.5 %

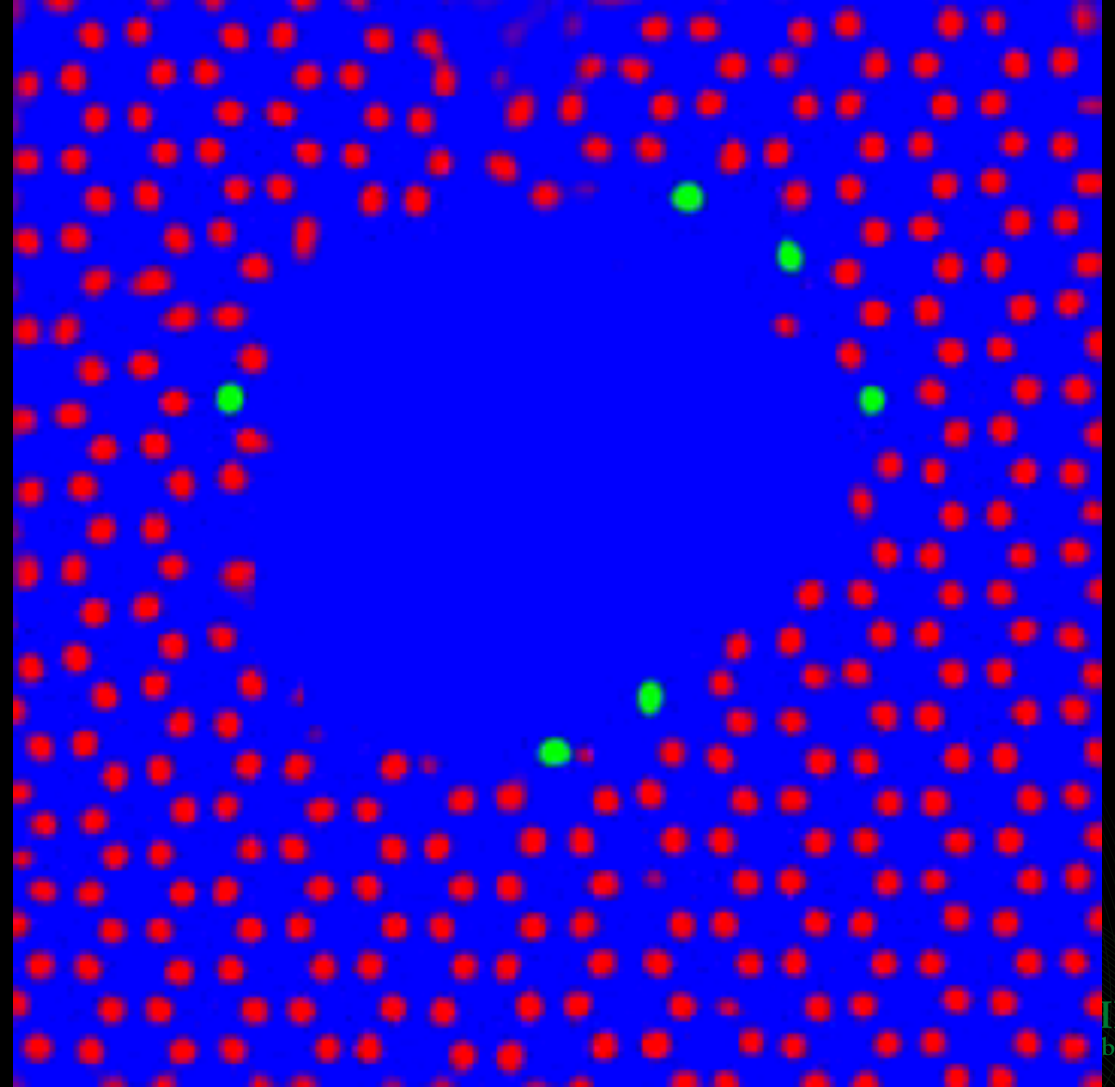
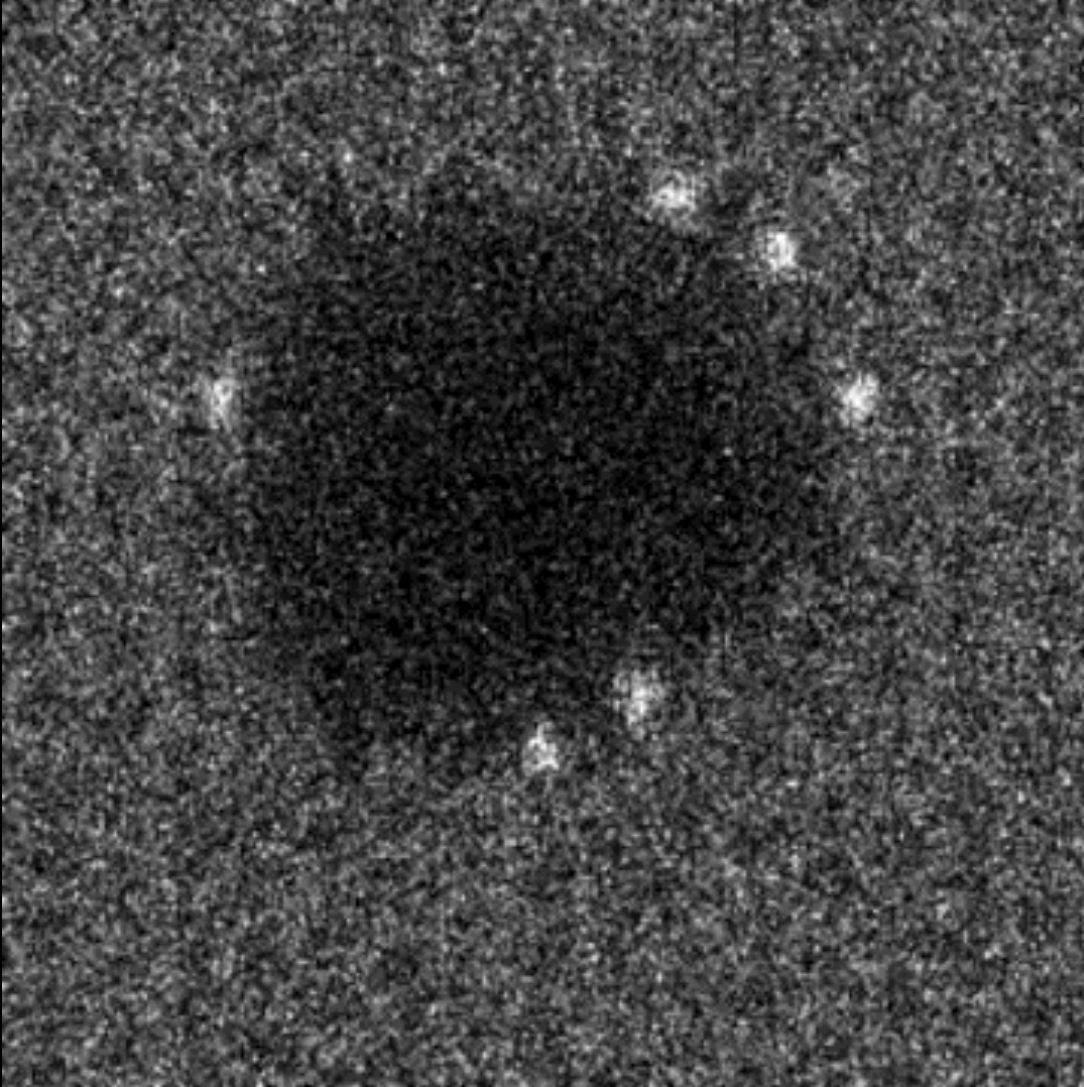
Window screen 3.8 %





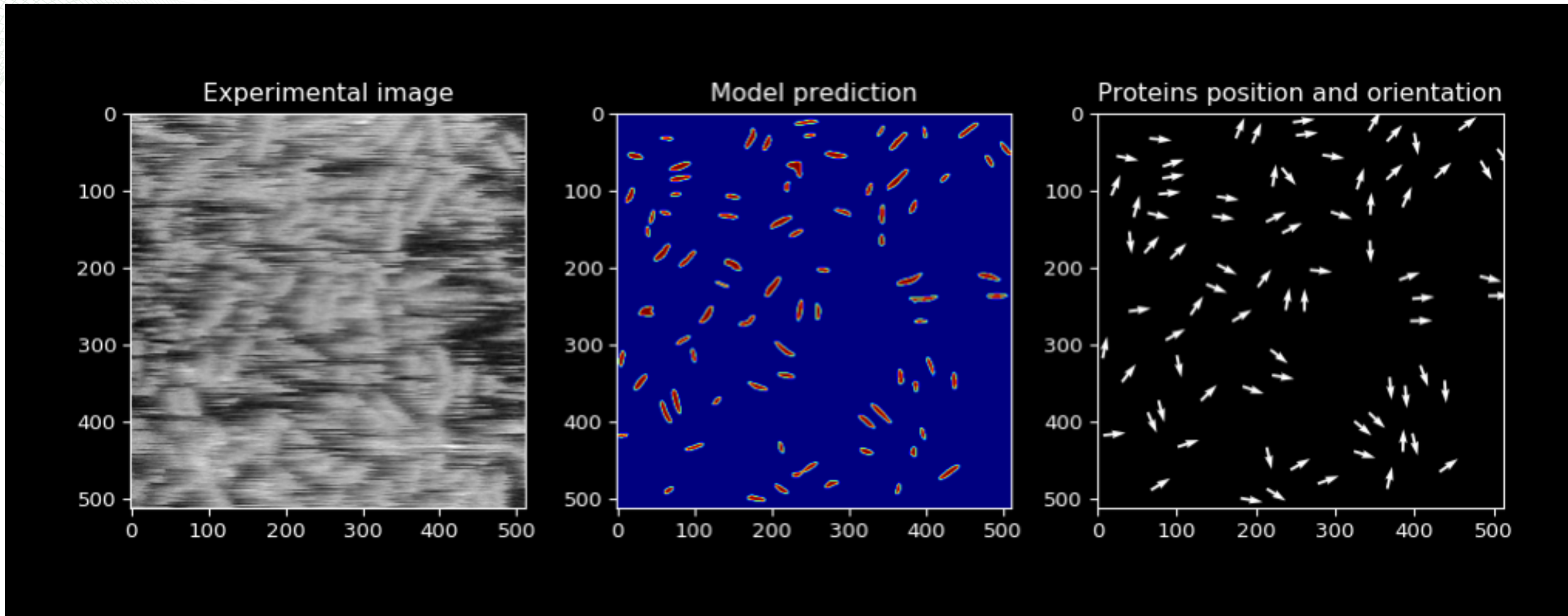
## Deep learning works like a charm for:

- Drift correction
- Denoising
- Data processing/dimensionality reduction
- Feature finding (physics is in the training set)



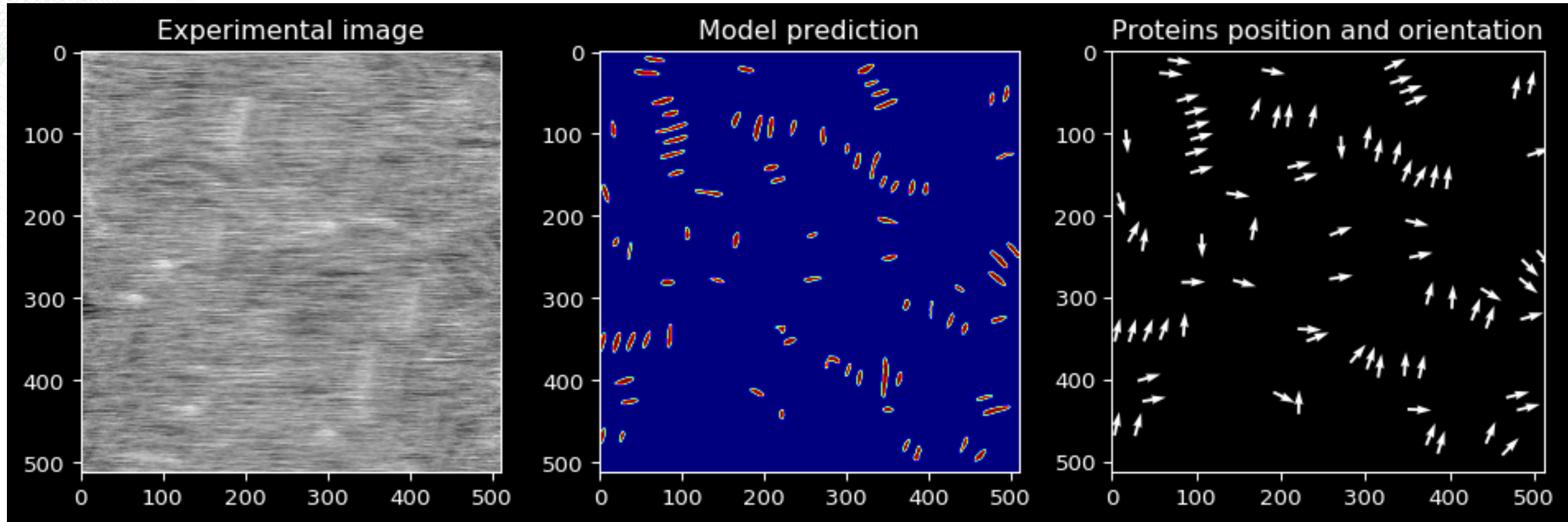
# Deep learning in AFM

Model trained on a single movie frame from the well-ordered phase and applied to the entire movie



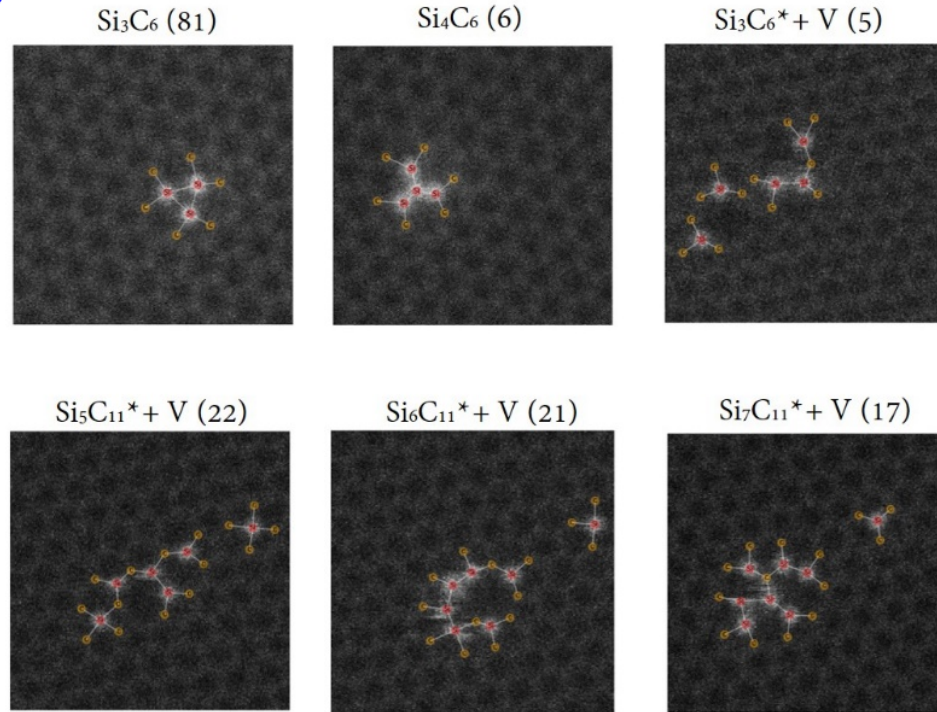
# Deep learning in AFM

Model trained on a single movie frame from the well-ordered phase and applied to the entire movie



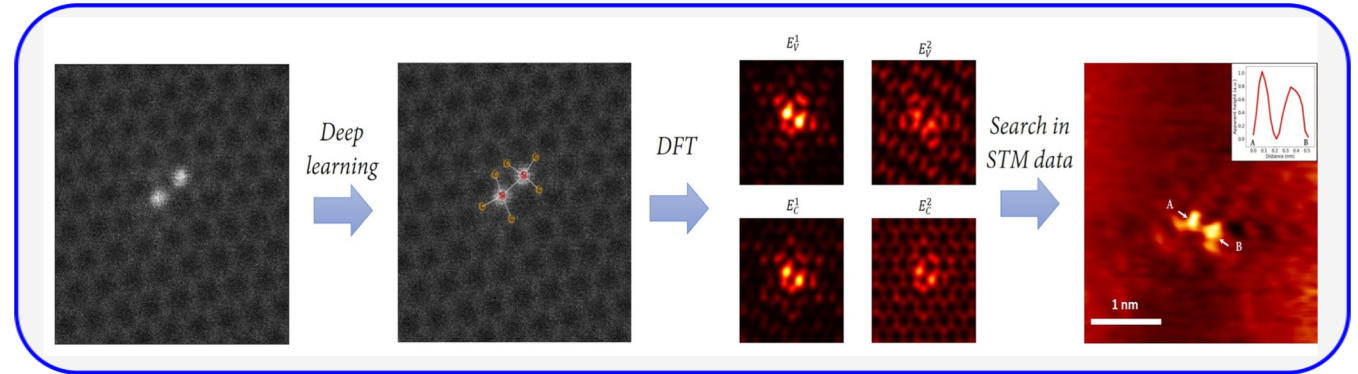


# Defect Libraries



- Identify the “real” defects
- Theory to get energies, strains, etc

## Analysis workflows



## Open data

CITRINE INFORMATICS

Si-Vacancy complexes in Graphene

ID: 164210 - Version 4

Description:

Scanning transmission electron microscopy of Si-vacancy complexes in monolayer graphene. Dynamic electron beam-induced transformations are used to create libraries of the possible Si and carbon vacancy defects in graphene monolayer. Automated image analysis and recognition based on deep learning networks is developed to identify and categorize the defects, creating a library of (meta) stable defect configurations. Prepared by Maxim Zlatilov, Ondrej Dyck, Sergei V. Kalinin, and Bobby G. Sumpter at the Oak Ridge National Laboratory. Cite this dataset: <https://doi.org/10.25920/ox3-8459>

Search this dataset

Material Name or Chemical Formula

Property Name

Units

Advanced Search Options

Search

Showing results 1 to 13 of 13

Si <sub>2</sub> C <sub>4</sub> +V	Name: Si <sub>2</sub> C <sub>4</sub> +V scan size (nm): 2.0
Si <sub>1</sub> C <sub>3</sub>	Name: Si <sub>1</sub> C <sub>3</sub> scan size (nm): 2.07
Si <sub>1</sub> C <sub>3</sub> +VT	Name: Si <sub>1</sub> C <sub>3</sub> +VT scan size (nm): 2.0
Si <sub>1</sub> C <sub>4</sub>	Name: Si <sub>1</sub> C <sub>4</sub> scan size (nm): 2.2

## Open code: Jupyter papers

Building Ferroelectric from the Bottom Up: The Machine Learning Analysis of the Atomic-Scale Ferroelectric Distortions

Project published on 18.04.2019, 15:28 and posted on 18.04.2019, 15:30 by Maxim Zlatilov, Christopher Nelson, Steve Bradburn, Dong Chen, Sergei Kalinin

Recent advances in scanning transmission electron microscopy (STEM) have enabled direct visualization of the atomic structure of ferroic materials, enabling the determination of atomic column positions with  $\sim 1$  Å precision. This, in turn, enabled direct mapping of ferroelectric and ferroelastic order parameter fields via the top-down approach, where the atomic coordinates are directly mapped in the microscopic order parameters. Here, we explore the alternative bottom-up approach, where the atomic coordinates derived from the STEM images are used to explore the related atomic displacement patterns in the material and build the collection of the building blocks for the distorted lattice. This approach is illustrated for the La doped BiFeO<sub>3</sub> system.

The full analysis procedure is available as an interactive paper in a form of a Google Colab (Jupyter) notebook where a classic paper organization is augmented with code cells that appear hidden by default (when viewed in Google Colab). This should allow a reader to interact with the analysis and, more importantly, it enables the readers to use the same codes for their data. The same paper is also available in a standard pdf format (without code).

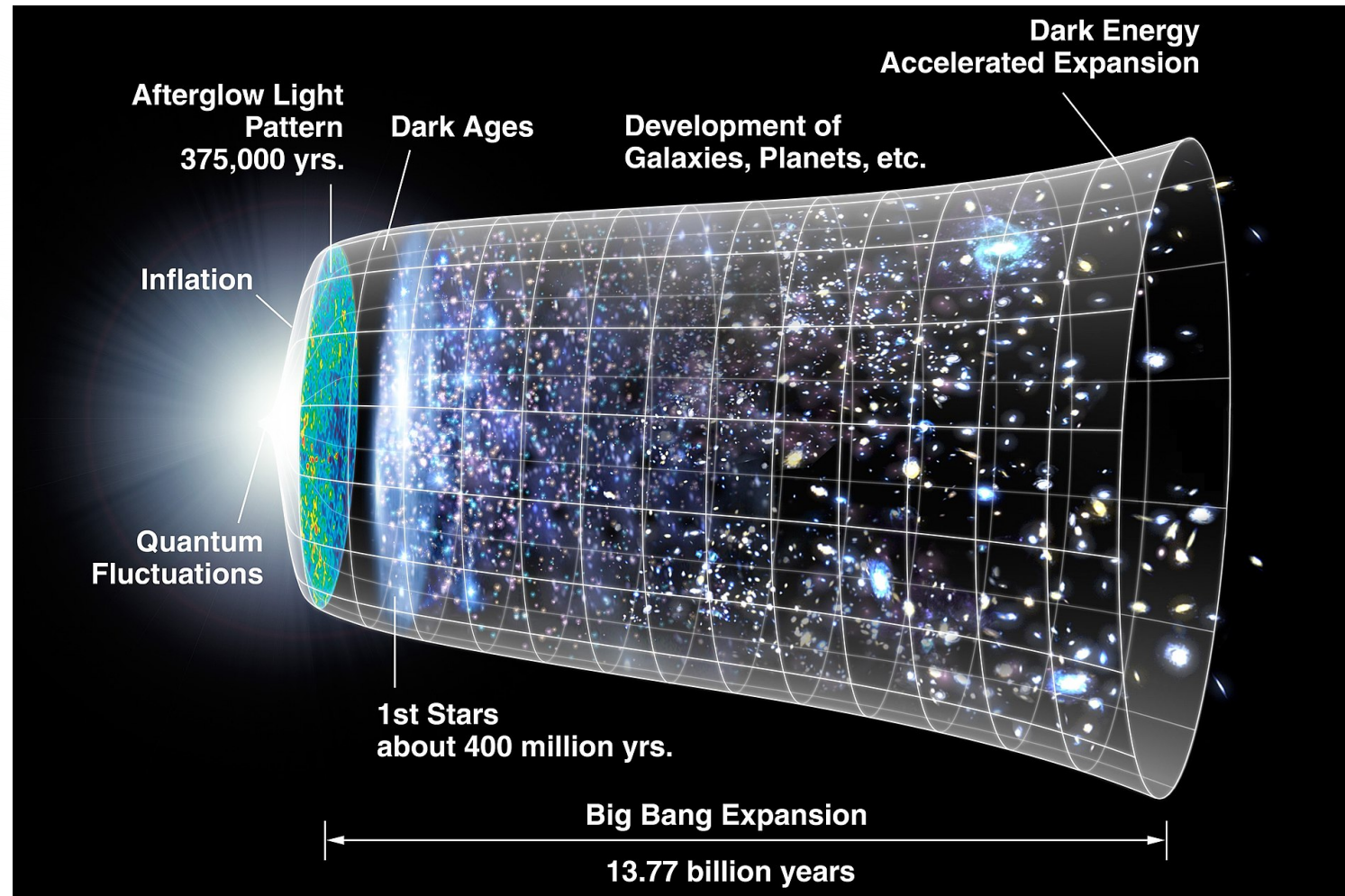
CATEGORIES:

- Materials Science: Ferroelectricity
- Machine Learning
- Physics

KEYWORDS:

- Machine Learning
- Materials Science: Ferroelectricity
- Machine Learning

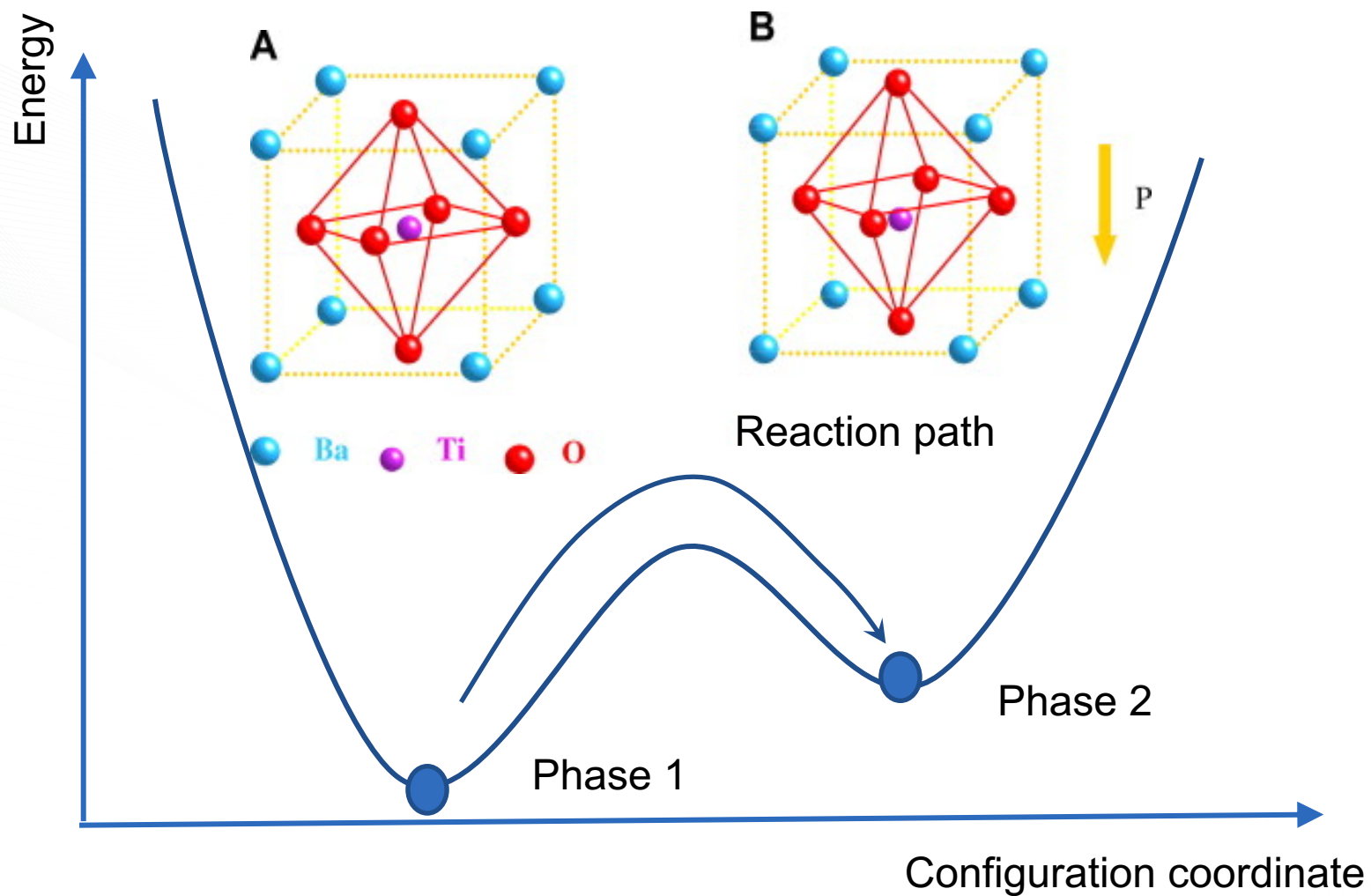
# From Correlation to Causation:



Wikipedia, NASA

- Problems can be intractable combinatorically, but have simple constitutive laws
- Low dimensional non-linear manifold in the very high dimensional linear spaces
- Observational data: astronomy



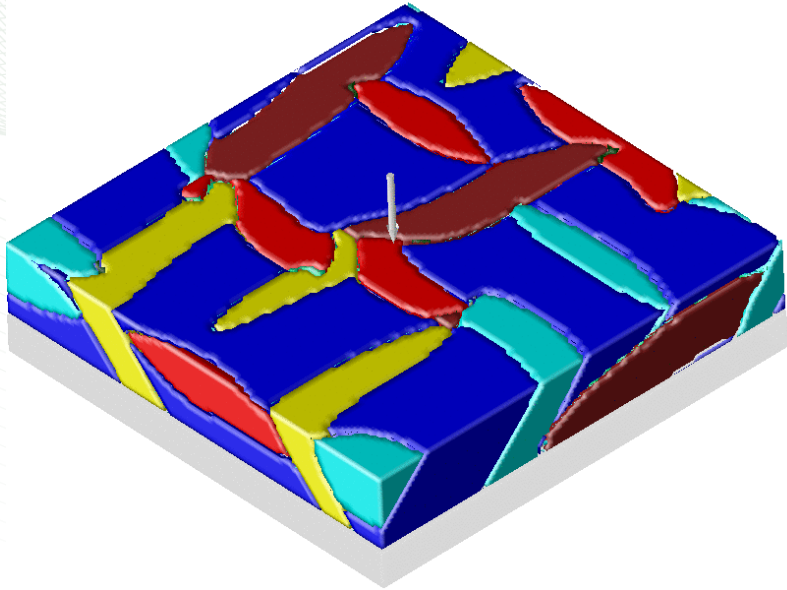


### Problems:

- There are  $10^{(10 - 23)}$  degrees of freedom
- Which (in most materials) correspond to very small number of collective variables (order parameters)
- Naturally, materials where this is likely not the case are really interesting



# Physics and chemistry from structural STEM data?



LGD equation:

$$F = \int_V \left[ f_{\text{bulk}}(P_i) + f_{\text{grad}}(P_{i,j}) + f_{\text{elas}}(P_i, \varepsilon_{kl}) + f_{\text{elec}}(P_i, E_i) \right] dV$$

LGD  
equation:

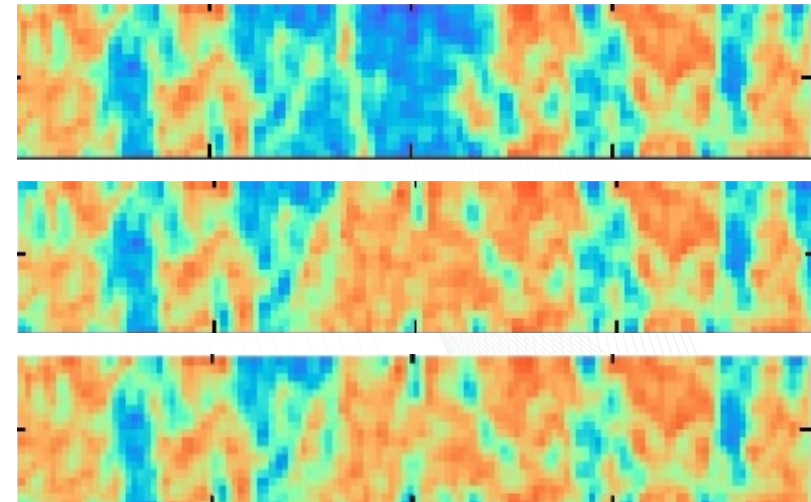
$$\frac{dP_i}{dt} = -L \frac{\delta F}{\delta P_i}$$

Conventional fixed  $\phi$  b.c.

$$\phi|_{z=L+\lambda} = V_{\text{planar}}$$

$$\left. \frac{\partial P_z}{\partial z} \right|_{z=0, L+\lambda} = 0$$

Disordered states



On mesoscale, materials functionalities can often be described via order parameter fields:

- Mesoscopic order parameter is often known from macroscopic measurements
- What are the boundary conditions at surfaces and interfaces
- What are the roles of defects?
- Can we describe spatially inhomogeneous states (relaxors, charge ordered materials, MPB systems)?

# Building the mesoscopic picture top down

**Model System:** Lanthanum-strontium cobaltite with topological defects and interfaces

## Antiphase domain boundary

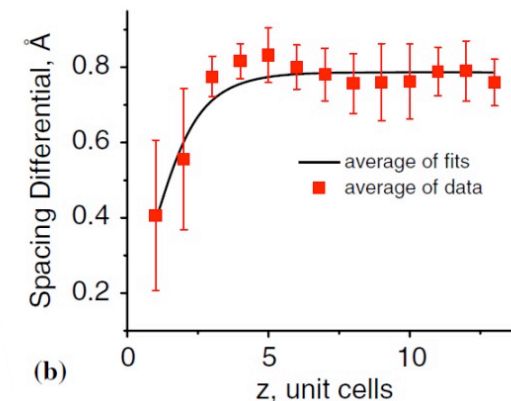
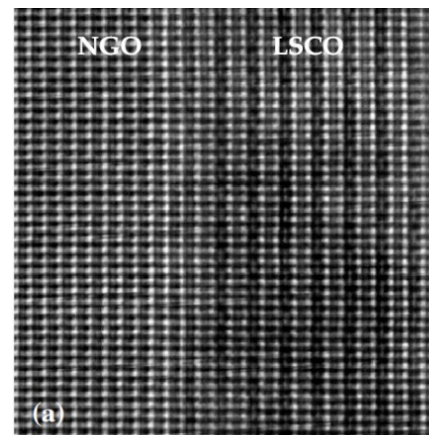
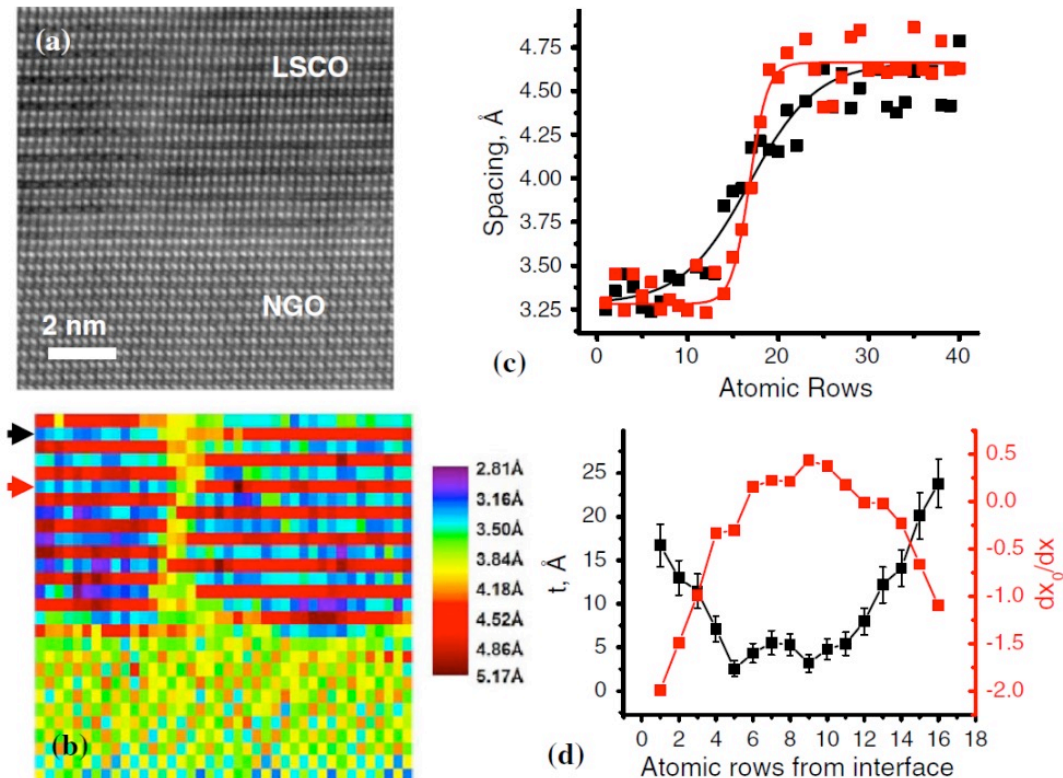
$$\eta(x, z) \approx \eta_{SD}(z) \tanh\left(\frac{x - x_0}{L_C(z)}\right)$$

Fitting the experimentally observed atomic profiles to the functional form of order parameter

$$\eta_{SD}(z) \approx \eta_b \left(1 - \frac{1}{1 + \sqrt{2}\lambda/L_C(0)} \exp\left(-\frac{\sqrt{2}z}{L_C(0)}\right)\right)$$

## Ordering behavior at the interfaces

$$\eta(z) = 0.787(1 - 0.5 \exp(-\frac{z-1}{1.1}))$$

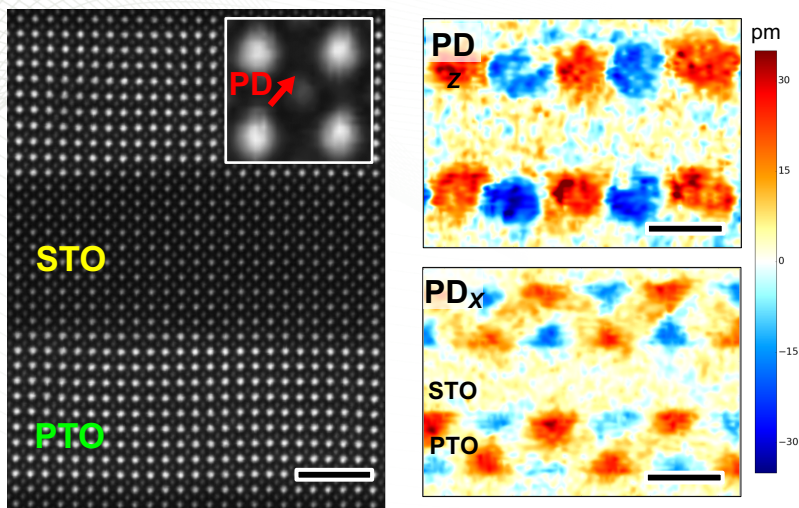


Allows to analyze the interplay between ordering, chemical composition, and mechanical effects at domain walls, interfaces and structural defects

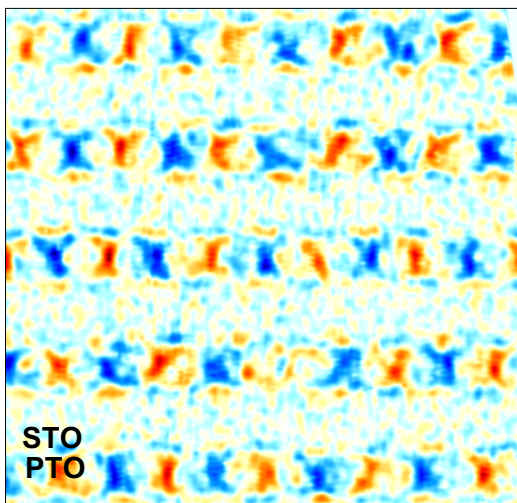


# Flexoelectricity by Computer vision

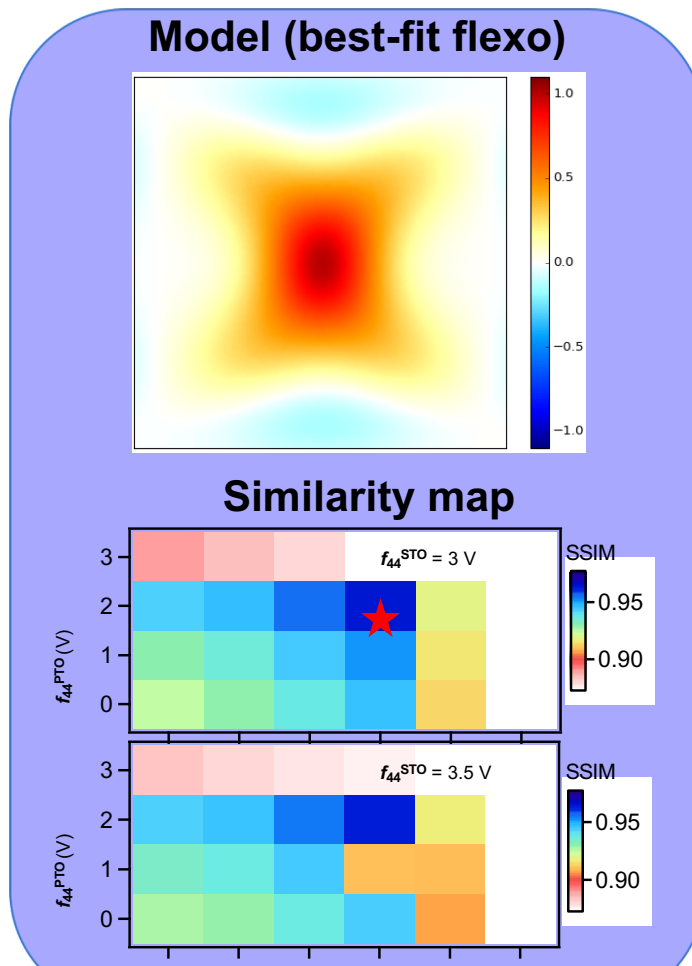
HAADF-STEM images



Vorticity/polar gradient



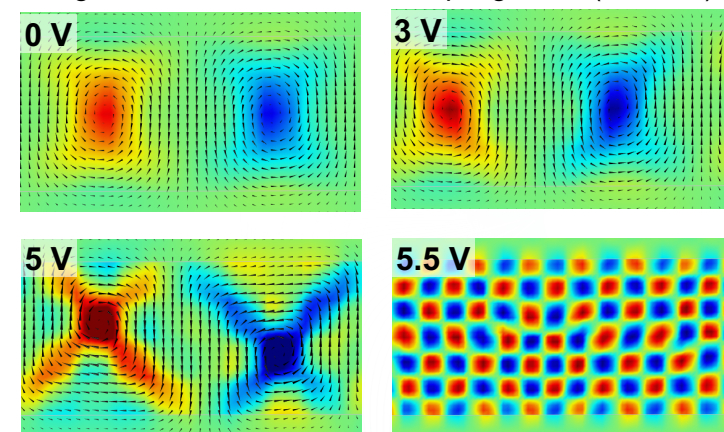
Theory-experiment matching



Effect of flexoelectricity

$$F = \alpha_{ij} P_i P_j + \alpha_{ijkl} P_i P_j P_k P_l + \alpha_{ijklmn} P_i P_j P_k P_l P_m P_n + \frac{1}{2} g_{ijkl} P_{i,j} P_{k,l} + \frac{1}{2} c_{ijkl} \varepsilon_{ij} \varepsilon_{kl} - q_{ijkl} \varepsilon_{ij} P_k P_l - \frac{1}{2} \kappa_0 E_i E_i - E_i P_i + \boxed{f_{ijkl} (P_{k,l} \varepsilon_{ij} - \varepsilon_{ij,l} P_k)}$$

Longitudinal/transverse flexocoupling  $f_{11}/f_{12}$  ( $f_{11} \sim -f_{12}$ )



*The flexocoupling in both PTO and STO layers is considered, revealing different modulation effects.*

Li et al., Nat. Comm. 2017

- Polarization and tilt behavior at interfaces and topological defects: gradient terms and physical BCs
- Coupling with (electro)chemical boundary conditions
- Defect effects: transition from localized perturbation to collective responses

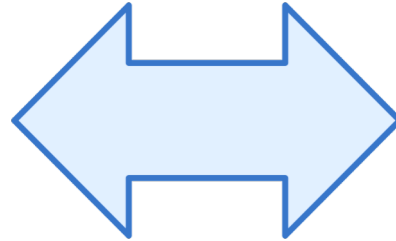


# Physics from Microscopic Degrees of Freedom

## Macroscopic observables:

- Property
- Average structure
- fluctuations

## Macroscopic measurements Scattering data

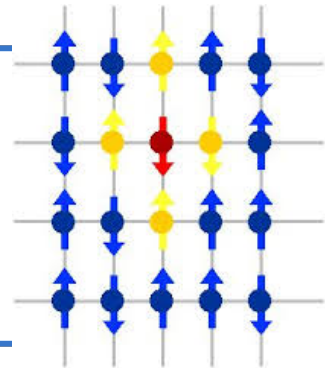


Can ML do it?  
(Melko 2017)



## Microscopic models

- Ising model
- Heisenberg model
- Kitaev model
- ....



## Real material



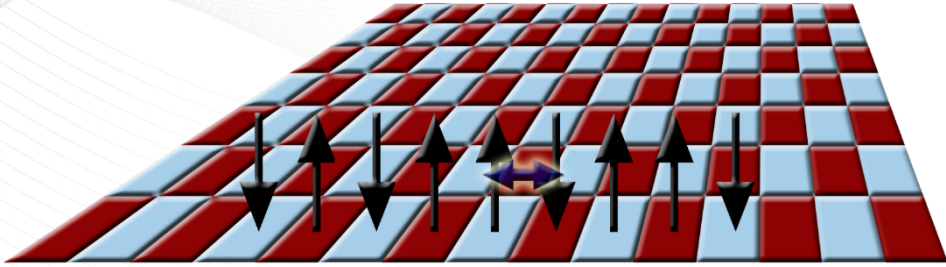
A. Sefat

**On atomistic scale, we often use lattice Hamiltonian models:**

- Can we determine local interactions from STEM or SPM data
- What if some information is lost ( $1 \ll \text{observables} \ll \text{degrees of freedom}$ )

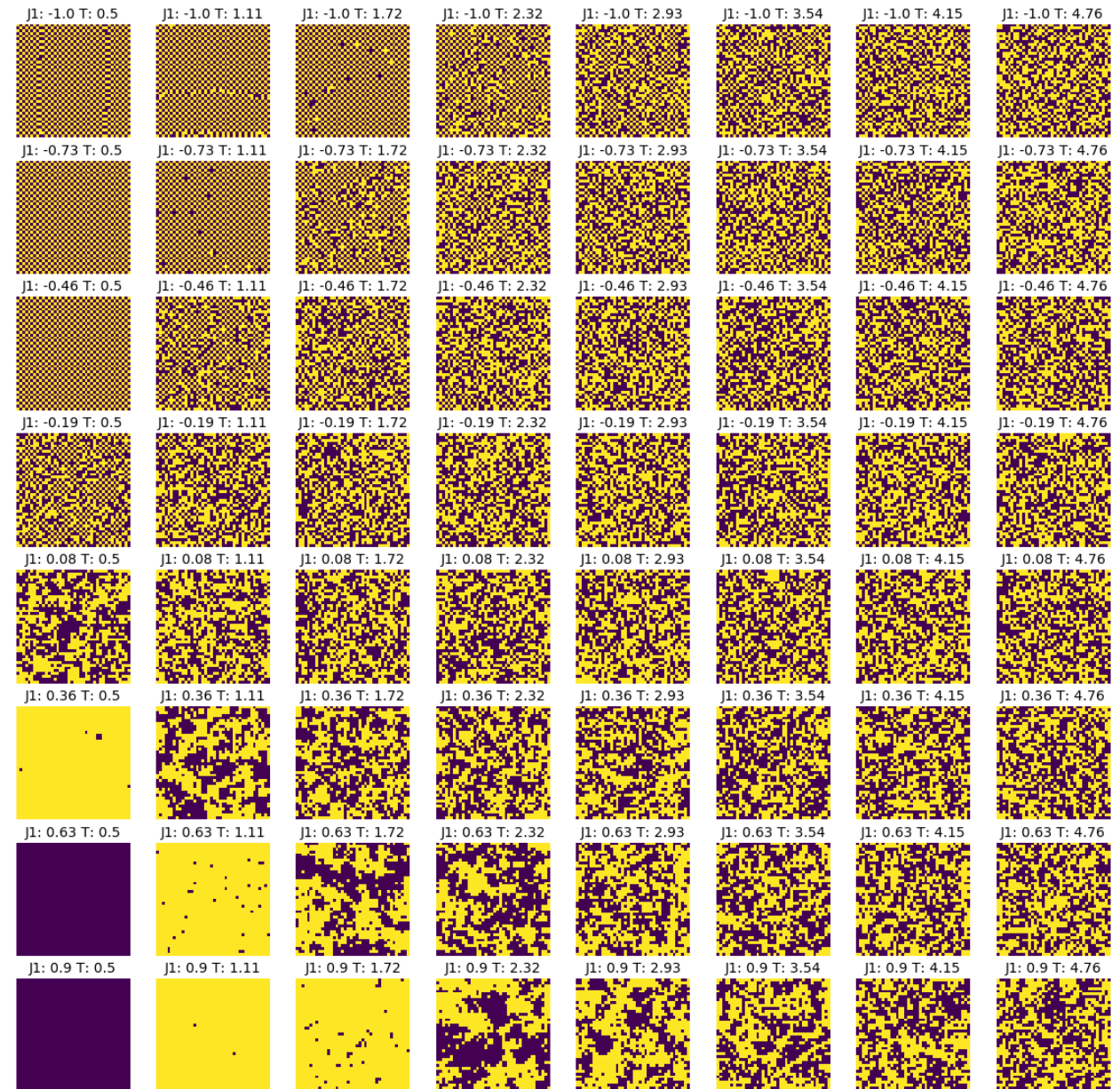
# Can we study physics and chemistry locally?

## Ising model



$$H(H) = \sum_{i,j} J_{ij} S_i S_j + \sum_i (h_i + H) S_i$$

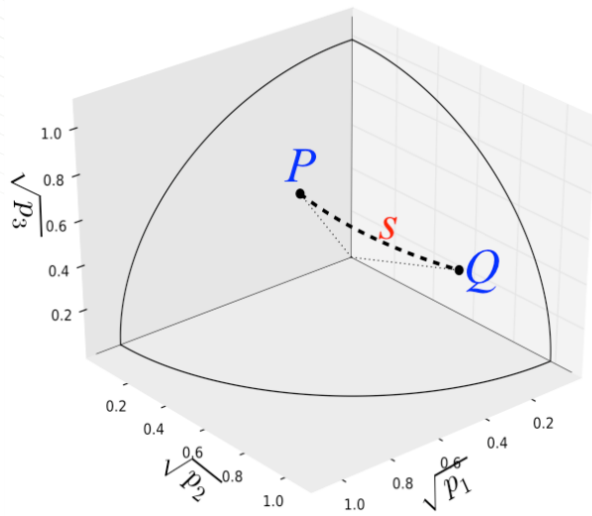
- Collection of two-state “spins” on a geometric lattice
- Interactions with nearest neighbors
- No long range depolarization fields
- Universal model that can represent physical (magnetism), chemical (alloys, surface adsorbates) systems



# Statistical distance minimization

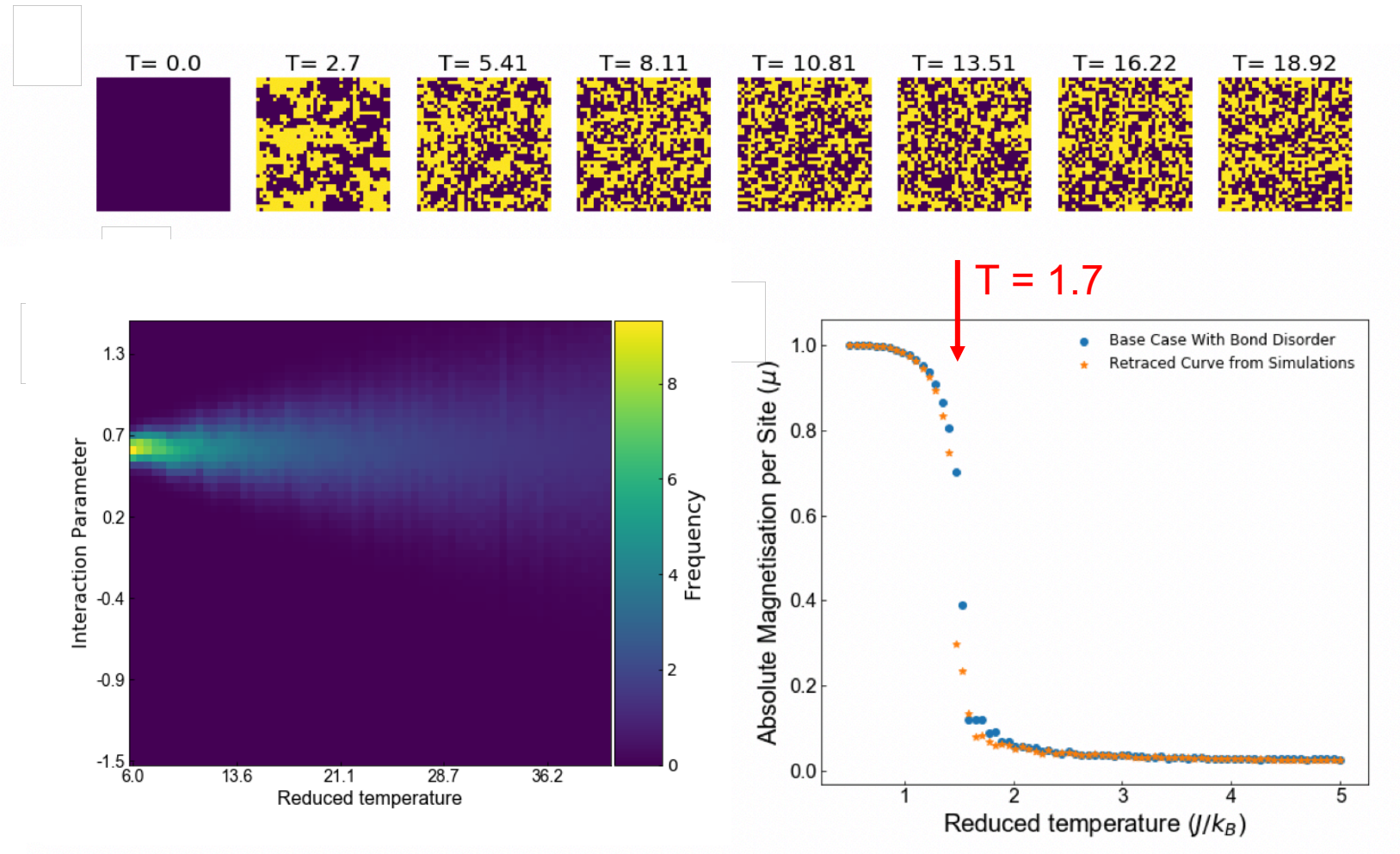
Minimize statistical distance between histograms

$$s = \arccos \left( \sum_{i=1}^k \sqrt{p_i} \sqrt{q_i} \right)$$



Vlcek et al., ACS Nano (2017)

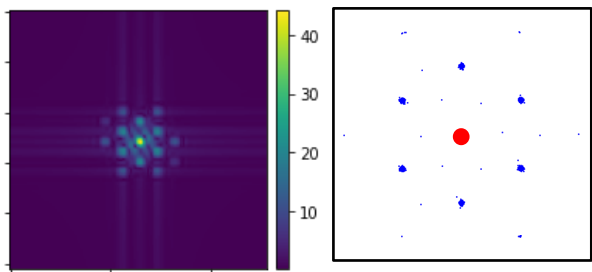
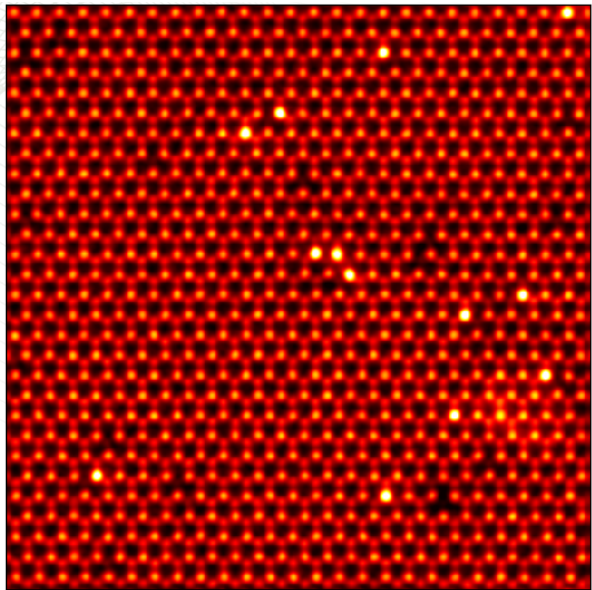
- Utilize all available statistical information in the image
- Using generative model, infer parameters from the experiment.



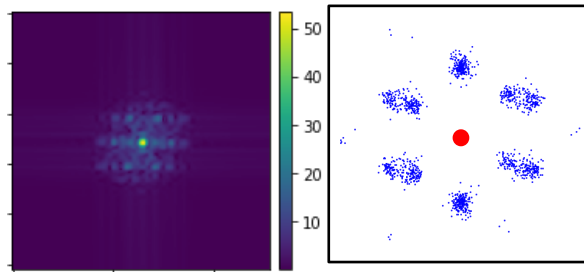
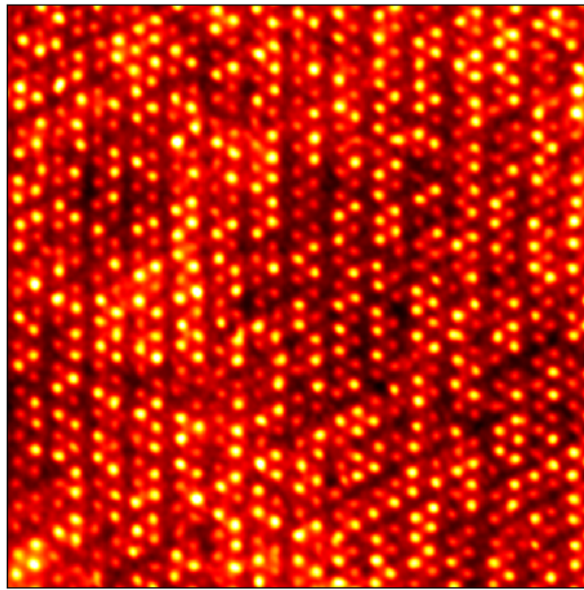


# MoS<sub>2</sub> – ReS<sub>2</sub> Solid solutions by STEM

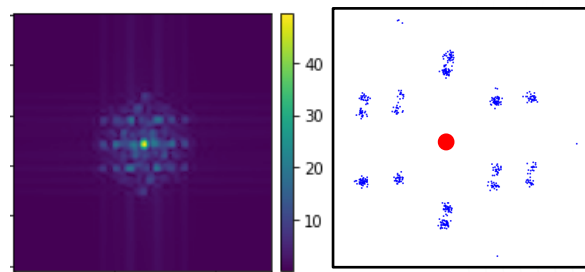
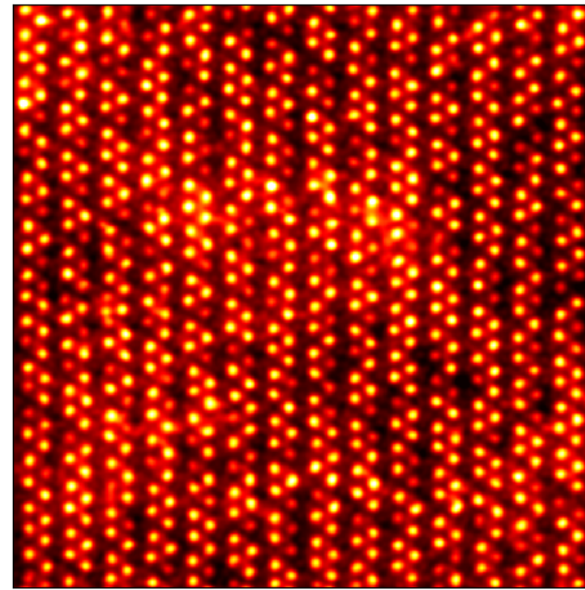
5% ReS<sub>2</sub>



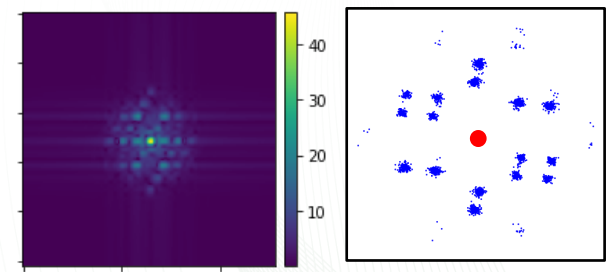
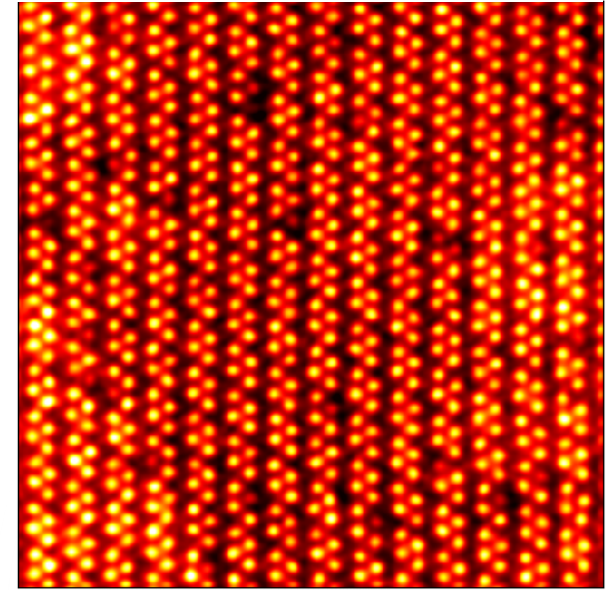
55% ReS<sub>2</sub>



78% ReS<sub>2</sub>

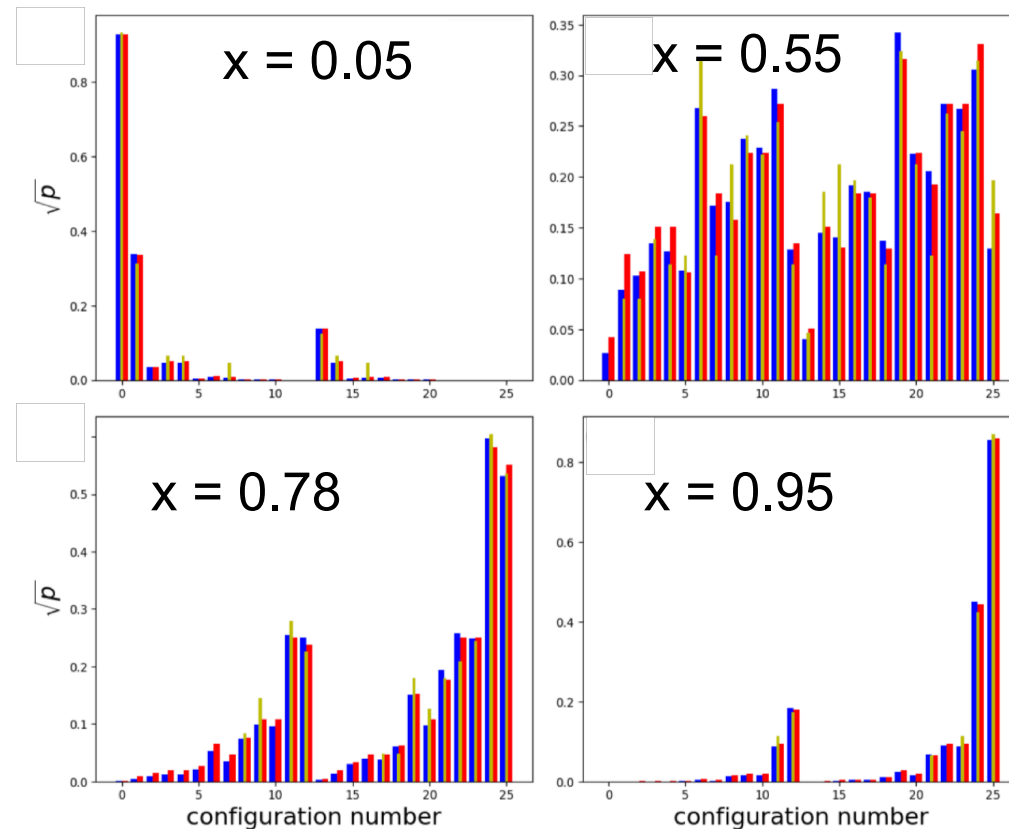


95% ReS<sub>2</sub>



Data by Shize Yang and Matt Chisholm

# Thermodynamics of solid solution



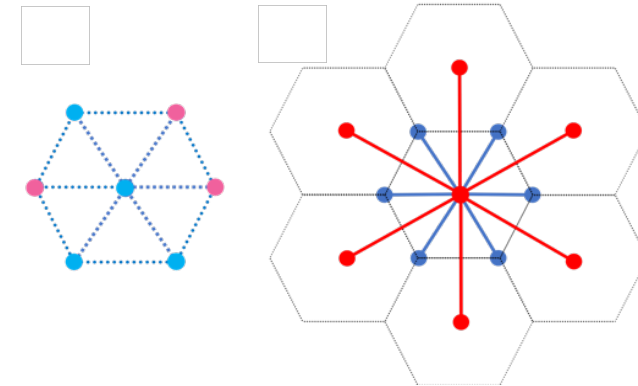
## 1. Pair-additive Model

$$u_i = w_1 \sum_{\{NN\}} \delta_{MoRe} + w_2 \sum_{\{NNN\}} \delta_{MoRe}$$

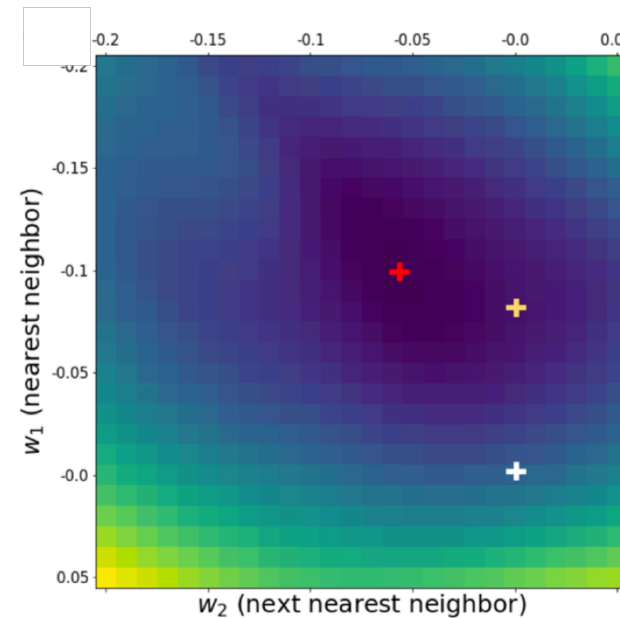
## 2. Many-body Model

$$u_i = w_1 \sum_{\{S\}} \delta_{MoMoRe} + w_2 \sum_{\{S\}} \delta_{MoReRe}$$

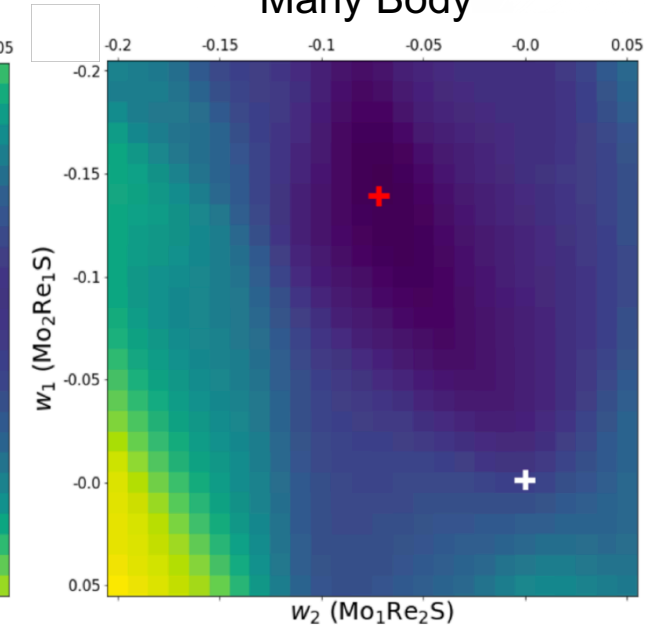
## Nearest and next-nearest interactions



## Pair Additive

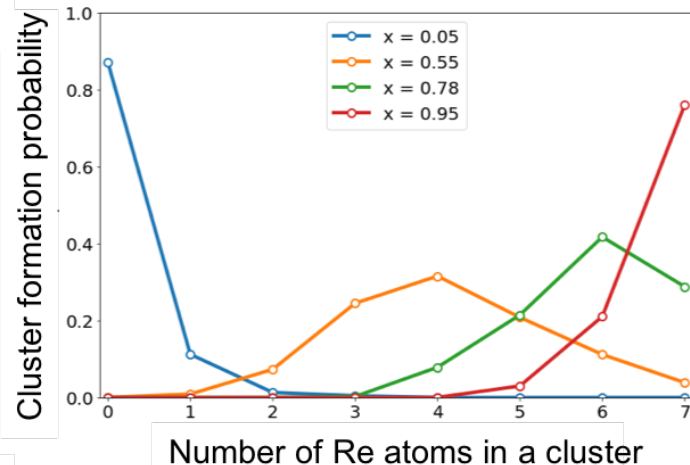
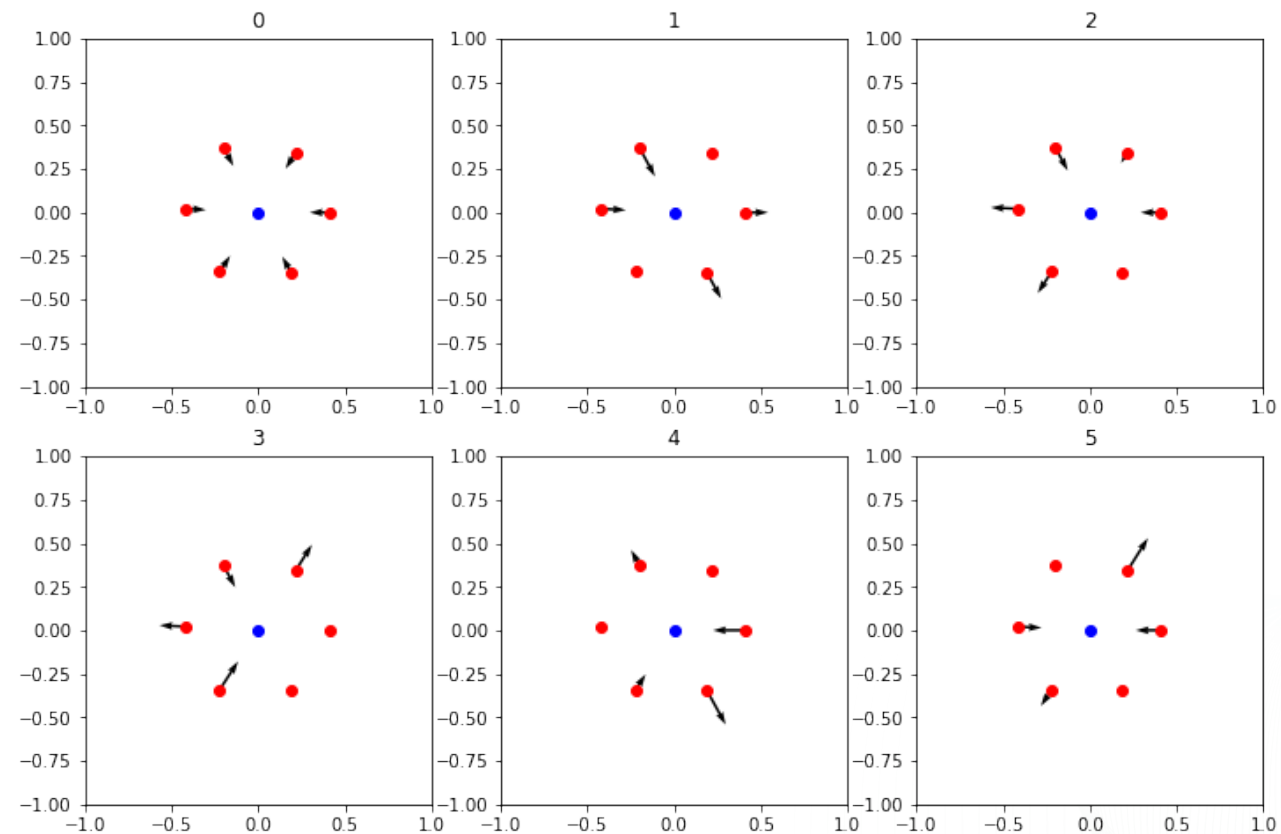
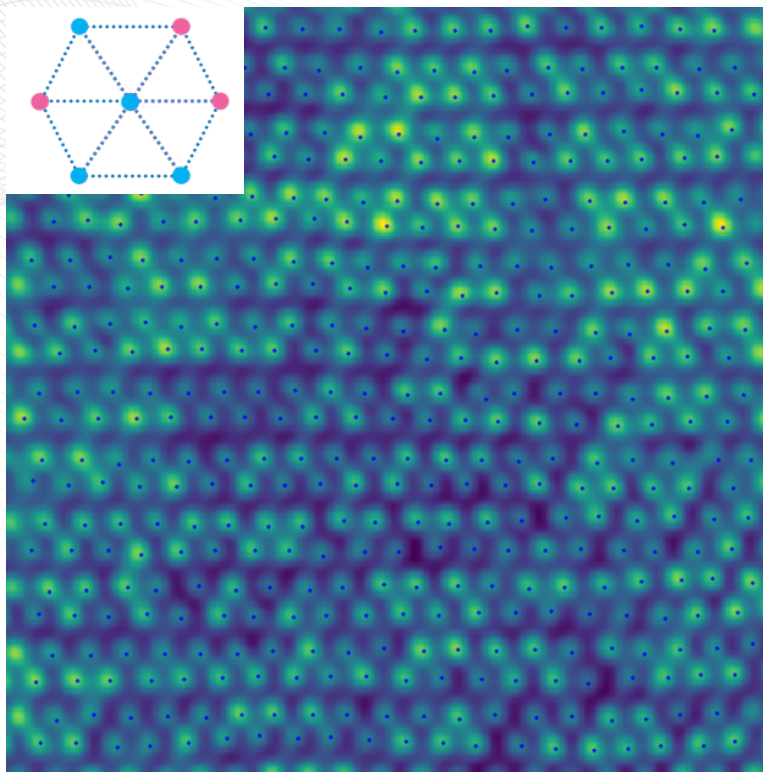


## Many Body



L. Vlcek, S. Yang, Y. Gong, P. Ajayan, W. Zhou, M. F. Chisholm, M. Ziatdinov, R. K. Vasudevan, S. V. Kalinin, [submitted], 2019.

# Exploring physics: statistical normal modes

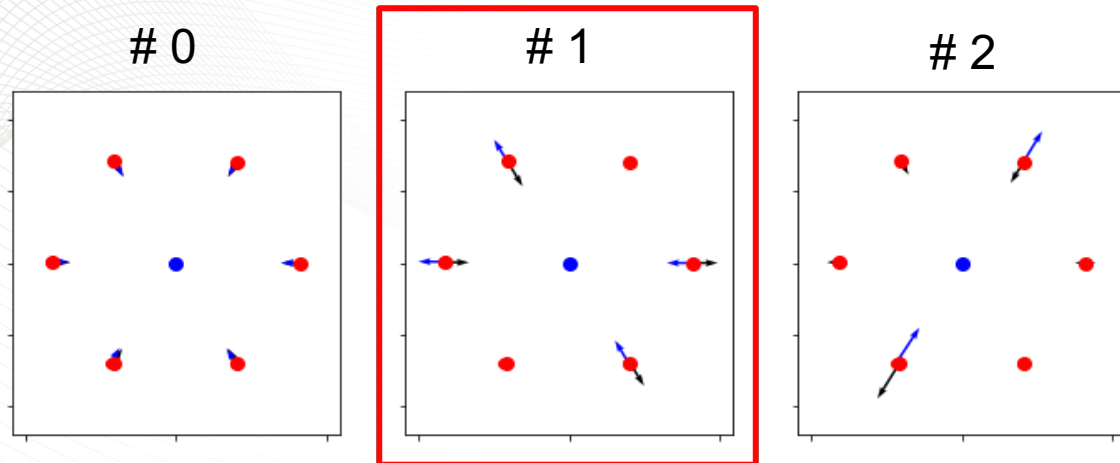


- Traditionally, the order parameter is defined based on symmetry and atomistic representation is established in the *ad hoc* manner
- But what if we define order parameter from the bottom up – based on the statistics of atomic distortions?
- And further correlate it to local chemical composition?

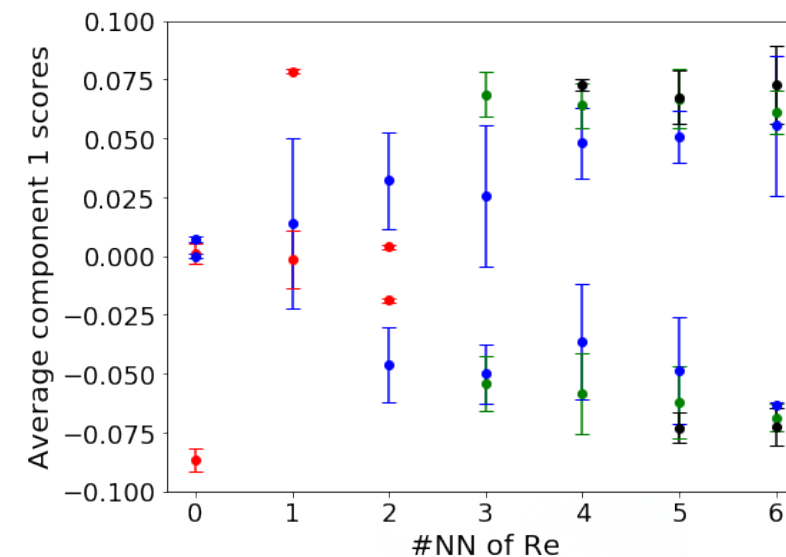


# Phase transition via statistical normal modes

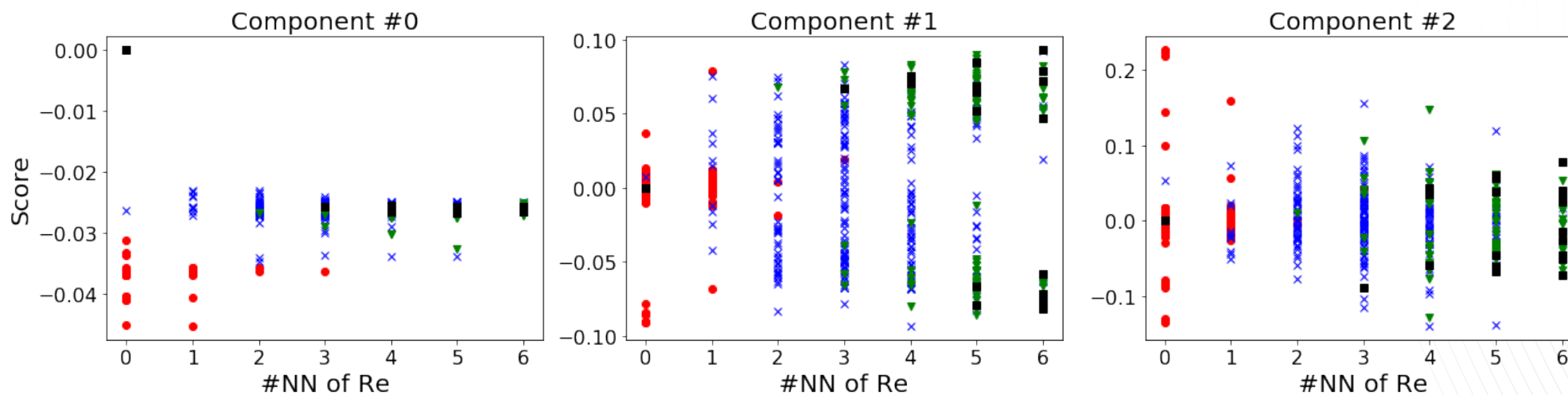
Three dominant distortion modes



Local symmetry breaking!



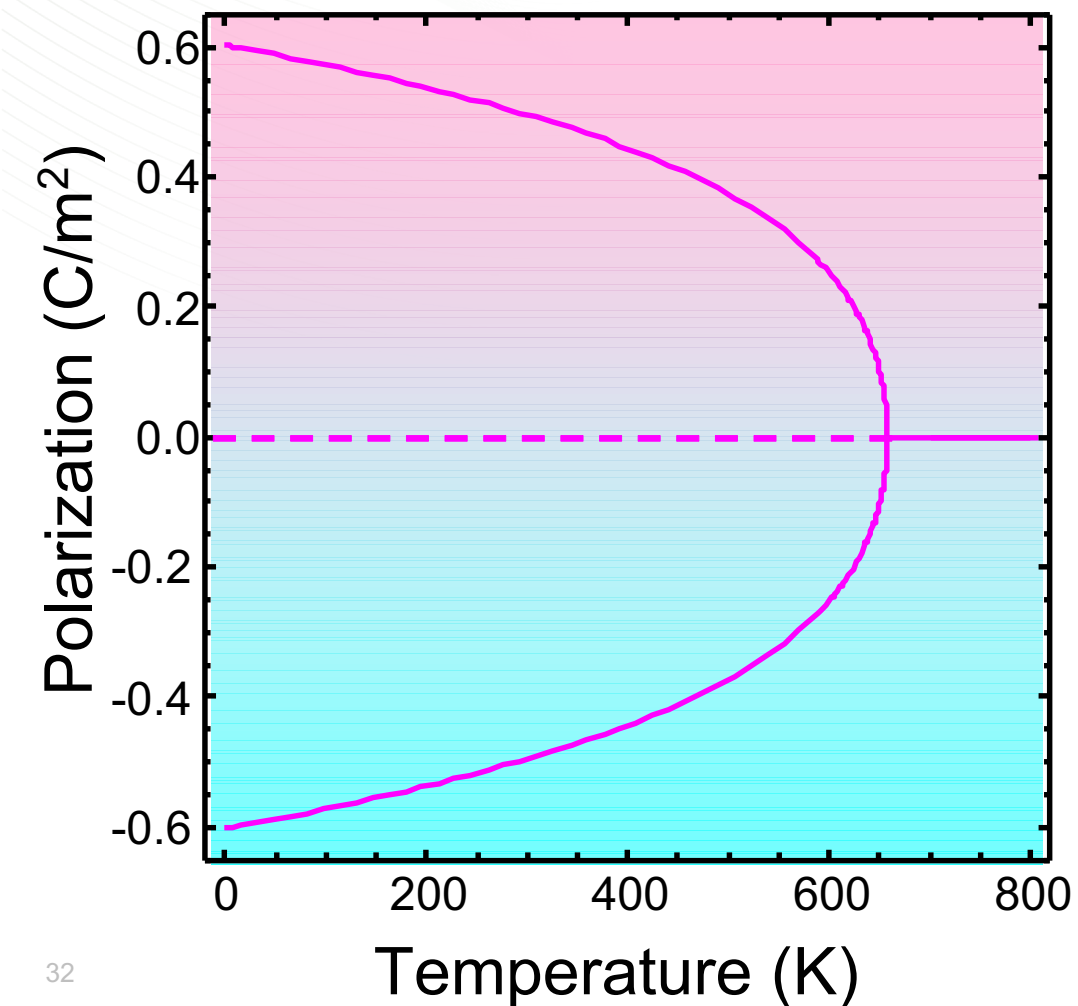
Mode distributions vs. global and local composition



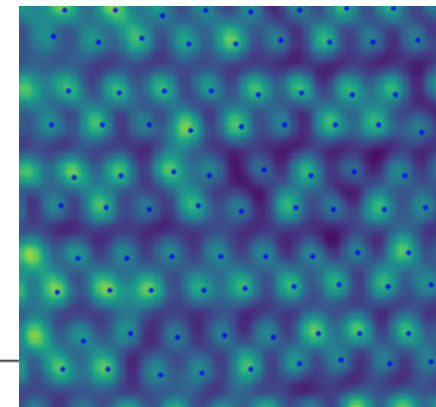
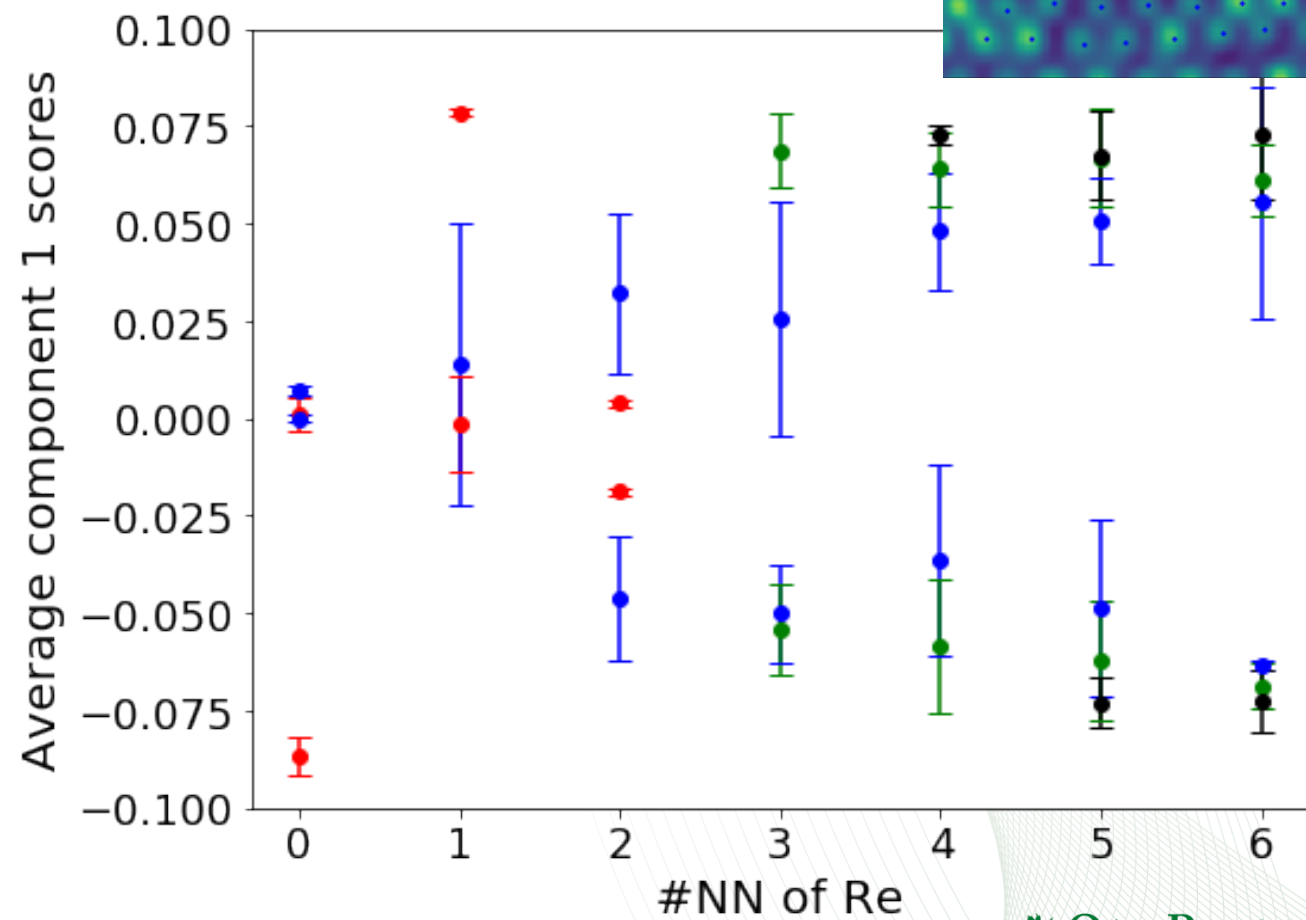
5% ReS<sub>2</sub>  
55% ReS<sub>2</sub>  
78% ReS<sub>2</sub>  
95% ReS<sub>2</sub>

# Phase transition from the bottom up

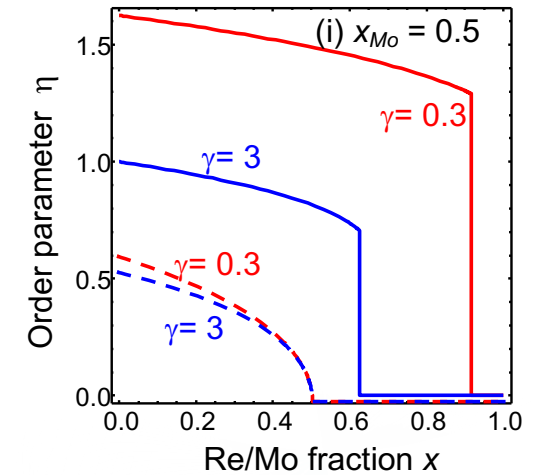
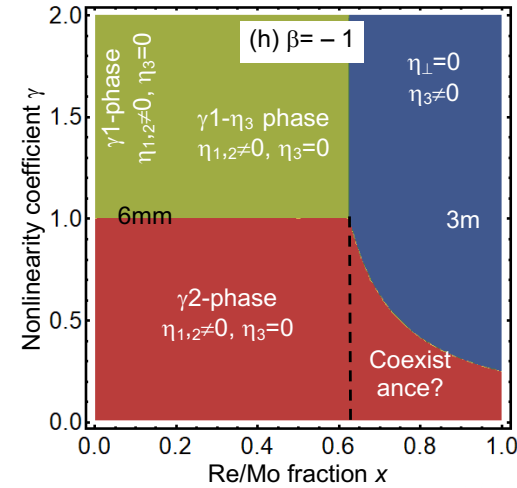
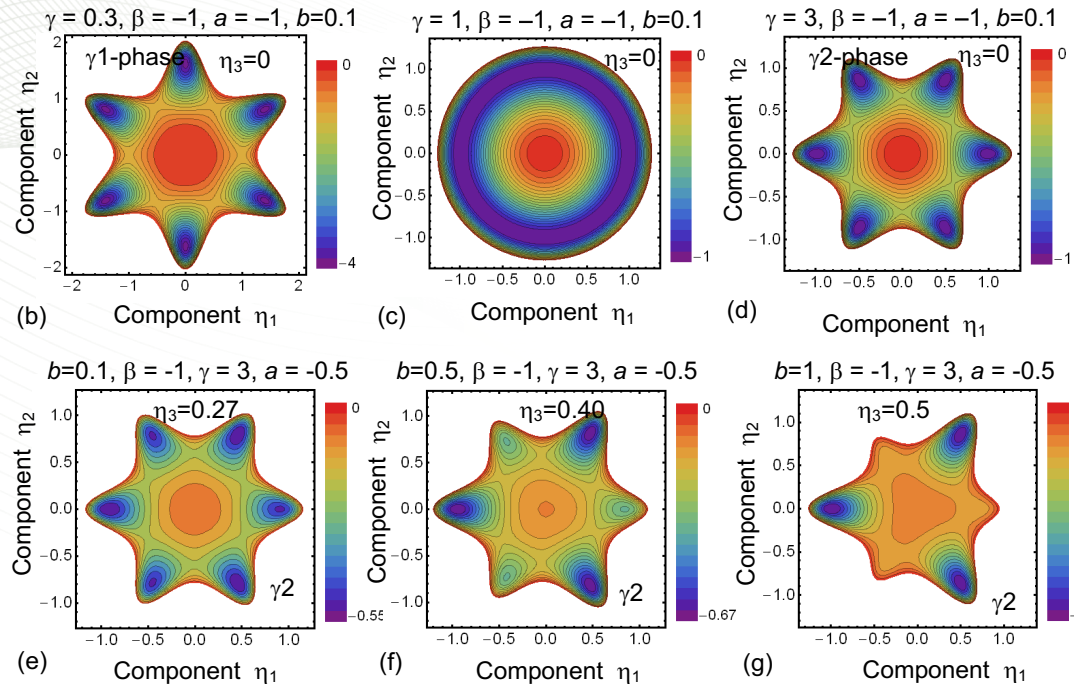
Global symmetry breaking



Local symmetry breaking!



# Transition to mesoscopic free energy



## Free energy functional for 2D materials

$$F[\boldsymbol{\eta}] = \int_S d^2r \left( f_L[\boldsymbol{\eta}(\vec{r})] + f_{EL}[\boldsymbol{\eta}(\vec{r})] + \frac{g_{ijkl}}{2} \frac{\partial \eta_i}{\partial x_j} \frac{\partial \eta_k}{\partial x_l} + \frac{f_{ijkl}}{2} \left( \frac{\partial \eta_i}{\partial x_j} u_{kl} - \eta_i \frac{\partial u_{kl}}{\partial x_j} \right) \right)$$

## Local free energy term:

$$f_L[\boldsymbol{\eta}(\vec{r})] = \alpha_{11}\eta_{\perp}^2 + \alpha_{33}\eta_3^2 + \beta_{11}\eta_{\perp}^4 + \beta_{13}\eta_{\perp}^2\eta_3^2 + \beta_{33}\eta_3^4 + \beta_{15}\eta_1\eta_3(\eta_1^2 - 3\eta_2^2) \\ + \gamma_{111}\eta_1^2(\eta_1^2 - 3\eta_2^2)^2 + \gamma_{222}\eta_2^2(3\eta_1^2 - \eta_2^2)^2 + \gamma_{113}\eta_{\perp}^4\eta_3^2 + \gamma_{133}\eta_{\perp}^2\eta_3^4 + \gamma_{333}\eta_3^6$$

## Opportunity:

- Phase diagrams
- Electronic phenomena induced by curvature via flexoelectric coupling
- Field- and doping induced transitions



# A Bit of (Ancient and Modern) Wisdom

*...there are known knowns; there are things we know we know. We also know there are known unknowns; that is to say we know there are some things we do not know. But there are also unknown unknowns—the ones we don't know we don't know.*

D. Rumsfeld

*Har kas ke bedanad va bedanad ke bedanad  
Asb-e kherad az gombad-e gardun bejahanad  
Har kas ke nadanad va bedanad ke nadanad  
Langan kharak-e khish be manzel beresanad  
Har kas ke nadanad va nadanad ke nadanad  
Dar jahl-e morakkab'abad od-dar bemanad*

Anyone who knows, and knows that he knows  
Makes the steed of intelligence leap over the vault of heaven  
Anyone who does not know, but knows that he does not know  
Can bring his lame little donkey to the destination nonetheless  
Anyone who does not know, and does not know that he does not know  
Is stuck forever in the double ignorance

Naser od-Din Tusi (1201-1274)



# The World is Bayesian: Physics from Observations

*...there are known knowns; there are things we know we know. We also know there are known unknowns; that is to say we know there are some things we do not know. But there are also unknown unknowns—the ones we don't know we don't know.*

D. Rumsfeld

**Hypothesis driven science:**  
What we want to learn

**Forward model:**  
Theory

**Domain expertise:**

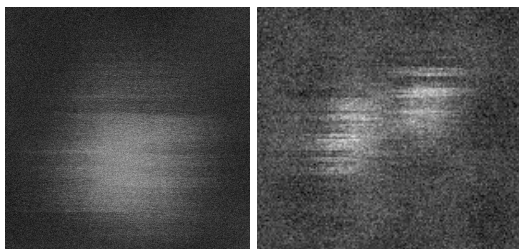
$$P(\text{Theory}|\text{Data}) = \frac{P(\text{Data}|\text{Theory})P(\text{Theory})}{P(\text{Data})}$$

- Experimentalists know the priors. Albeit they do not know that they know it, or how to convert them to algorithmic form
- **How can we add Bayesian priors to:**
  - Reinforcement learning (functionality optimization in experiment)
  - GANs (inverse problems, image reconstruction)
  - VAEs (physical constraints on latent variables)

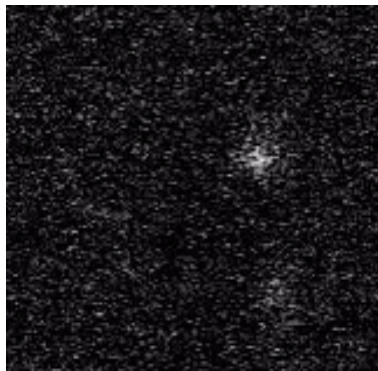
**High Performance  
Computing**

# The chemistry challenge

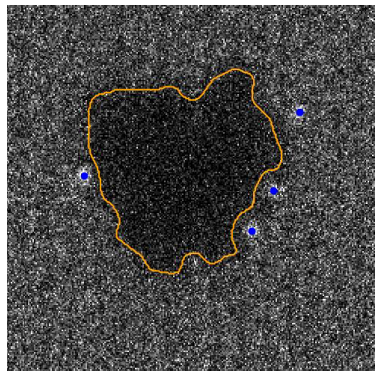
R. Ishikawa



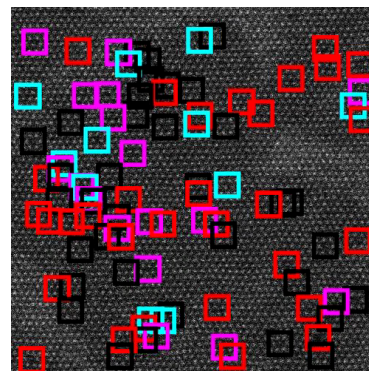
O. Dyck



O. Dyck

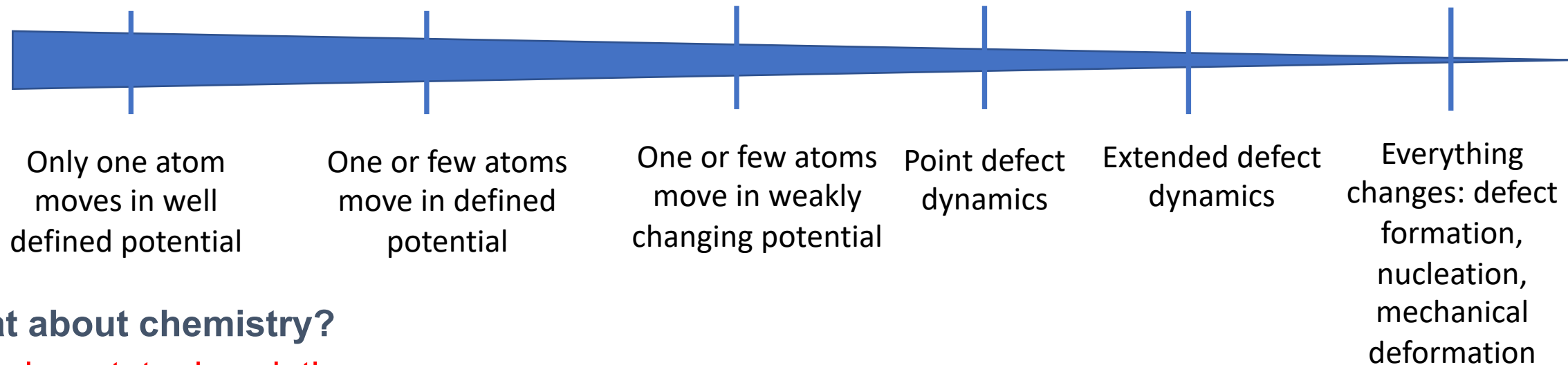


O. Dyck



A. Lupini

M. Chi



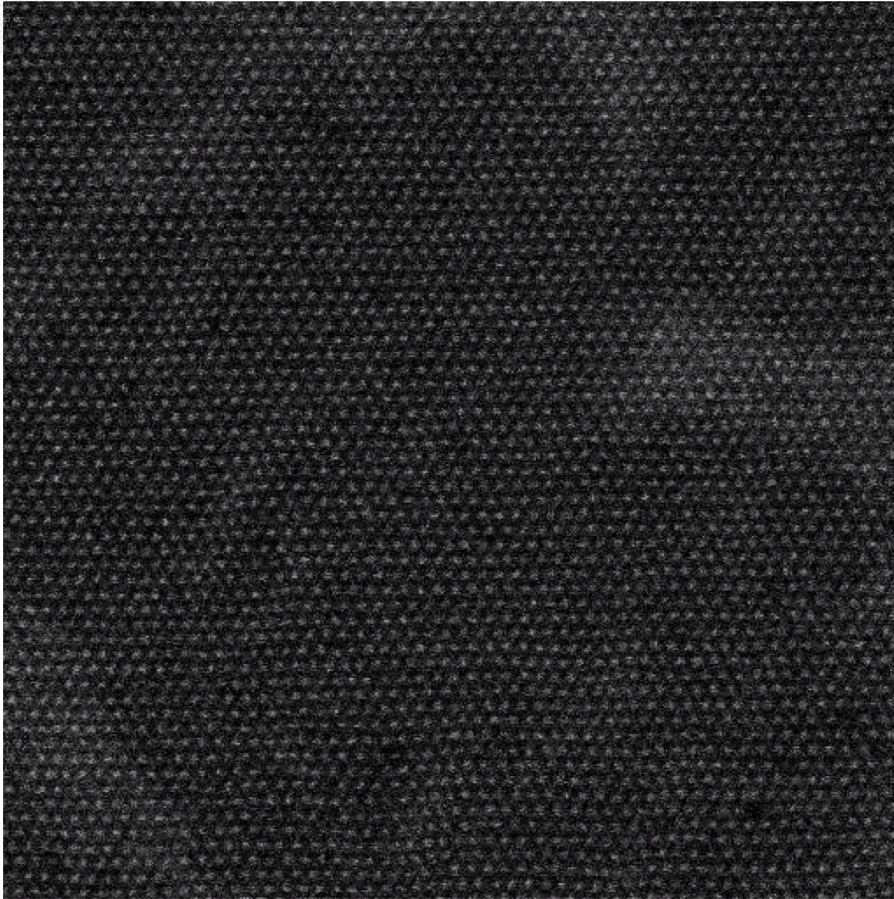
## What about chemistry?

- Markov state descriptions
- Potential energy landscape reconstructions
- Force fields (and excitation) from observed dynamics



# Learning the defect evolution

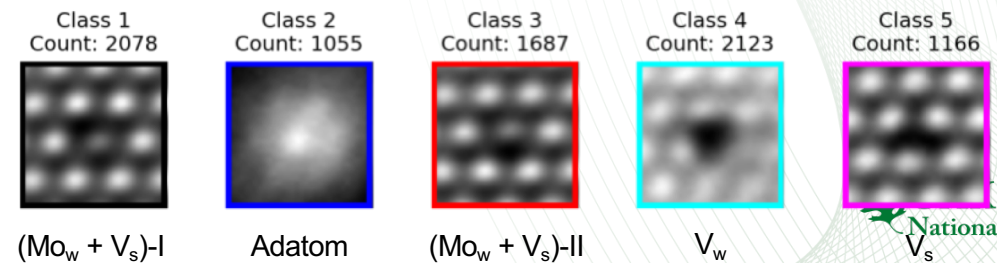
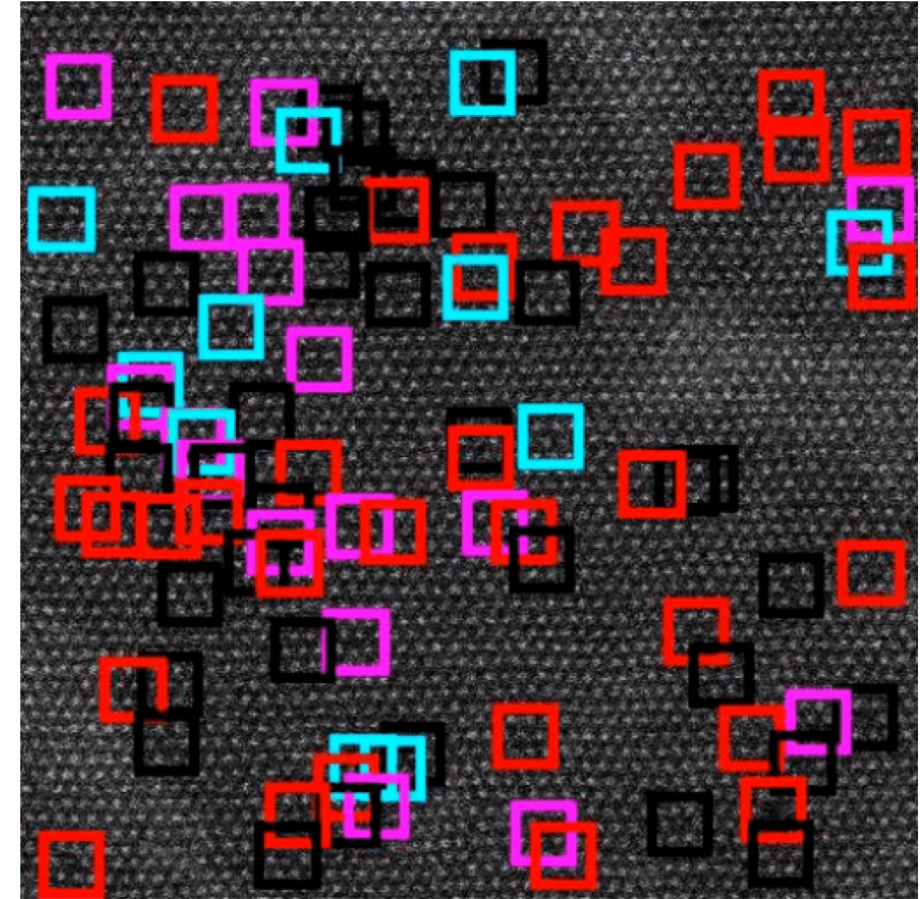
Experimental



Sample: WS2  
E-beam energy: 60 kV

Data collected by Ondrej Dyck (CNMS/ORNL)

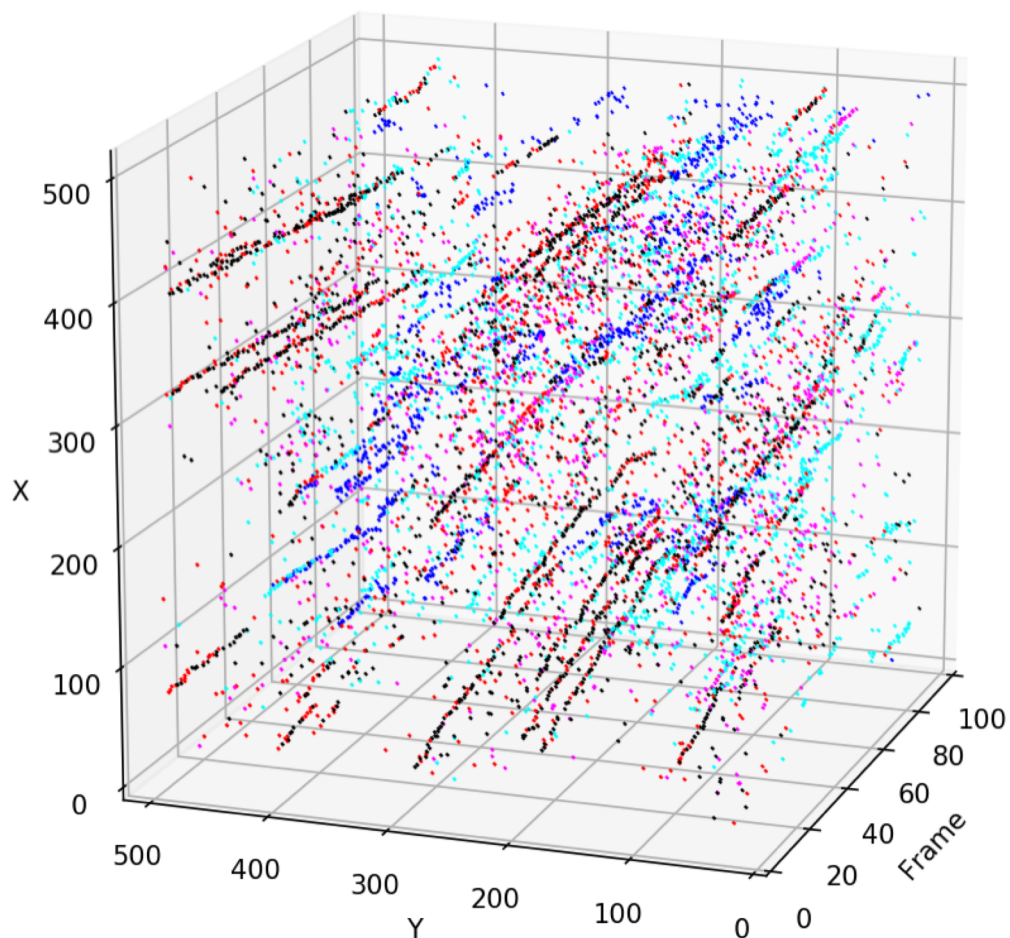
Decoded





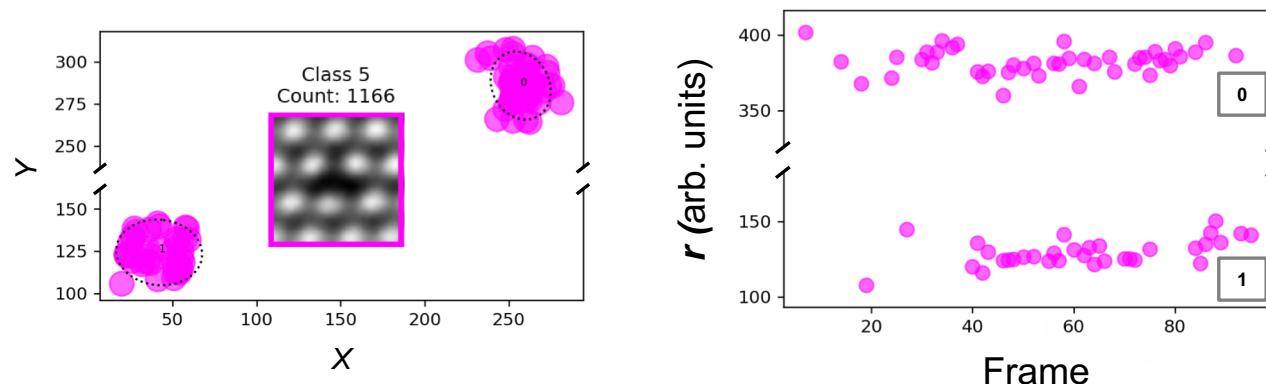
# Physics extraction

## Spatio-temporal trajectories



Maksov *et al.*, npj Computational Materials 5, 12 (2019)

## Diffusion parameters for selected defect types

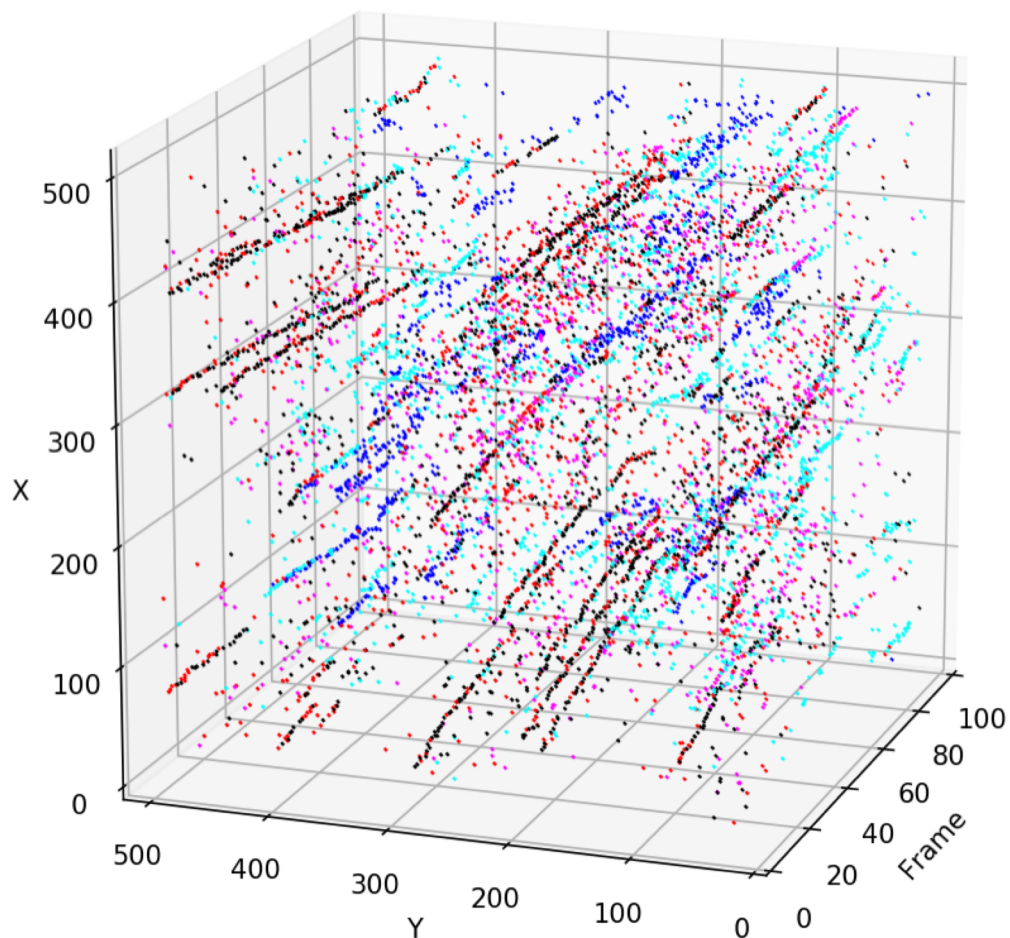


Diffusion coefficient:  $3 \times 10^{-4} \text{ nm}^2/\text{s}$  -  $6 \times 10^{-4} \text{ nm}^2/\text{s}$   
(within 2D random walk approximation)

- Identification of dominant point defects and their characteristic statistical behaviors
- Analysis of diffusion parameters for the selected defect species
- Study of transformation pathways and transition probabilities for composite defects

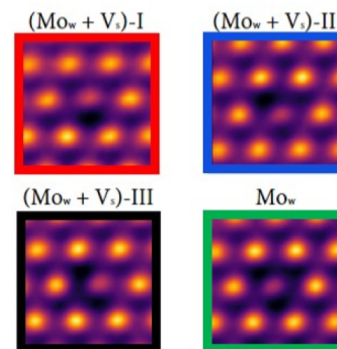
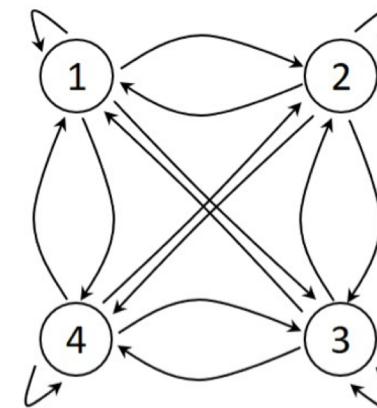
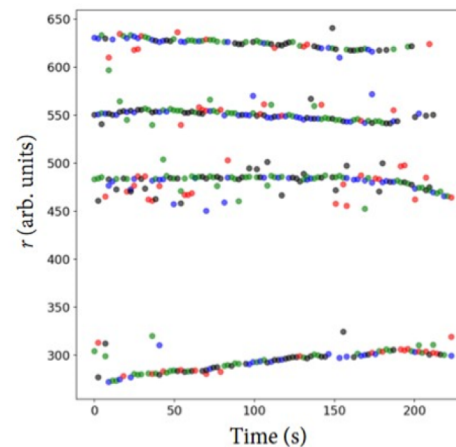
# Physics extraction

## Spatio-temporal trajectories



Maksov *et al.*, npj Computational Materials 5, 12 (2019)

## Evolution of defects as Markov process



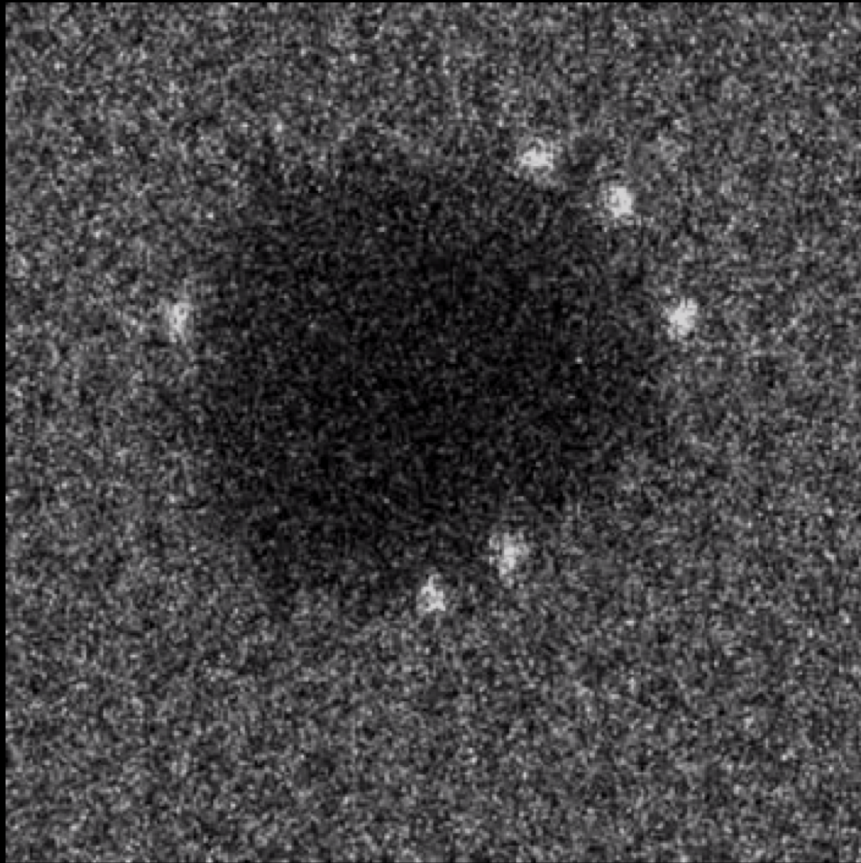
Starting class	$(Mo_w + V_s)$ -I	$(Mo_w + V_s)$ -II	$(Mo_w + V_s)$ -III	$Mo_w$
$(Mo_w + V_s)$ -I	0.11	0.32	0.27	0.31
$(Mo_w + V_s)$ -II	0.12	0.18	0.36	0.34
$(Mo_w + V_s)$ -III	0.18	0.23	0.27	0.32
$Mo_w$	0.17	0.20	0.28	0.35
Transition class	$(Mo_w + V_s)$ -I	$(Mo_w + V_s)$ -II	$(Mo_w + V_s)$ -III	$Mo_w$



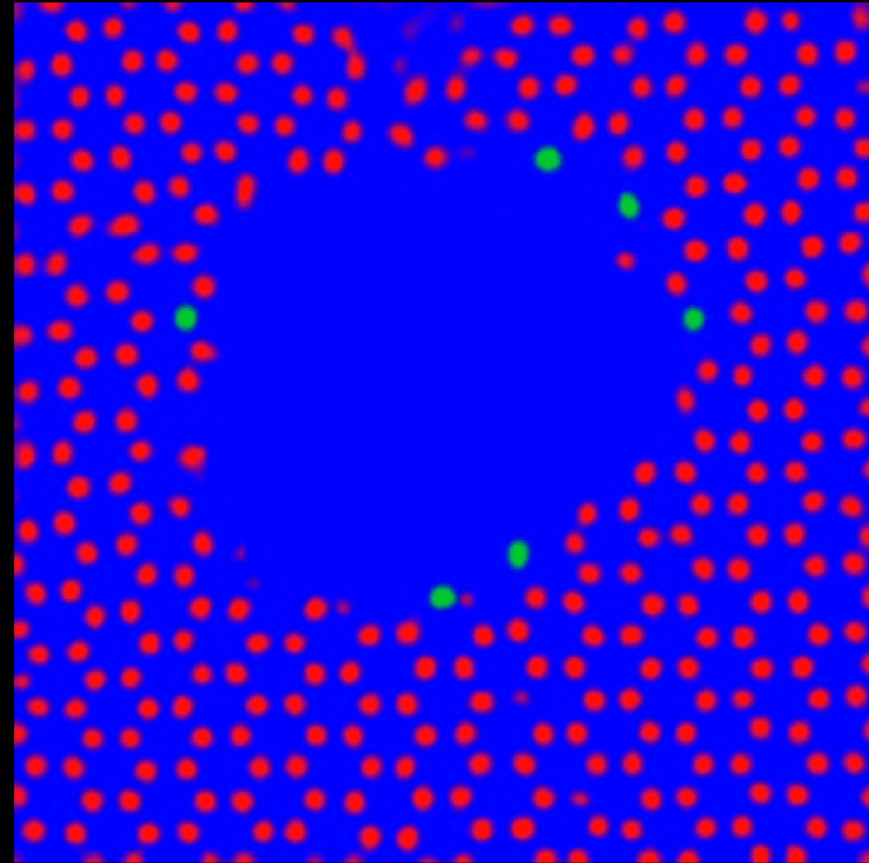
# Beam induced reactions of Si atoms on the edge of graphene

Reconstructing Si impurity configurations at graphene edge

Experimental data



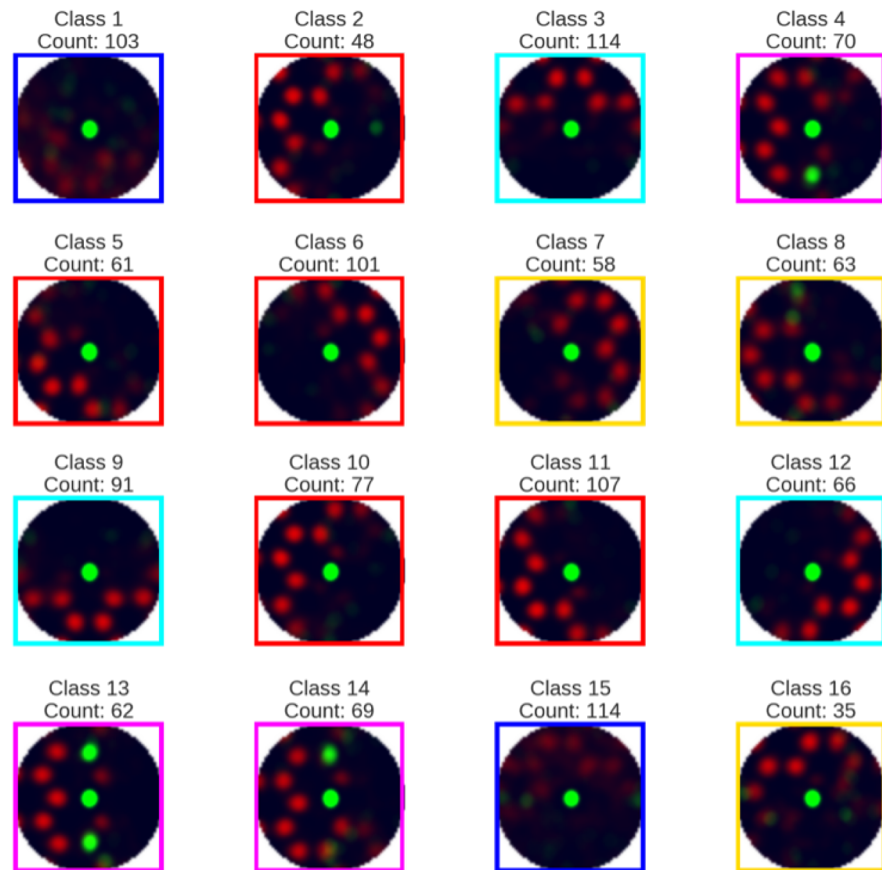
Network's output



Ziatdinov *et al.*, ArXiv:1901.09322 (2019)

# Chemical transformations on the edge

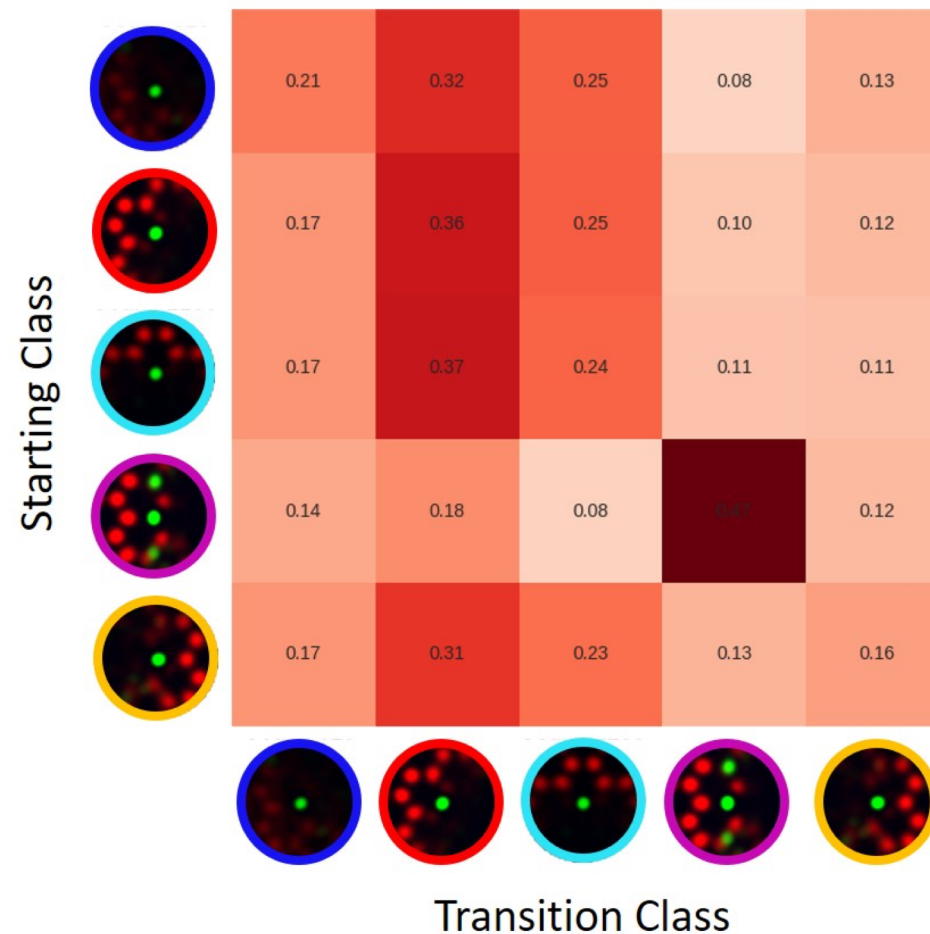
Derived classes of Si-C edge configurations



- Gaussian mixture model

- Discrete rotation symmetry +  
structural similarity algorithm

Transition probabilities matrix



- Markov state analysis

“What I cannot create, I do  
not understand.”  
— *Richard Feynman*

Molecular machines  
2016 Nobel Prize



“What I cannot create, I do  
not understand.”

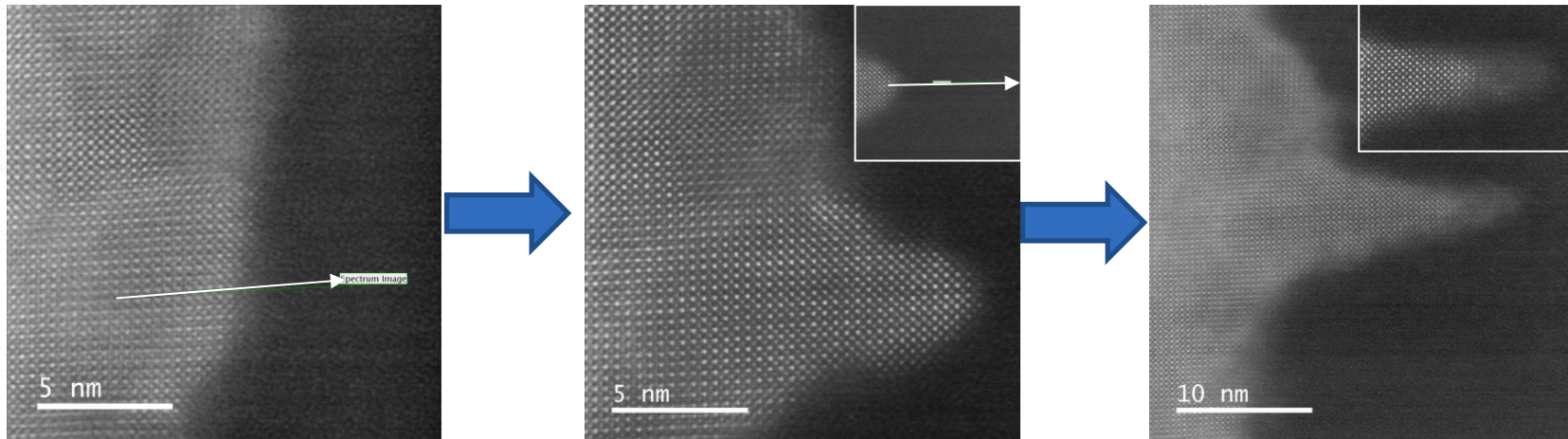
J. P. Sauvage, F. Stoddard, and B. Feringa

— *Richard Feynman*

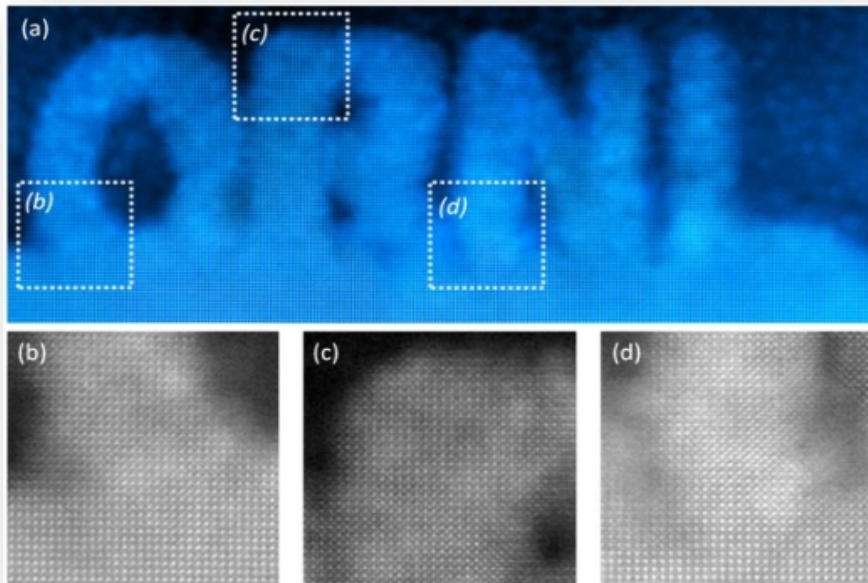
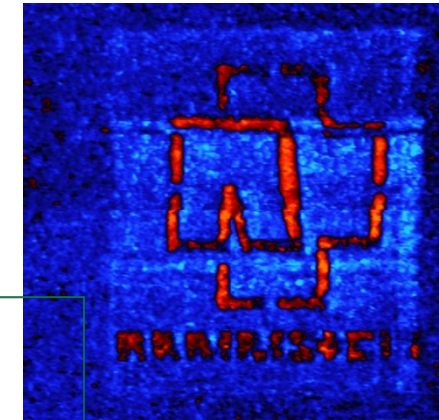
Is There a Third Way?



# Synergy of STEM and SPM: Sculpting

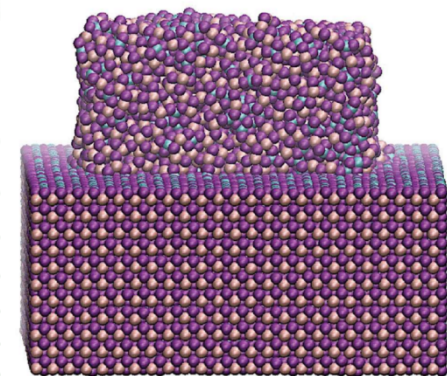


We induce local amorphous/crystalline transition in a defined area

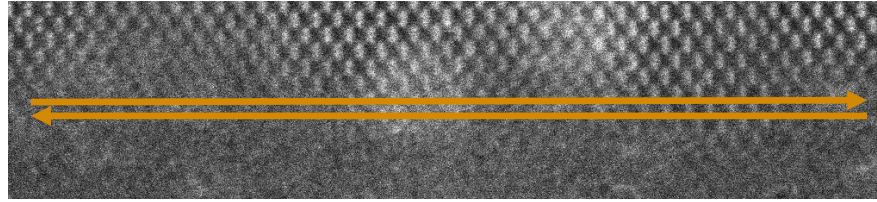


- No crystallization observed away from interface
- Electron-beam induced solid phase epitaxy
- Control of beam position and speed
- We chose to explore, exploit, and understand this behavior

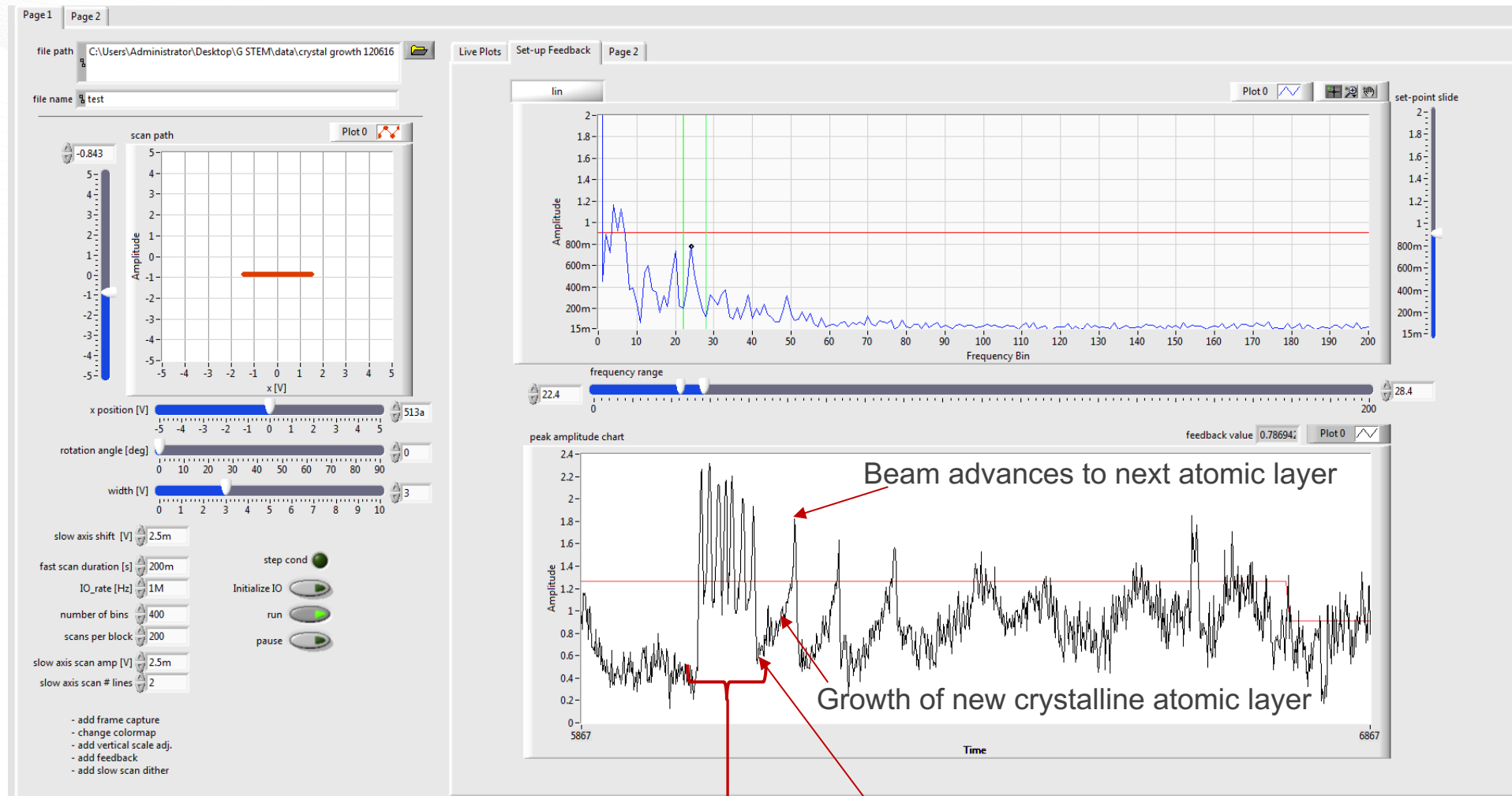
- Formation of 3D structures inside solid
- Epitaxial registration
- MD simulations have confirmed experimental observations
- Foundation for making atomically precise, multi-component, multi-layer systems



# Beam Induced Transformations in Solids: *Feedback*



Stephen Jesse



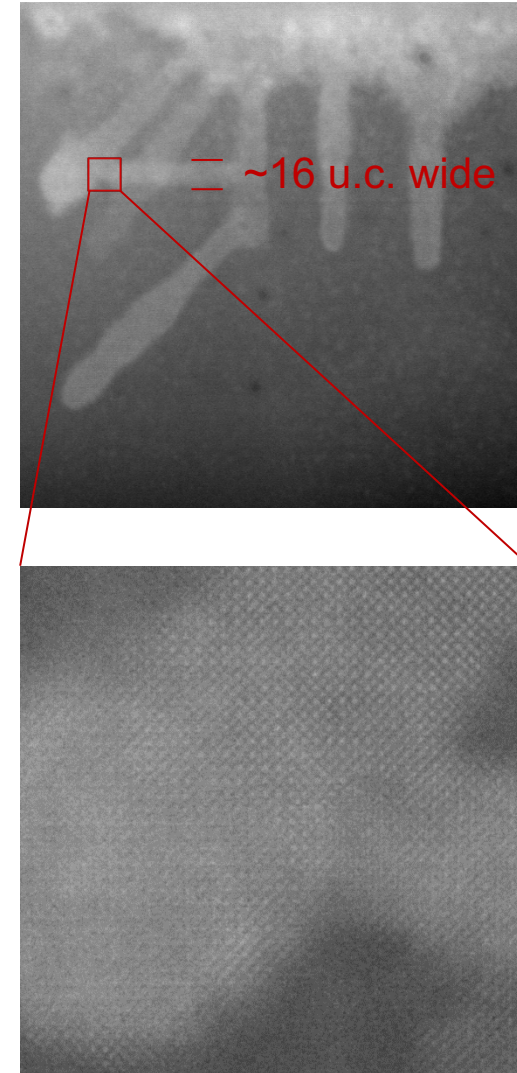
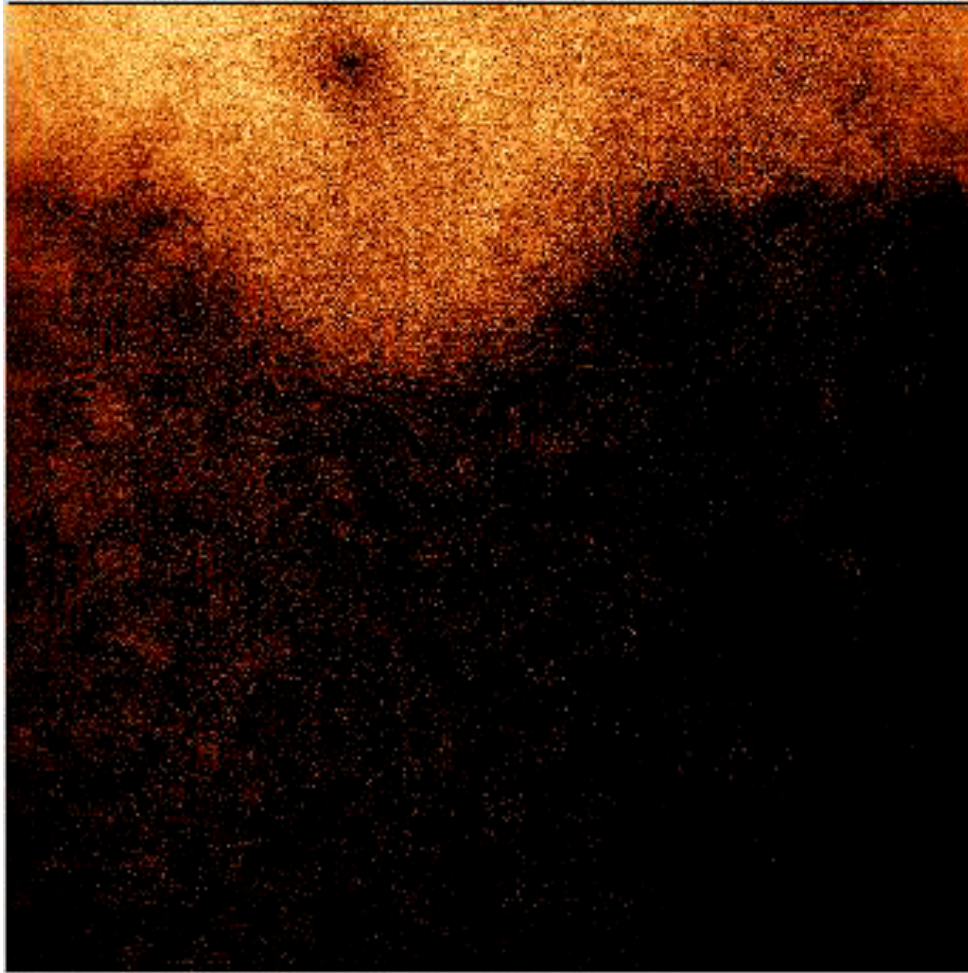
Start inside crystal,  
Fast advance

Amorphous/x-line  
interface



# Manufacturing at the Atomic Scale with Beams

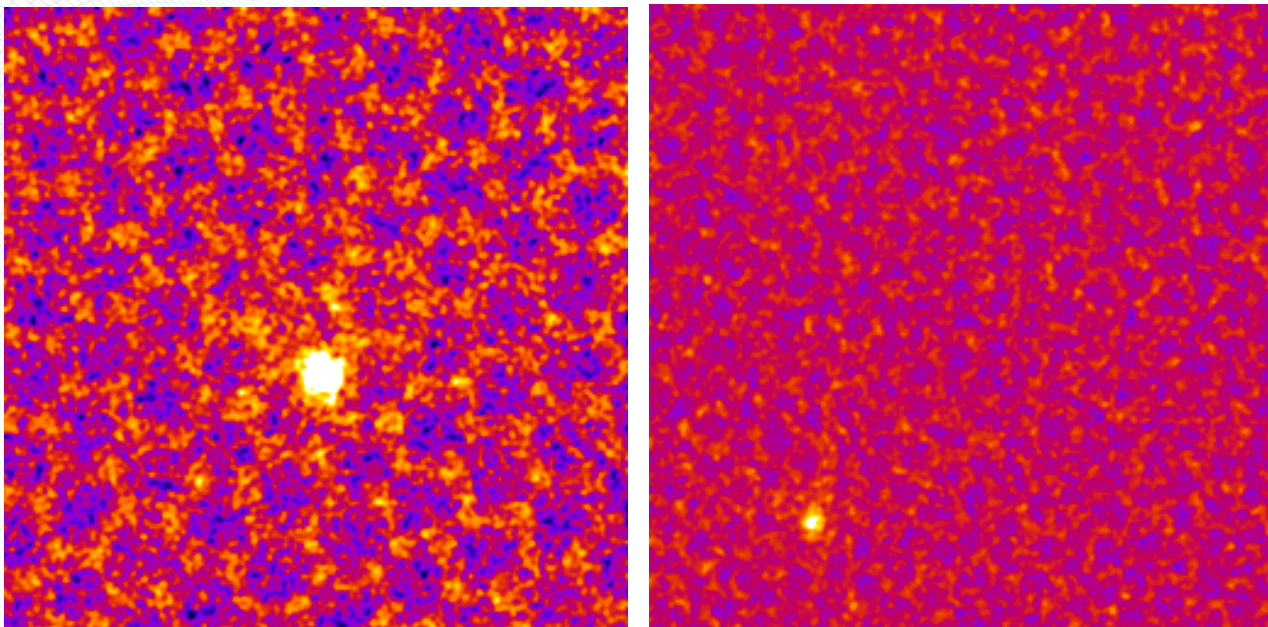
Directed Crystal Growth in STO



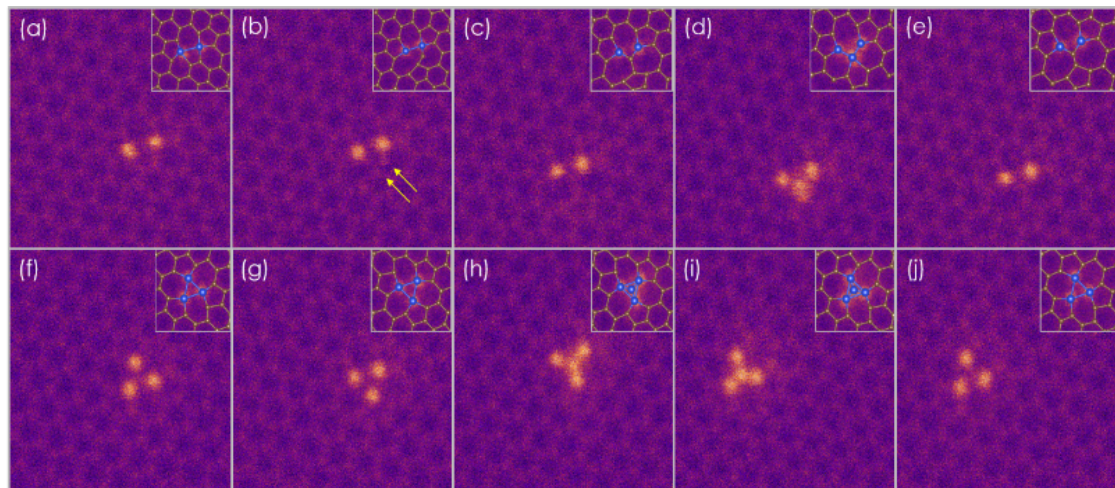


# Atomic manipulations

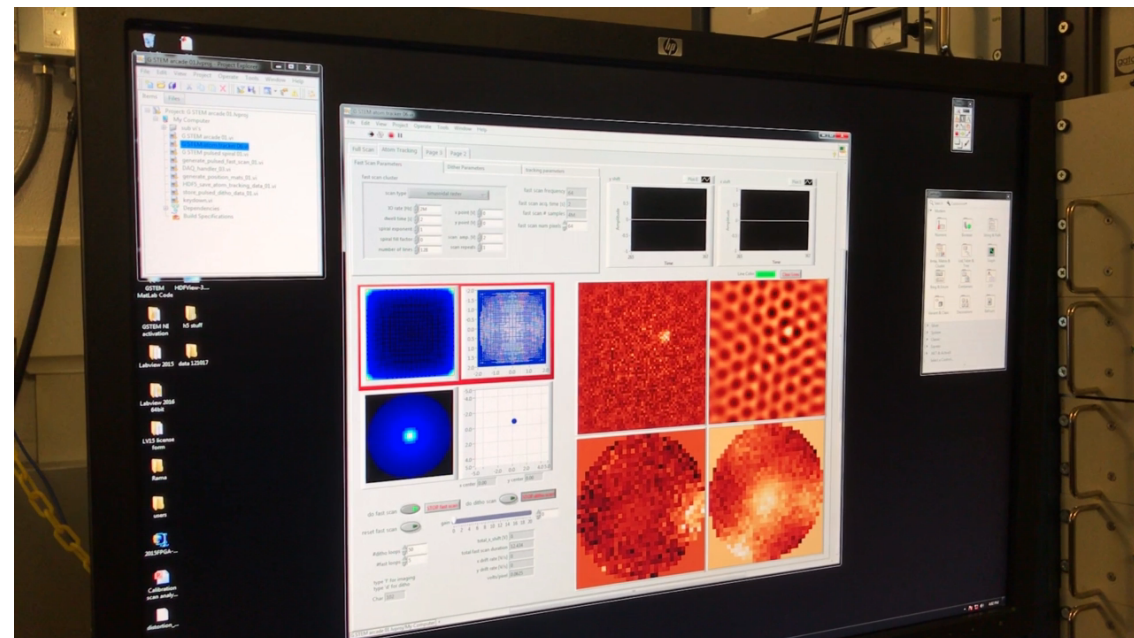
## Moving Si dopants



## Assembling primitive structures



Custom platform for dopant atom movement with real-time image updates during manipulation



Ondrej  
Dyck



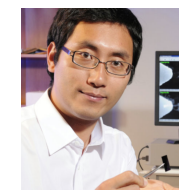
Stephen  
Jesse



Elisa Jimenez-  
Izal



Anastasia N.  
Alexandrova



Songkil  
Kim

Dyck, O., Kim, S., Jimenez-Izal, E., Alexandrova, A. N., Kalinin, S. V. & Jesse, S. Building Structures Atom by Atom via Electron Beam Manipulation. *Small* **14**, 1801771, doi:doi:10.1002/smll.201801771 (2018).



# STEM: The Lab on the Beam

*What I cannot create, I do not understand*  
R. Feynman

**X-ray and neutron scattering:**  
where the atoms are on average

**Electron and probe microscopy:** where exactly are the atoms

**Dynamic microscopies:**  
what the atoms do

**Need to find out :**

- What are local atomic functionalities
- Why do atoms do it ?

- How we direct them to do what we want ?

- Nanotechnology
- Beyond Moore
- Molecular machines
- Materials design

Time

Present

# Concluding:

- Data matters (but only if physics and chemistry is analyzed!)
- Building physics and chemistry from atomic level up
- E-beam atomic fabrication



See [cnms.ornl.gov](http://cnms.ornl.gov) for more information.

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[sergei2@ornl.gov](mailto:sergei2@ornl.gov)



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