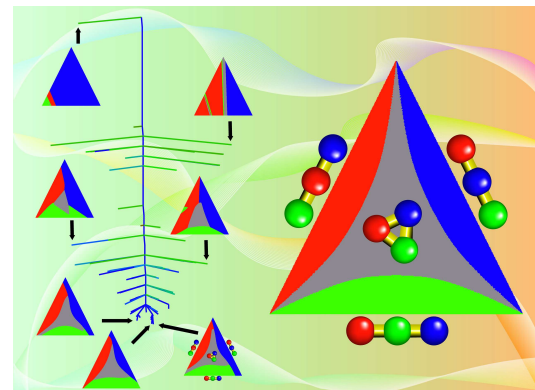
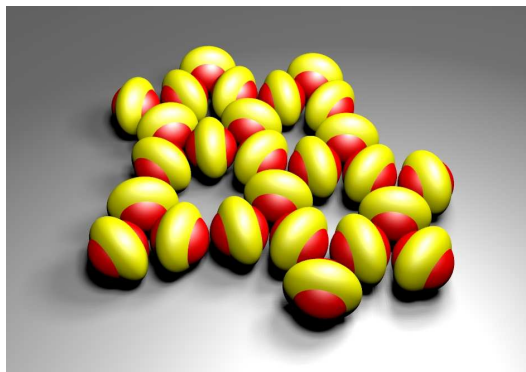
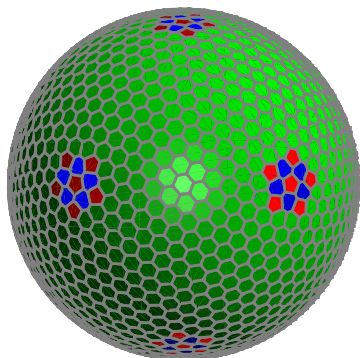
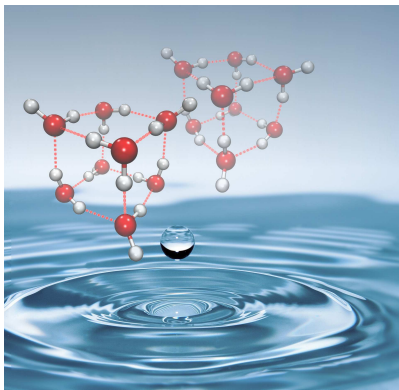


Energy Landscapes: Structure, Dynamics, and Thermodynamics

Recap: exploit **stationary points** (minima and transition states) as a **conceptual** and **computational** framework (*J. Phys. Chem. B*, **110**, 20765, 2006)

- **Basin-hopping** for global optimisation (*J. Phys. Chem. A*, **101**, 5111 1997)
- **Basin-sampling** for global thermodynamics (*J. Chem. Phys.*, **124**, 044102, 2006)
- **Superposition Enhanced Nested Sampling** (*Phys. Rev. X*, **4**, 031034, 2014)
- **Discrete path sampling** for global kinetics (*Mol. Phys.*, **100**, 3285, 2002)
- **Coarse-grained** models employ generalised local **rigid bodies** and **ellipsoidal** building blocks (*JCTC*, **9**, 4026, 2013; *Soft Matter*, **11**, 6663, 2015)



Symmetrised Basin-Hopping (*Phys. Chem. Chem. Phys.*, **15**, 3965, 2013)

Searches are **biased** towards geometries with a higher **approximate** symmetry measure, defined in terms of a **cluster core** and an **orbit** analysis.

- Justification from the *principle of maximum symmetry*.
- The bias is **not** specific to any particular **point group** or **morphology**.

Mean first encounter times for the global minima of LJ_N clusters averaged over **100** random starting points (Intel Xeon E5405 cpu running at 2.00 GHz):

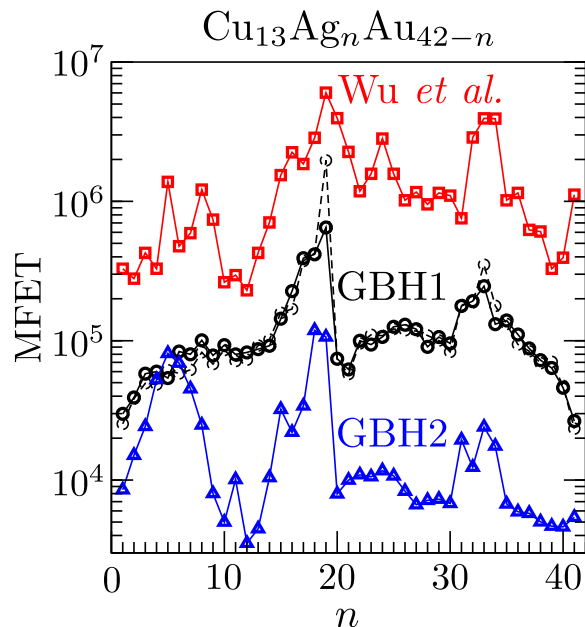
N	method	energy evaluations	minimisations	cpu time/s
38	none	185,493	1,271	4.4
38	core orbits symmetrisation	20,655	142	0.5
38	continuous symmetry measure	4,369	34	0.2
75	none	8,230,648	61,668	688.7
75	core orbits symmetrisation	50,229	338	4.2
98	none	7,017,387	48,301	1,314.3
98	core orbits symmetrisation	109,753	563	18.5

Generalised Basin-Hopping (*PRL*, **113**, 156102, 2014; *PCCP*, **17**, 902, 2016)

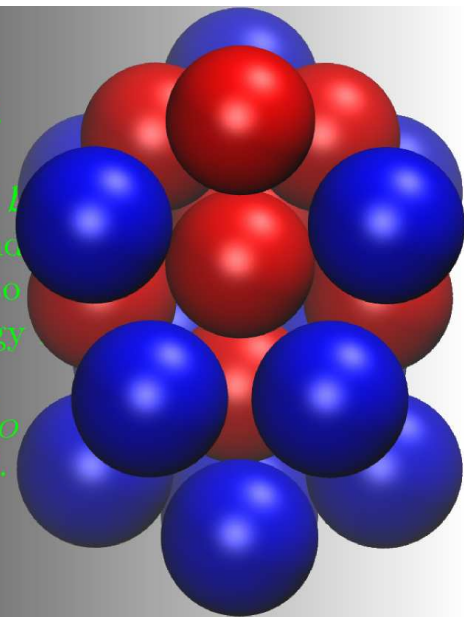
In **GBH** we focus on **biminima**, defined as structures where the energy cannot be lowered by **interchanging** inequivalent particles and **requenching**.

For **multicomponent** nanoalloy metal clusters fewer biminima exist for **segregated** systems with **lower** lattice mismatch.

In GBH1 the **biminimum** property is **verified** by scanning the local neighbourhood, while in **GBH2** a subset is considered. (*PCCP*, **17**, 28331, 2015)

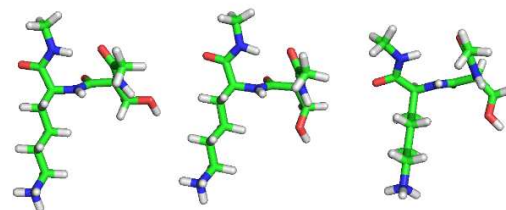
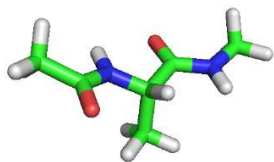
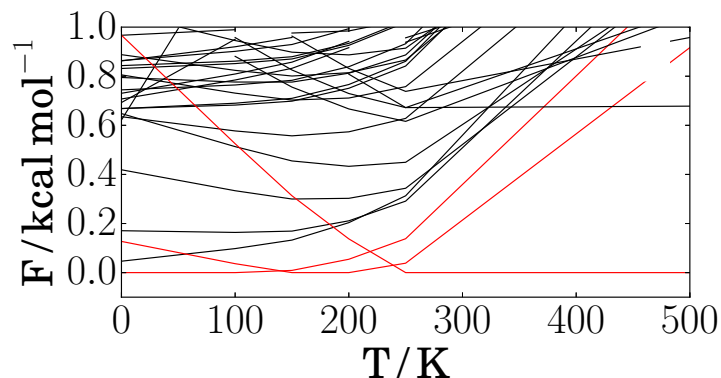
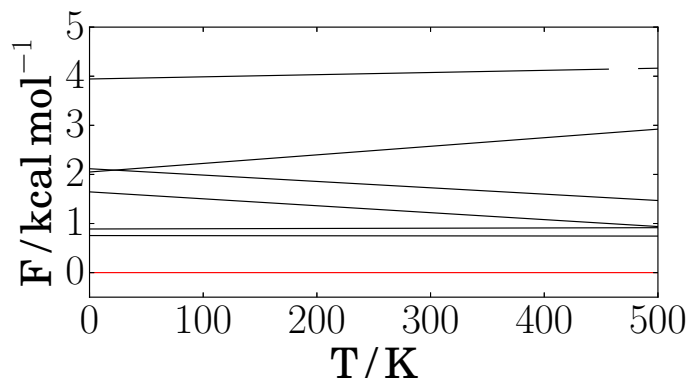


```
while C is FALSE:
  Update C to TRUE.
  Set  $\mathcal{A}' \leftarrow \mathcal{A}$  and  $\mathcal{B}' \leftarrow \mathcal{B}$ .
  do min( $N_A, N_B$ ) times:
    Pick  $a'_I \in \mathcal{A}'$  and  $b'_J \in \mathcal{B}'$ .
    Remove  $a'_I$  from  $\mathcal{A}'$  and
    Move  $a_I$  to  $\mathcal{B}$  and  $b_J$  to  $\mathcal{A}$ .
    Update the total energy
    if  $E < E_O$  then:
      Update  $E_O \leftarrow E$ ,  $\mathcal{A}_O$ 
      Reset C to FALSE.
    end if.
  end do loop.
end while loop.
```



Free Energy Basin-Hopping (Chem. Phys. Lett., **625**, 1, 2015)

Consider $F_{\text{new}}(T) - F_{\text{old}}(T) = V_{\text{new}} - V_{\text{old}} + k_B T \ln(o_{\text{new}} \bar{\nu}_{\text{new}}^\kappa / o_{\text{old}} \bar{\nu}_{\text{old}}^\kappa)$.



For the **serine-lysine dipeptide** (right) the free energy minimum changes **twice** with temperature, as determined by the **vibrational entropy**. We can exploit **sparsity** to calculate the vibrational partition function. (CPL, **685**, 288, 2017)

Using optimal **Tsallis** weights rather than **Boltzmann** can **halve** the mean first encounter time for the global minimum (JCP, **141**, 071104, 2014).

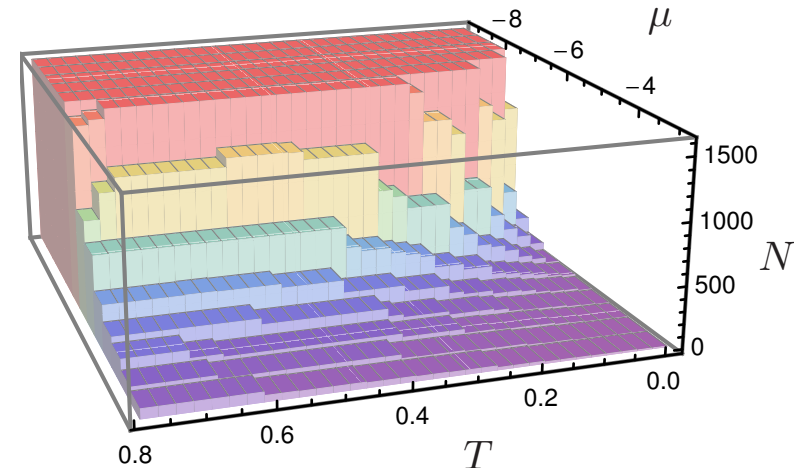
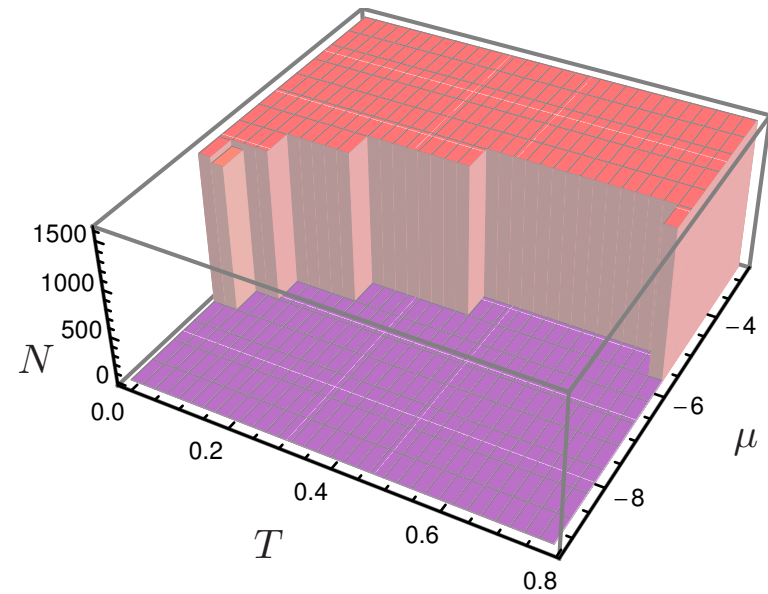
Grand and Semi-Grand Canonical Basin-Hopping (*JCTC*, 12, 902, 2016)

The **accept/reject** condition for **GCBH** employs the local **grand potential**:

$$\xi_{\alpha} = V_{\alpha} - \mu N - k_B T \ln \frac{n_{\alpha} \sqrt{8\pi} |\mathbf{I}_{\alpha}|^{1/2} (k_B T)^{3/2}}{\hbar^3 (\beta \hbar \bar{\nu}_{\alpha})^{\kappa}},$$

which includes the **rigid rotor** partition function, with inertia tensor \mathbf{I}_{α} .

Blocks of conventional BH steps are employed between changes of N .

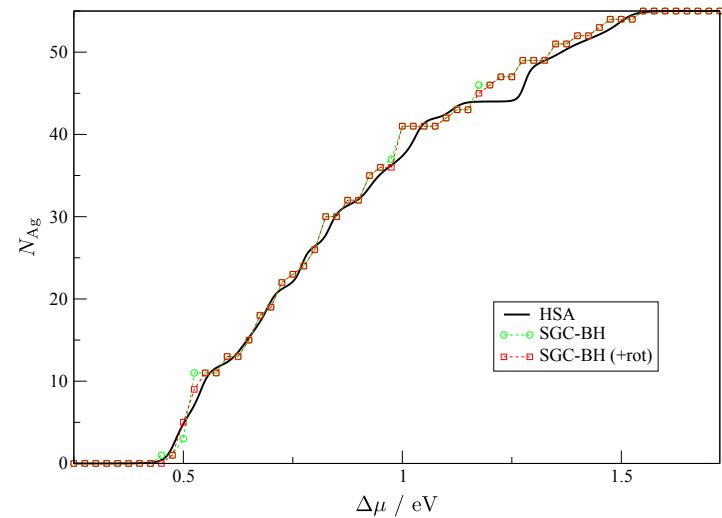
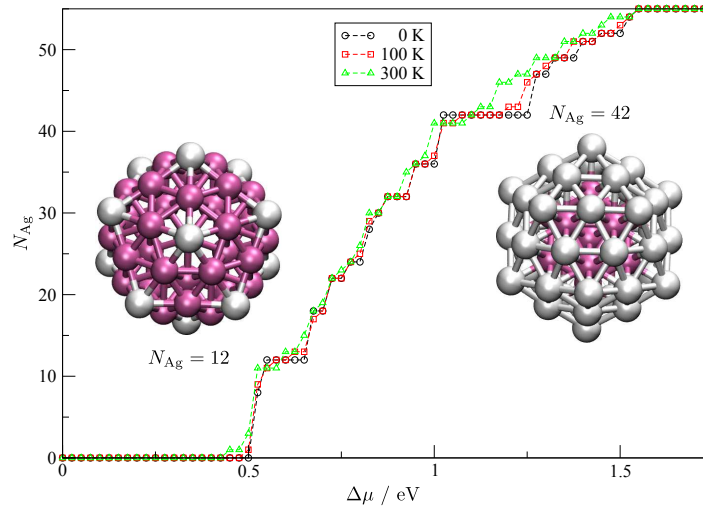


The **most** probable cluster size (left) corresponds to the **smallest** or **largest** size permitted. However, the **least** probable size can occur between the limits (right), suggesting a free energy **barrier** to **nucleation** and cluster growth.

For a **binary semigrand** canonical potential we consider **fixed** $N = N_A + N_B$ and **variable** N_A and N_B with potential

$$\xi_\alpha = V_\alpha - \Delta\mu N_B - k_B T \ln \frac{(k_B T)^{3/2} n_\alpha \sqrt{8\pi} |\mathbf{I}_\alpha|^{1/2}}{\hbar^3 (\beta \hbar \bar{\nu}_\alpha)^\kappa}.$$

$\Delta\mu = \mu_B - \mu_A$ is the chemical potential **difference**; $n_\alpha = 2N_A!N_B!/o_\alpha$.



Left: the most favourable composition for icosahedral $\text{Ag}_n\text{Pd}_{55-n}$ as a function of $\Delta\mu$ exhibits **steps** at $N_{\text{Ag}} = 12$ and $N_{\text{Ag}} = 42$.

Right: **lumping** probabilities for minima with the same composition and including non-icosahedral structures **smooths** the steps.

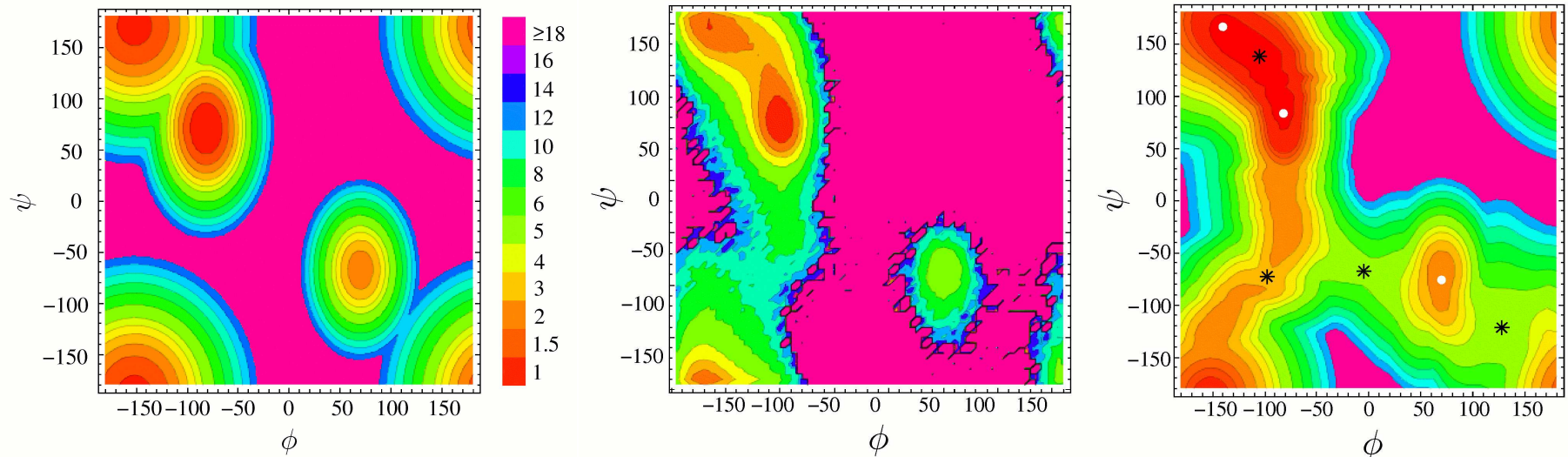
A Modified Superposition Approach (*Chem. Phys. Lett.*, **466**, 105, 2008)

Here the partition function is broken down into contributions from local **minima** and **pathways** as a function of an order parameter, a , with terms

$$Z_i(a, T) = \left(\frac{kT}{h\bar{\nu}_i} \right)^{3N-6} \frac{\exp(-V_i/kT)}{\sqrt{2\pi kT A_i}} \exp \left[-\frac{(a - a_i)^2}{2kT A_i} \right],$$

where A_i is a weighted sum of order parameter derivatives.

Free energy surfaces for **alanine dipeptide** (CHARMM22/vacuum) from **superposition**, **replica exchange**, and **reaction path Hamiltonian** superposition:



Basin-Sampling for Global Thermodynamics (*CPL*, 584, 1, 2013)

Broken ergodicity is treated using basin-hopping, while the configuration space corresponding to high temperature is sampled by parallel tempering.

We define a two-dimensional density of states using systematic local minimisation, and couple these statistics to a model anharmonic form connected to the low-temperature limit for structures obtained via global optimisation.

The number of visits to quench potential energy bin q from instantaneous potential energy bin i in replica r is \mathcal{N}_{iqr} : a two-dimensional histogram.

The corresponding canonical probability distribution for replica r is

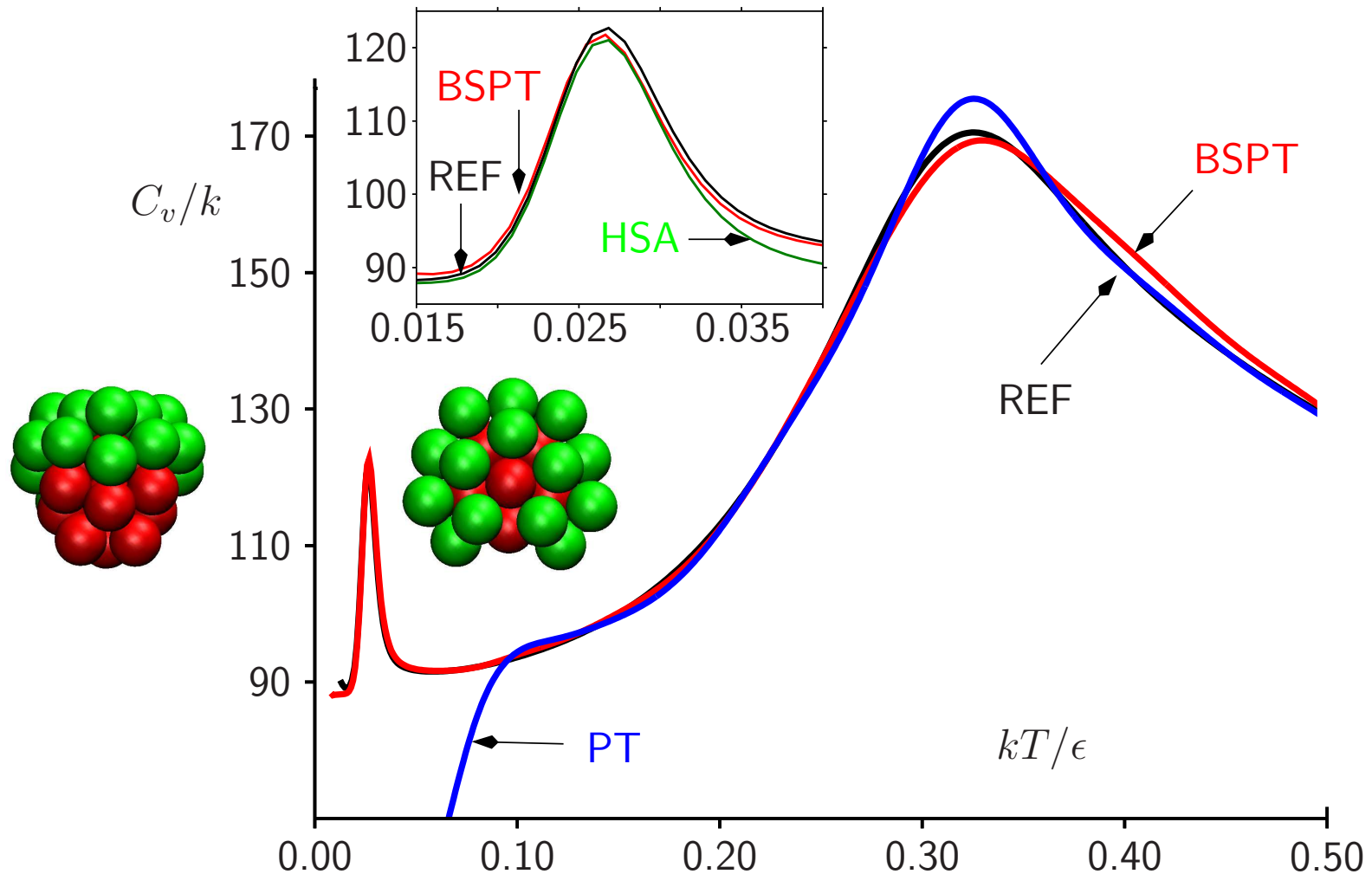
$$P(V_i^I, V_q^Q, T_r) = \mathcal{N}_{iqr} / \mathcal{N}_r \propto \Omega_c(V_i^I, V_q^Q) e^{-V_i^I / kT_r}.$$

The analytical density of states for a Morse potential suggests a model anharmonic representation for each quench bin with two fitting parameters:

$$\ln \Omega_c(V_i^I, V_q^Q) = [(3N - 6)/2 - 1 + e^{A_q} (V_i^I - V_q^Q)] \ln (V_i^I - V_q^Q) + B_q.$$

Accurate thermodynamics were obtained for the **solid-solid** phase transition in **LJ₃₁** with 10^6 equilibration steps (discarded), 5×10^6 parallel tempering steps, and 5×10^6 BSPT steps using 24 replicas and **quenching** every 30 steps.

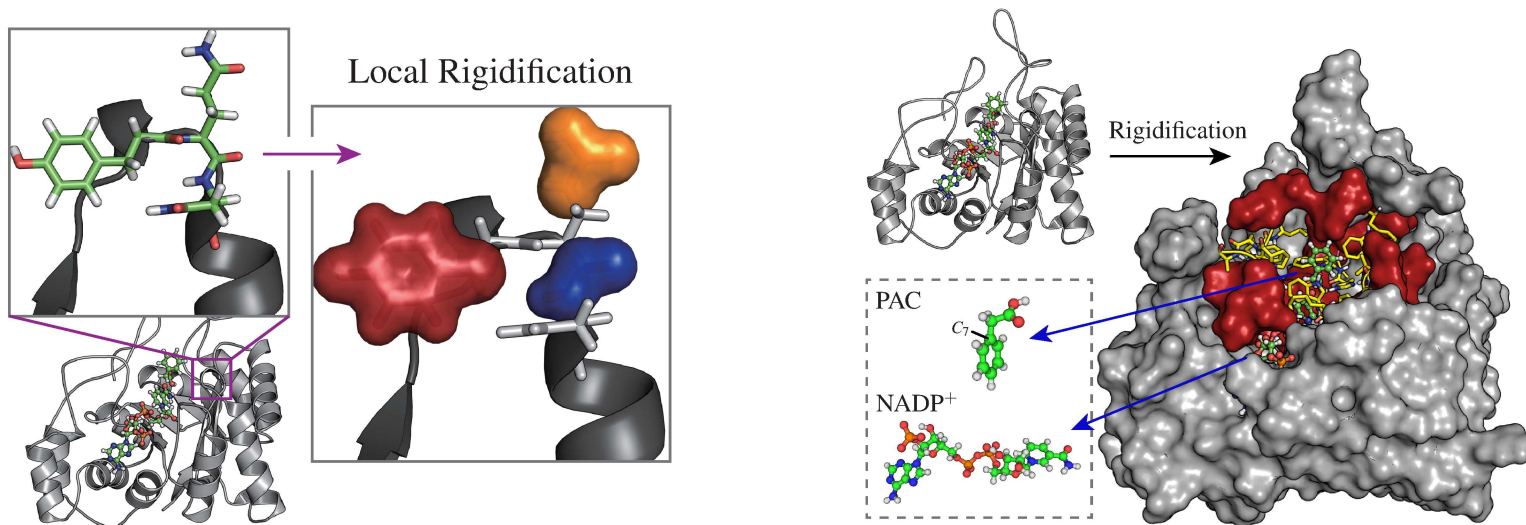
This run took **21.8 minutes** compared to **110.5 hours** for **parallel tempering**.



Factorised Superposition for Ligand Binding (*PCCP*, **16**, 2842, 2014)

The factorisation separates weakly coupled degrees of freedom for consistent reference conformations, based on local rigidification (*JCTC*, **8**, 5159, 2012).

For phenyl acetic acid/aldose reductase the binding free energy converges for factorisation of groups beyond 12 Å to 14 Å from the binding site.



Left: local rigidification of the TYR47 ring, the peptide bond between GLN48 and ASN49, and the amide group of GLN48 (sp^2 centre), residues 47-49.

Right: the unconstrained inner region of the complex, the locally rigidified intermediate region, and the rigidified outer region (80% of the protein).

Discrete Path Sampling (*Mol. Phys.*, **100**, 3285, 2002; **102**, 891, 2004)

Discrete path sampling constructs databases of stationary points that are relevant to global kinetics (*Int. Rev. Phys. Chem.*, **25**, 237, 2006).

No reaction coordinate is required. No projection error.

Phenomenological $A \leftrightarrow B$ rate constants can be formulated as sums over discrete paths, defined as sequences of local minima and the transition states that link them, weighted by equilibrium occupation probabilities, p_b^{eq} :

$$k_{AB}^{\text{SS}} = \frac{1}{p_B^{\text{eq}}} \sum_{a \leftarrow b} P_{ai_1} P_{i_1 i_2} \cdots P_{i_{n-1} i_n} P_{i_n b} \tau_b^{-1} p_b^{\text{eq}} = \frac{1}{p_B^{\text{eq}}} \sum_{b \in B} \frac{C_b^A p_b^{\text{eq}}}{\tau_b},$$

where $P_{\alpha\beta}$ is a branching probability and C_b^A is the committor probability that the system will visit an A minimum before it returns to the B region.

The paths that make the largest contributions to k_{AB}^{SS} can be extracted using the Dijkstra or recursive enumeration algorithms, using edge weights $-\ln P_{\alpha\beta}$ (*J. Chem. Phys.*, **121**, 1080, 2004; *J. Phys. Chem. B*, **112**, 8760, 2008).

Rates from Graph Transformation (*JCP*, **124**, 234110, 2006; **130**, 204111, 2009)

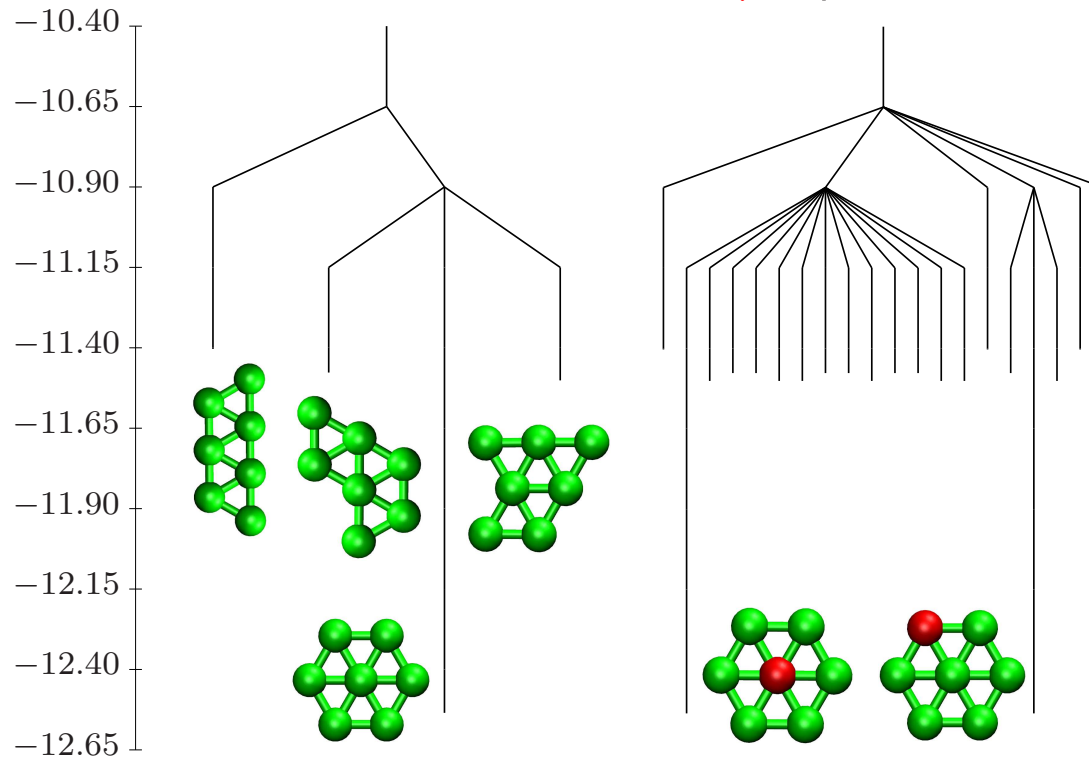
The deterministic **graph transformation** procedure is **non-stochastic** and **non-iterative**. Minima, x , are progressively **removed**, while the branching probabilities and waiting times in adjacent minima, β , are **renormalised**:

$$P'_{\gamma\beta} = P_{\gamma\beta} + P_{\gamma x}P_{x\beta} \sum_{m=0}^{\infty} P_{xx}^m = P_{\gamma\beta} + \frac{P_{\gamma x}P_{x\beta}}{1 - P_{xx}}, \quad \tau'_{\beta} = \tau_{\beta} + \frac{P_{x\beta}\tau_x}{1 - P_{xx}}.$$

Each transformation **conserves** the **MFPT** from every reactant state to the set of product states with an execution time **independent** of temperature:

kT/K	$\Delta F_{\text{barrier}}$	N_{min}	N_{ts}	NGT/s	SOR/s	KMC/s
298	5.0	272	287	8	13	85,138
298	4.5	2,344	2,462	8	217,830	
1007	-	40,000	58,410	35	281	1,020,540
1690	-	40,000	58,410	39	122,242	

Permutational Isomerisation of $\text{LJ}_7^{2\text{D}}$ (*Mol. Phys.*, **100**, 3285, 2002)



Disconnectivity graphs for $\text{LJ}_7^{2\text{D}}$. Left: **permutation-inversion** isomers of the four local minima are collected **together**. Right: one of the atoms is **tagged**, lowering the permutational degeneracy.

The fastest ten paths contribute about **74%** of the total rate constant at $kT/\epsilon = 0.05$. Various combinations of **diamond-square-diamond** rearrangements make significant contributions.

Benchmarks for Landscape Exploration

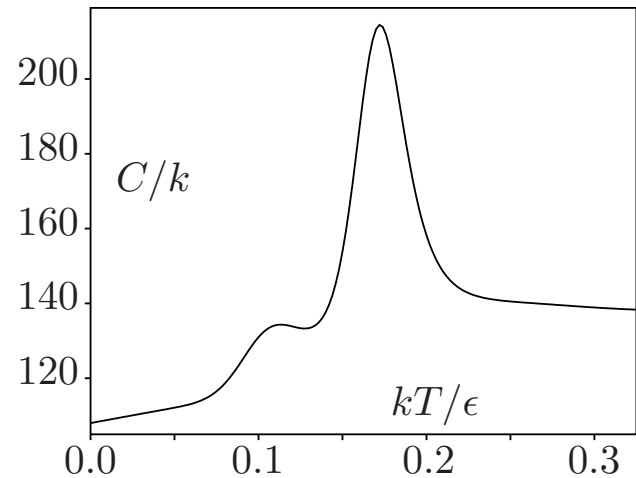
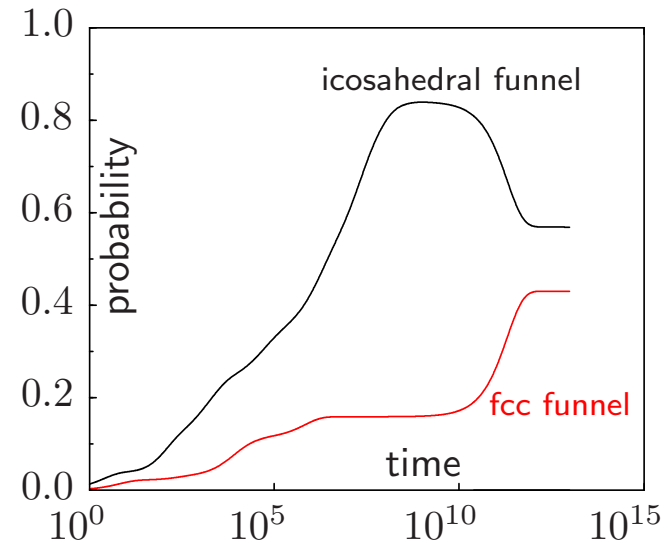
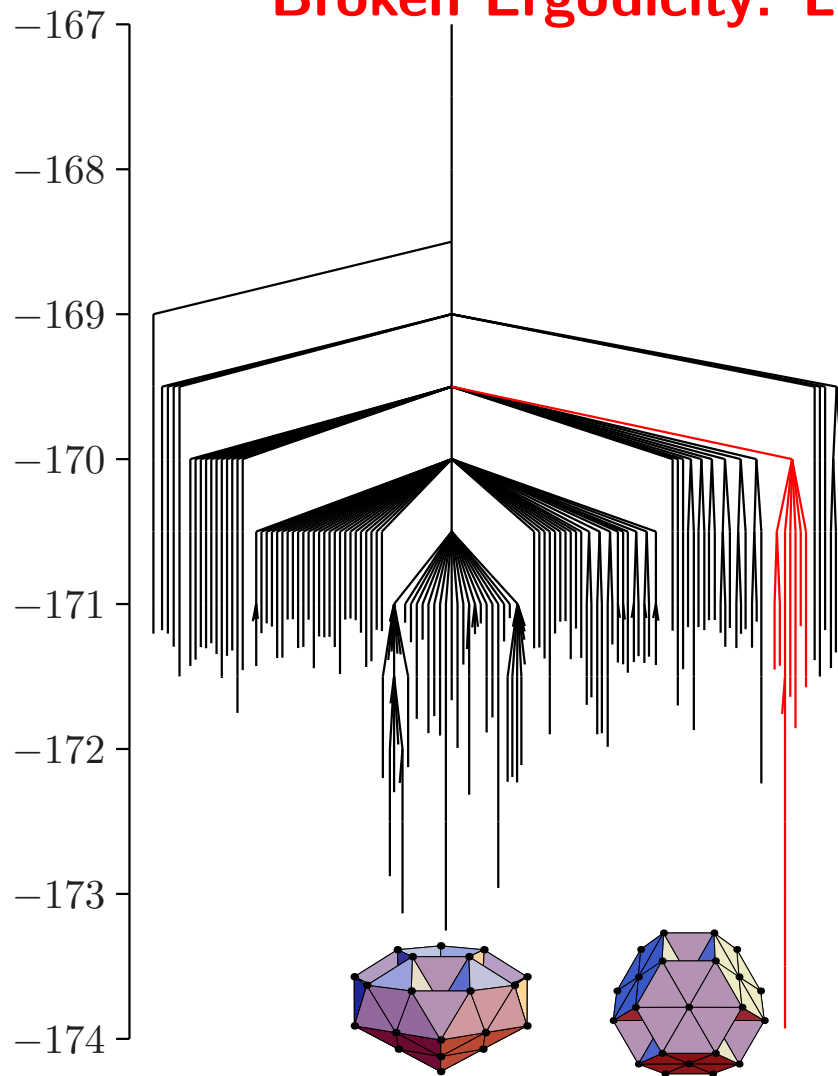
Minimisation: Nocedal's algorithm, **LBFGS**, with line searches removed.

Transition states: single-ended searches use **hybrid eigenvector-following** (*PRB*, **59**, 3969, 1999; *JCP*, **111**, 7010, 1999; *CPL*, **341**, 185, 2001), double-ended searches use the **doubly-nudged** elastic band approach (*JCP*, **120**, 2082, 2004; **140**, 044115, 2014).

The **GMIN** (global optimisation), **OPTIM** (transition states and pathways) and **PATHSAMPLE** (discrete path sampling) programs are available from the **Cambridge Landscape Database** under the **Gnu** General Public License.

- Interfaces to many **electronic structure** codes are included.
- Current **svn** tarball image: <http://www-wales.ch.cam.ac.uk>
- <http://www-wales.ch.cam.ac.uk/tsbenchmarks.html> **Peptide** examples
- <http://theory.cm.utexas.edu/benchmarks/index.html> **OptBench** test suite
- <https://github.com/wales-group/examples> **Curated examples**

Broken Ergodicity: LJ_{38} (*Phys. Rev. E*, **60**, 3701, 1999)



LJ_{38} exhibits a **double funnel** due to competition between icosahedral and truncated **octahedral** morphologies. The interconversion rate for Ar_{38} is calculated as 55 s^{-1} at 14 K where a **solid-solid** transition occurs.

Simulating structural transitions by direct transition current sampling: The example of LJ₃₈

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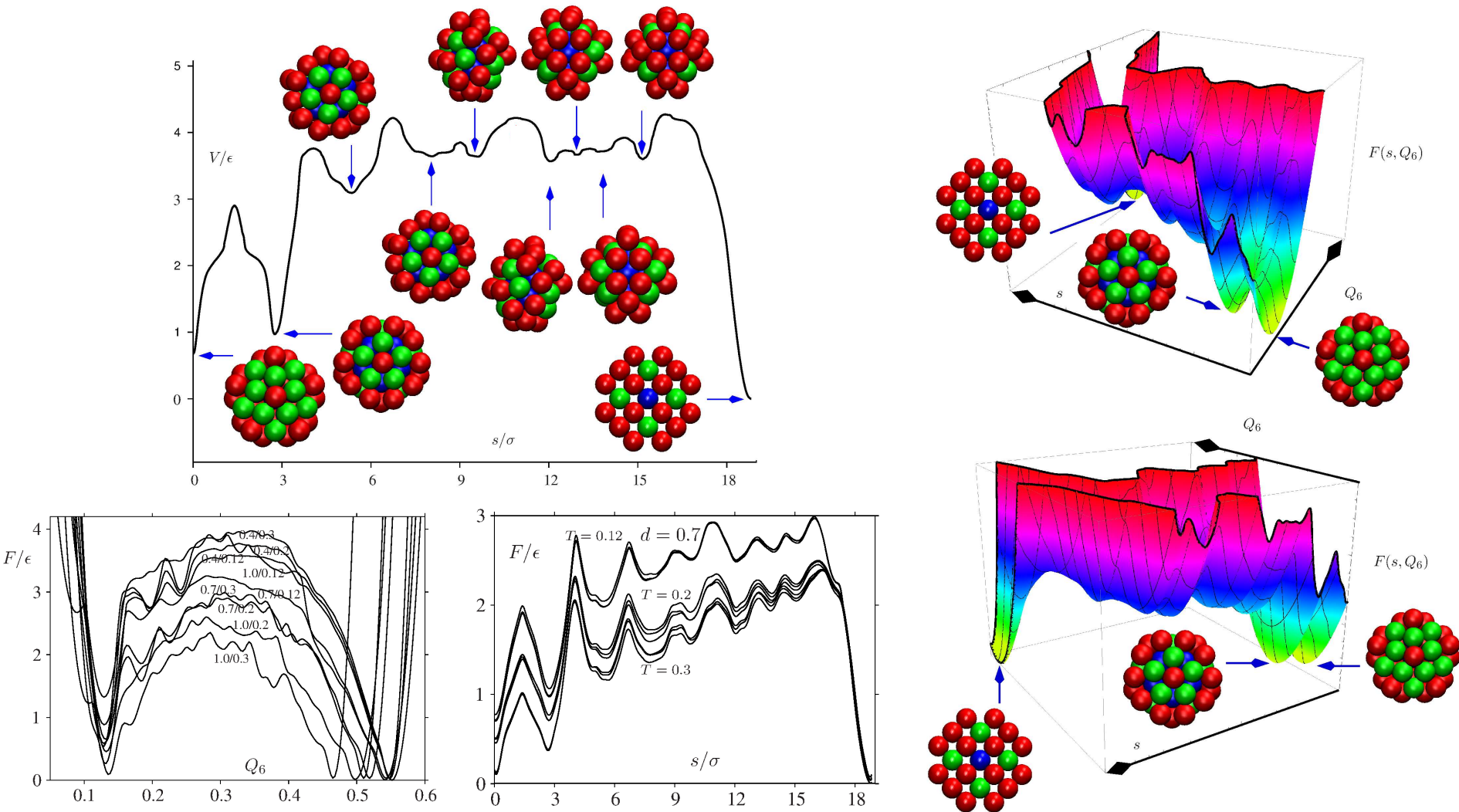
²*CNRS; ESPCI, 10 rue Vauquelin, UMR 7636 PMMH, 75005 Paris, France*

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(Received 2 March 2011; accepted 21 June 2011; published online 20 July 2011)

Another attempt to study the transitions between the two funnels of LJ₃₈ relies on the use of transition path sampling.³³ Because of the number of metastable states separating the two main basins, the traditional shooting and shifting algorithm failed here, despite previous success for smaller LJ clusters.³⁹ The authors thus developed a two-ended approach which manages to successfully locate reaction paths between the two basins: they started from a straight trial trajectory linking the two minima, and obtained convergence towards trajectories of energies similar to those obtained in the discrete path sampling approach.³³ Although the authors point out the lack of ergodicity in the sampling within their approach and the sensitivity on the “discretization” of the trajectories, this is nevertheless a progress and the main drawback remains the high computational cost (the work needed 10⁵ h of central processing unit (cpu) time) to obtain such converged trajectories. In contrast, the simulations we present below required less than 10² h of cpu time.

Free Energy Profiles for an LJ₃₈ Pathway (*JCP*, 142, 130901, 2015.)

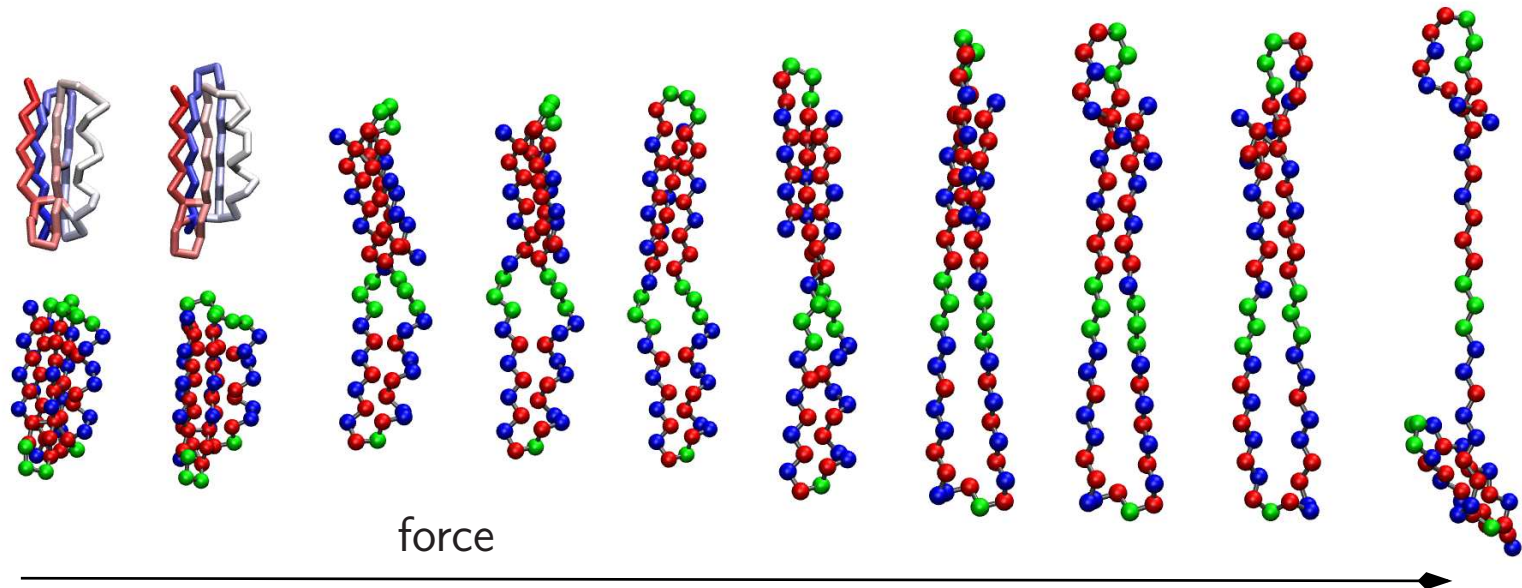


Projection onto the **bond order** parameter Q_6 averages over **surface reorganization** and other mechanistic details. However, the profile based on pathways defined **geometrically** faithfully reflects the underlying **barriers**.

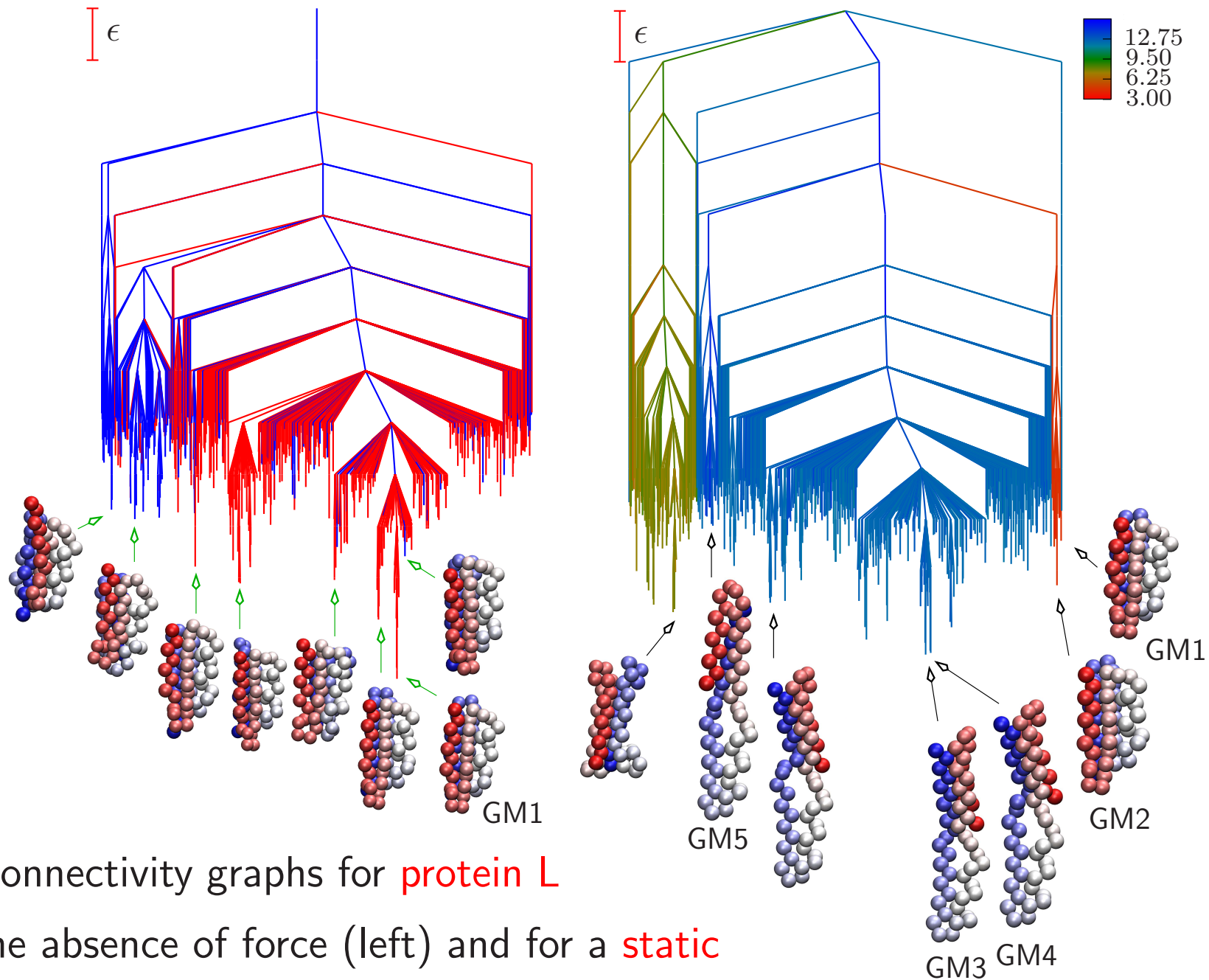
Folding and Pulling for Protein L and Protein G

Folding pathways and the evolution of the energy landscape as a function of static force have been analysed for protein L and protein G using a sequence-dependent BLN model. B=hydrophobic, L=hydrophilic, and N=neutral.

Protein L forms the N-terminal hairpin 1 first, followed by the C-terminal hairpin 2. The order is reversed for protein G, with an early intermediate.

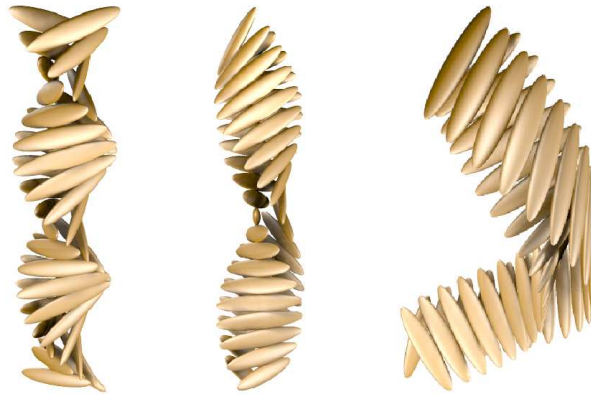


Distinct global minima for protein L as a function of increasing static pulling force applied between beads 10 and 32. (*J. Phys. Chem. B*, **136**, 8394, 2012)



Disconnectivity graphs for **protein L** in the absence of force (left) and for a **static pulling force** (right) applied between beads **10** and **32**.

Helical Bilayers From Frustrated Building Blocks (*JPCB*, **117**, 7918, 2013)



helical fibre morphologies

bilayer filaments

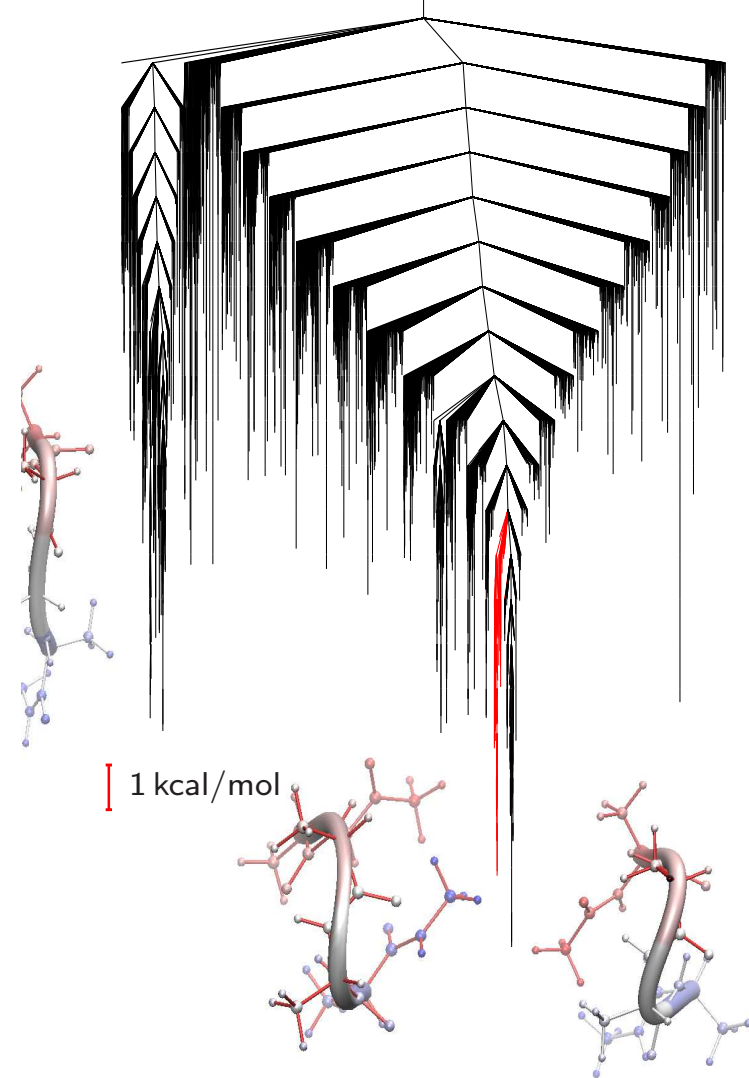
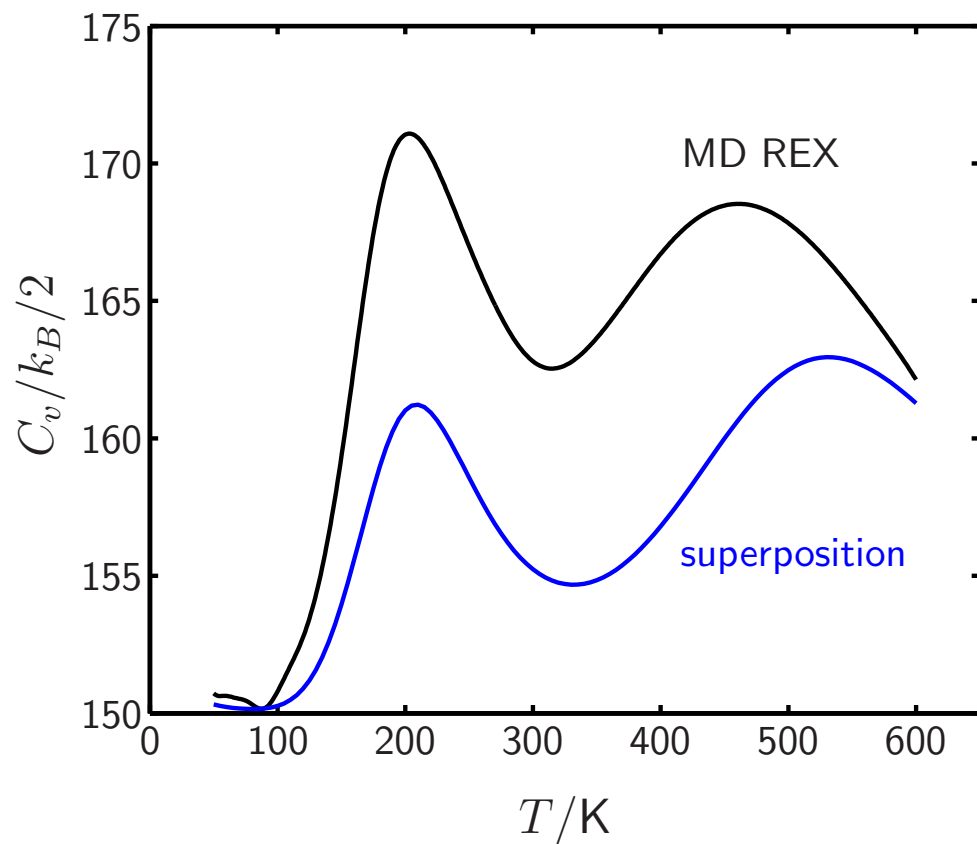
magnetic bilayers

Left: introduction of a **cytochrome** domain into an **amyloid fibre** can change the morphology from **twisted** to **spiral** ribbons and induce systematic **kinking**.

Centre: rigid building blocks consisting of **two ellipsoids** can reproduce these structures, which are also observed for **Bauhinia seedpods**.

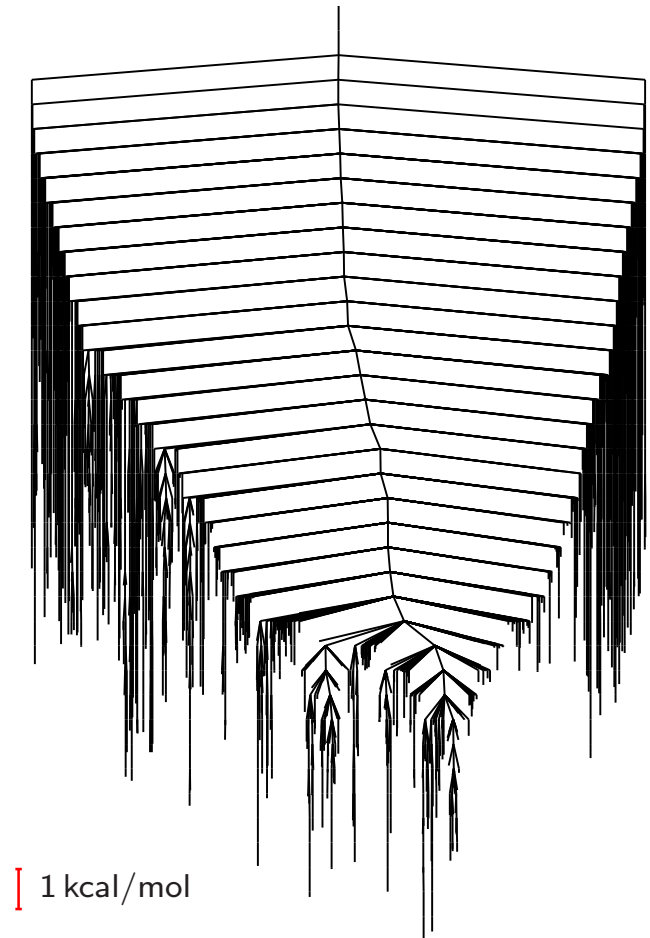
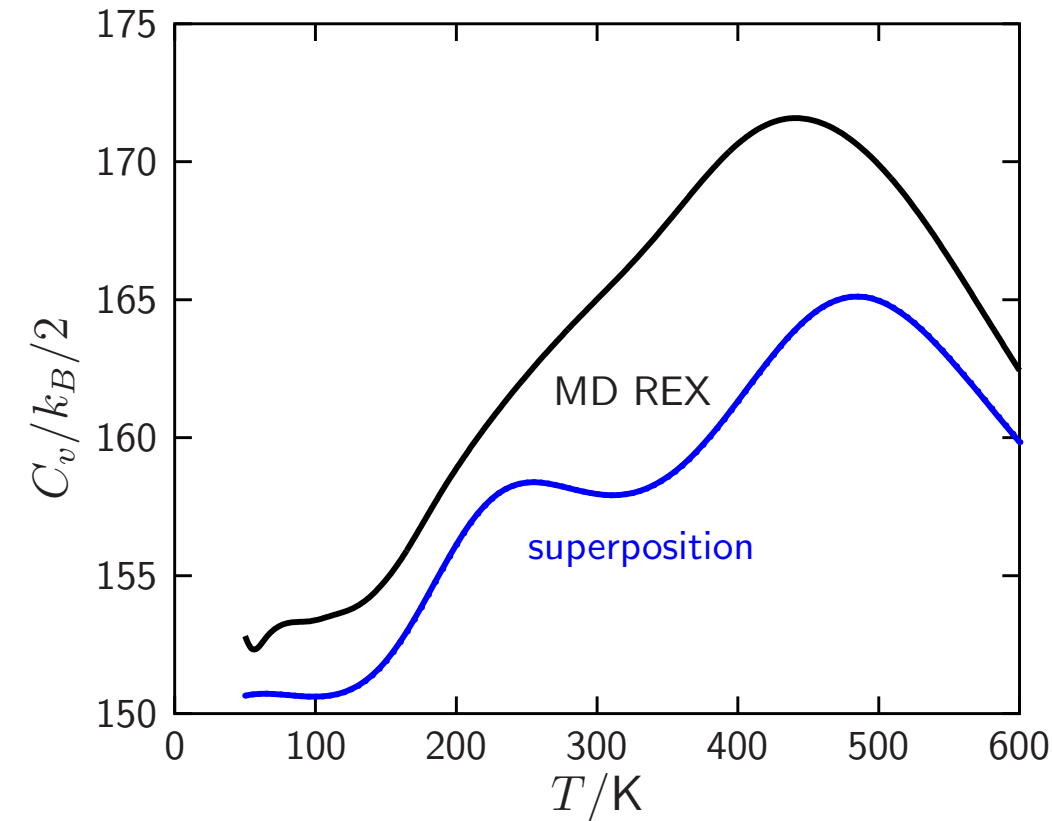
Right: the structure depends mostly on the **internal geometry** of the building blocks, rather than details of the **potential**. The **design** principles extend to **macroscopic** helices formed from elliptical **magnets**.

Thermodynamics for Ala_4 (*J. Chem. Phys.*, **139**, 121909, 2013)



Ala_4 in vacuum (**charmm27**) has a **low** temperature C_v **peak**, corresponding to the hundred or so lowest minima in the **disconnectivity graph**. The **high** temperature peak corresponds to the finite system analogue of **melting**.

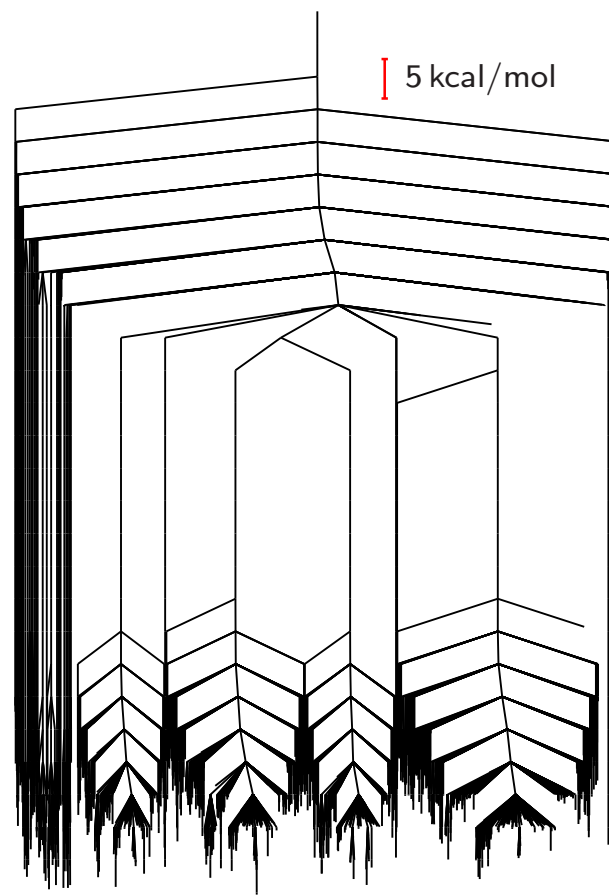
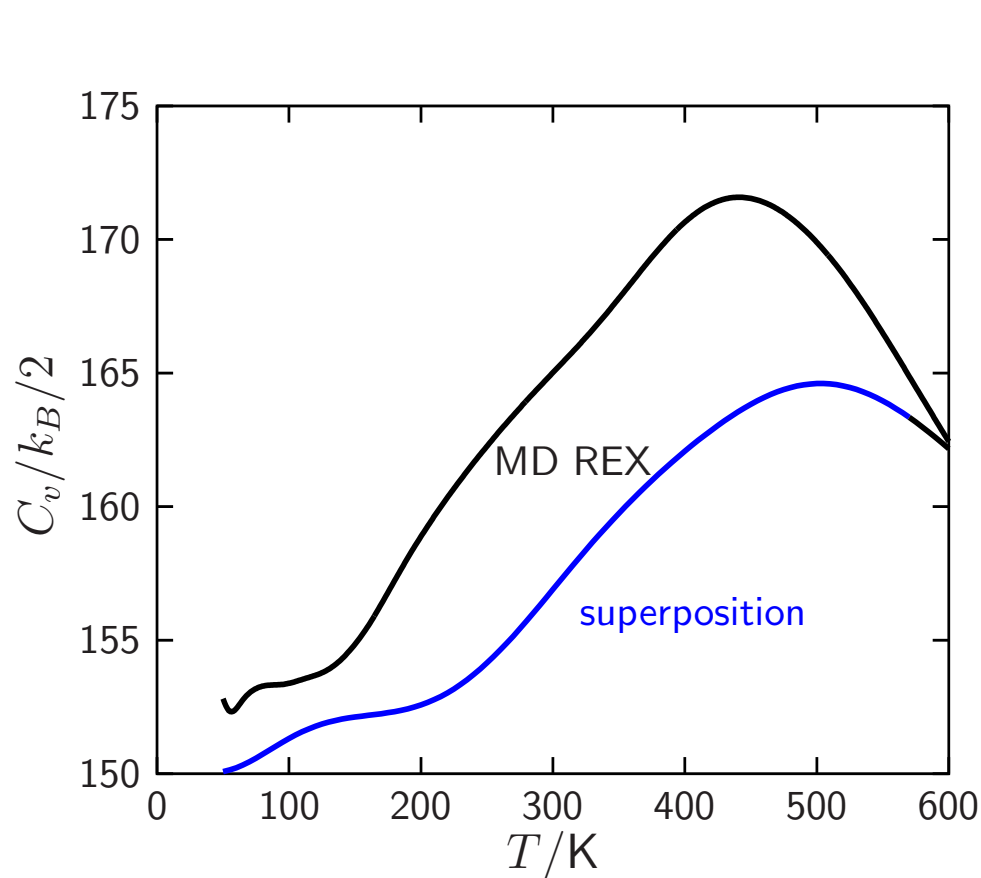
Thermodynamics for Ala₄ in Vacuum: AMBER



Replica exchange (**REX**) and **superposition** results for the heat capacity C_v of ala₄ in vacuum (**amber99sb**) appear similar to **CHARMM**.

(*J. Chem. Phys.*, **139**, 121909, 2013)

Thermodynamics for Ala₄ in Vacuum: AMBER



The **global minimum** for this potential has a mixture of **L** and **D** amino acids. The landscape separates into regions with different **L/D** composition, separated by barriers of order **90 kcal/mol**. (*J. Chem. Phys.*, **139**, 121909, 2013)

Symmetry and Energetics (*Chem. Phys. Lett.*, **285**, 330, 1998; **294**, 262, 2008)

‘The symmetry characteristic of a phenomenon is the **maximal symmetry** compatible with the existence of the phenomenon’ (**Pierre Curie**)

‘The perfection of mathematical beauty is such... that whatsoever is most beautiful and regular is also found to be most useful’ (**D’Arcy Thompson**)

However, many systems do **not** exhibit their highest possible symmetry.

We can write the total energy as a **sum** over contributions from a many-body expansion, involving single atom, pairwise and three-body terms, etc.

If these terms are drawn from the same **distribution** then geometrical symmetry (degeneracies) would be manifested as **correlation**.

The **variance** is **larger** when correlation is present. Symmetrical structures are therefore more likely to have particularly **high** or particularly **low** energy.

Low-lying structures are therefore likely to exhibit symmetry.

More formally, denote the **mean** and **variance** of a variable, X , drawn from probability distribution, $p(X)$, as μ and σ^2 .

The variance of a **sum** of N such variables, X_i , is then

$$\text{Var} \left(\sum_{i=1}^N X_i \right) = N\sigma^2 + N(N-1)\rho\sigma^2,$$

where the **correlation** ρ is defined by

$$\rho\sigma^2 = \int (X - \mu)(X' - \mu)p(X, X')dXdX'.$$

For $\rho = 0$ the variance is $N\sigma^2$, but for $\rho = 1$ it rises to $N^2\sigma^2$.

Open question: how should we treat **approximate** geometrical symmetry, for example, in large **biomolecules** or **condensed matter**?

Open question: is it possible to build a more **rigorous** theory, and make connections with concepts such as **designability**?

Connecting Dynamics and Thermodynamics (*Science*, **293**, 2067, 2001)

The organisation of a PES is governed by its **stationary points**, where Taylor expansions provide local descriptions in terms of **Hessian matrices**.

The organisation of **families** of PES's as a function of **parameters** in the potential is determined by the stationary points that possess additional zero Hessian eigenvalues, known as **non-Morse** points.

Catastrophe theory provides a local representation of the PES around non-Morse points as a function of **both** atomic coordinates and parameters.

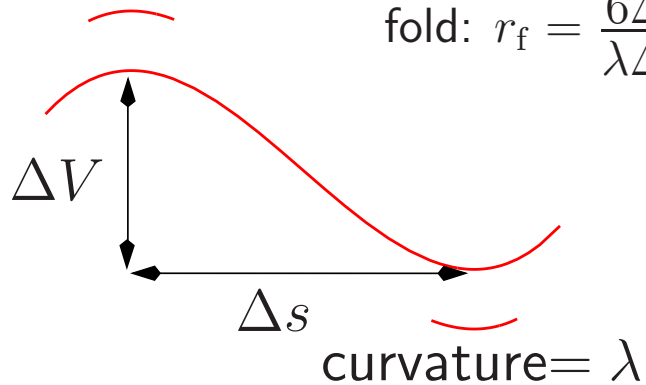
The **splitting lemma** reduces the dimensionality to the **essential** variables, while **transversality** guarantees that the resulting classifications are **universal**.

The simplest one-parameter catastrophes are the **fold**, $f(x) = \frac{1}{3}x^3 + ax$, and the symmetrical **cusp**, $f(x) = \frac{1}{4}x^4 + \frac{1}{2}ax^2$.

Geometries of the **fold** and **cusp** catastrophes.

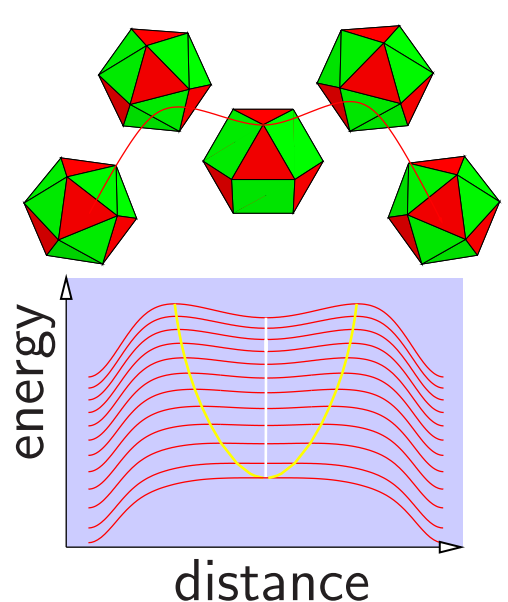
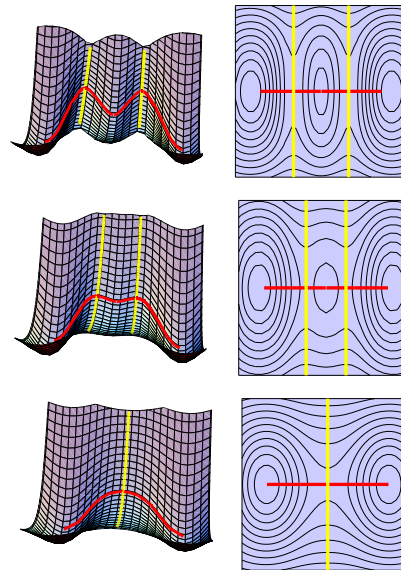
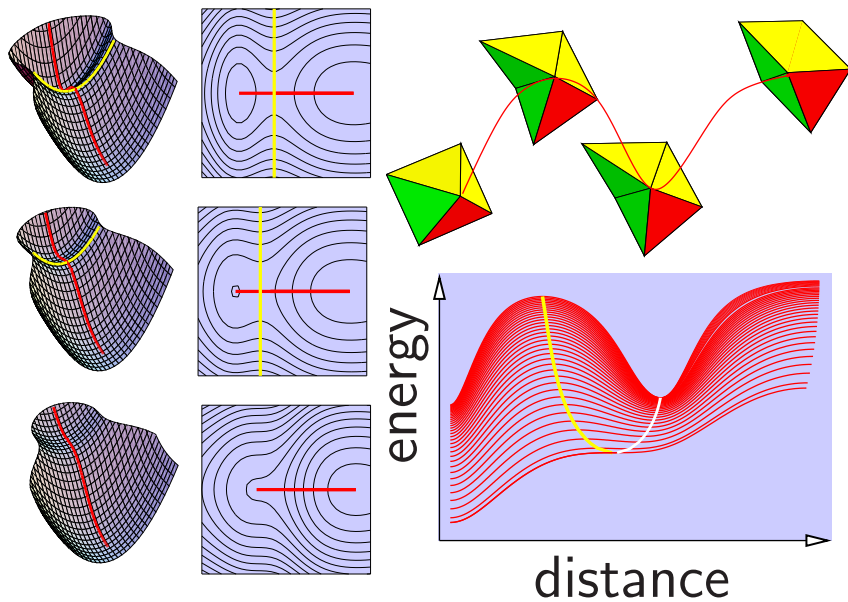
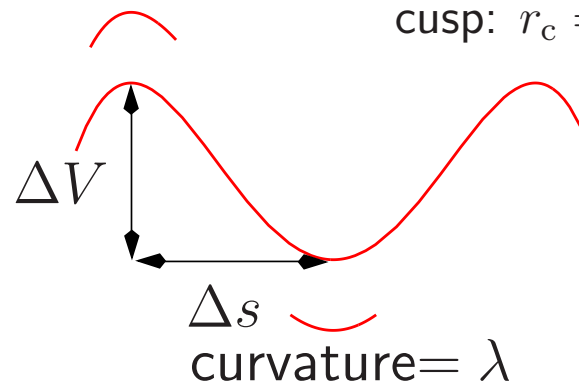
curvature = $-\lambda$

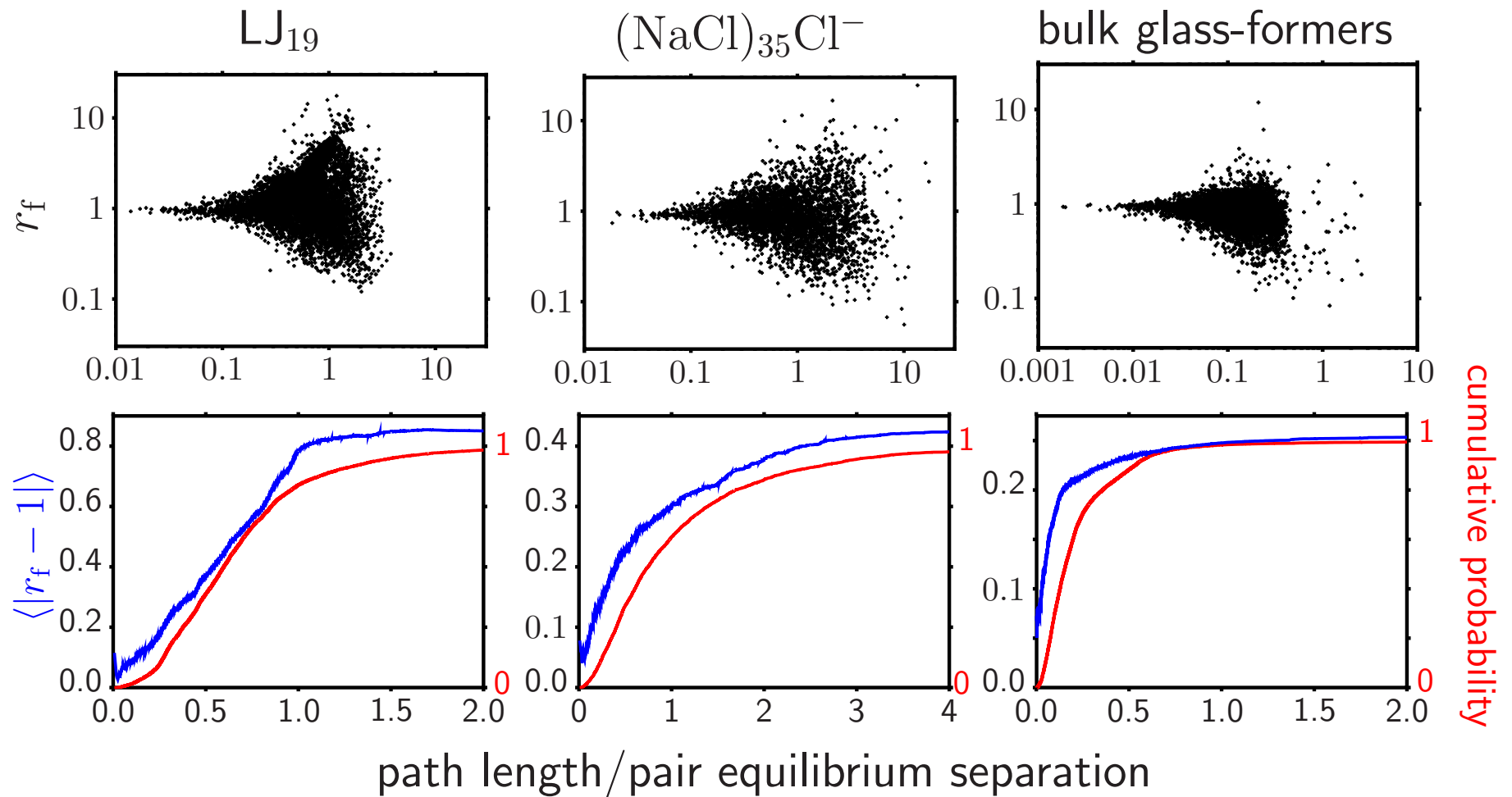
fold: $r_f = \frac{6\Delta V}{\lambda\Delta s^2} = 1$



curvature = -2λ

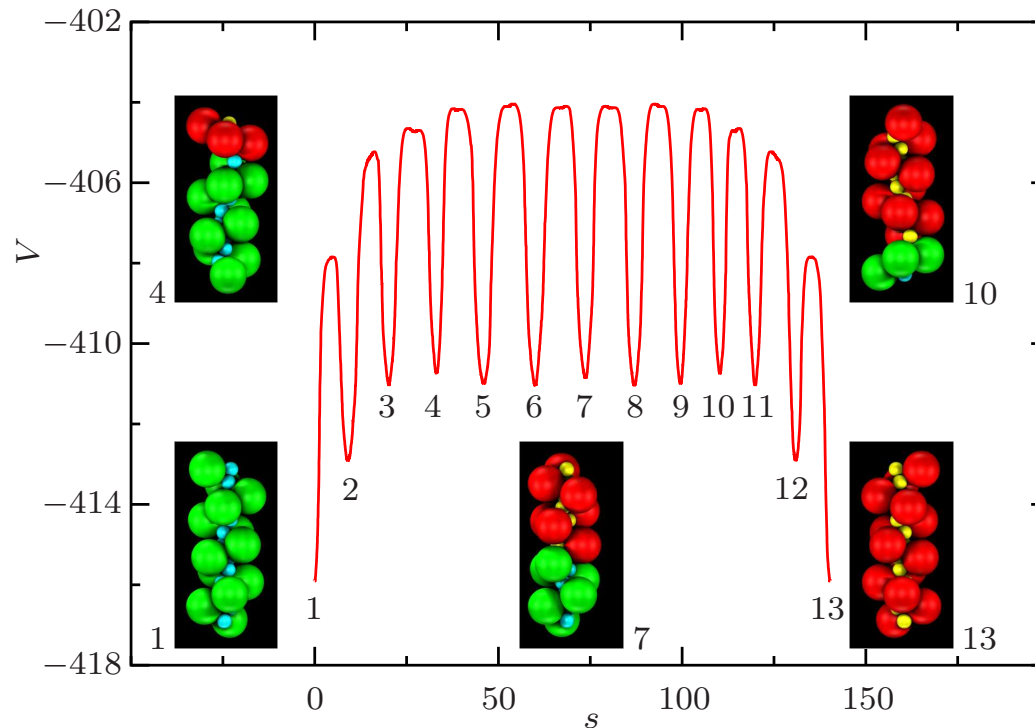
cusp: $r_c = \frac{4\Delta V}{\lambda\Delta s^2} = 1$





For systems with a fixed potential we effectively have a **snap-shot** of parameter space. On average, r_f remains **close to unity** for many pathways in both model clusters and bulk, providing an explanation for **Hammond's postulate**.

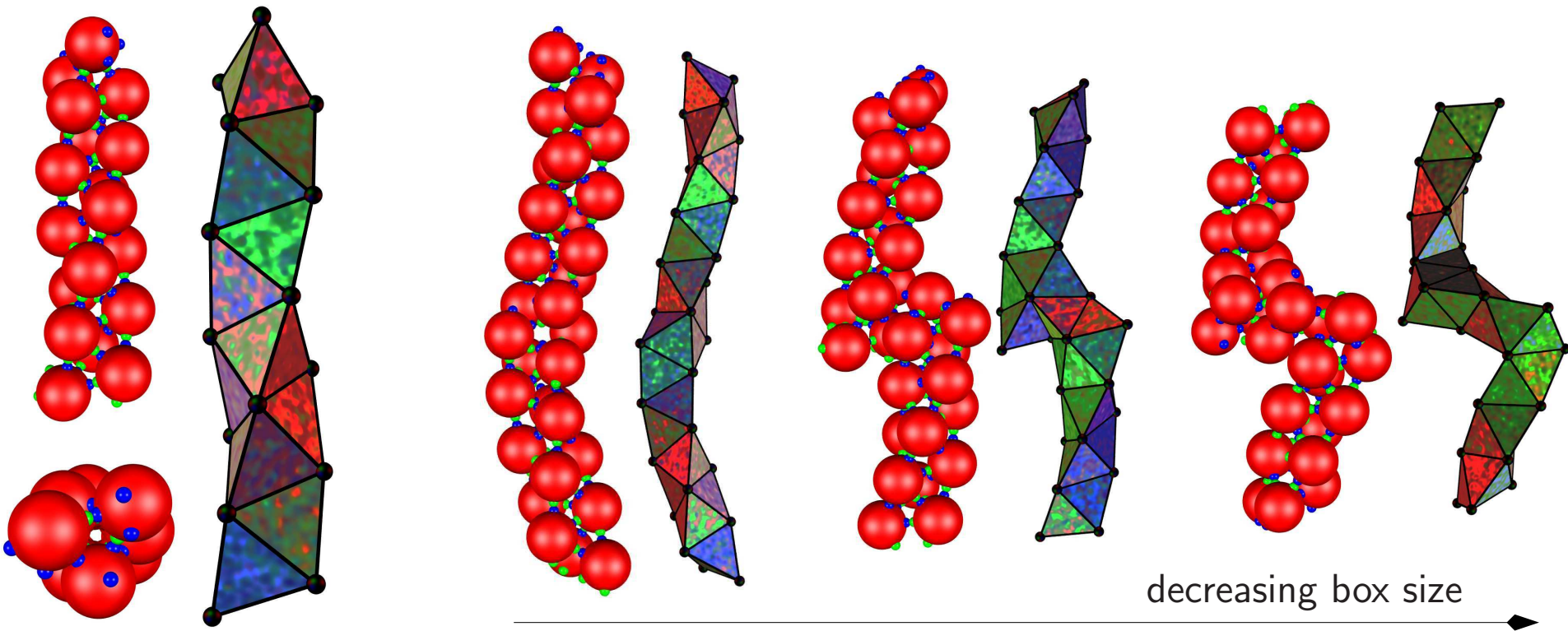
A Nanodevice (*Soft Matter*, **7**, 2325, 2011)



Coupled **linear** and **rotary** motion has been characterised for a helix composed of 13 asymmetric **dipolar dumbbells** in the presence of an **electric field**.

The helix changes **handedness** as the boundary between segments propagates along the strand via successive steps that switch the dumbbells.

Designing a Bernal Spiral (ACS Nano, 7, 1246, 2013)

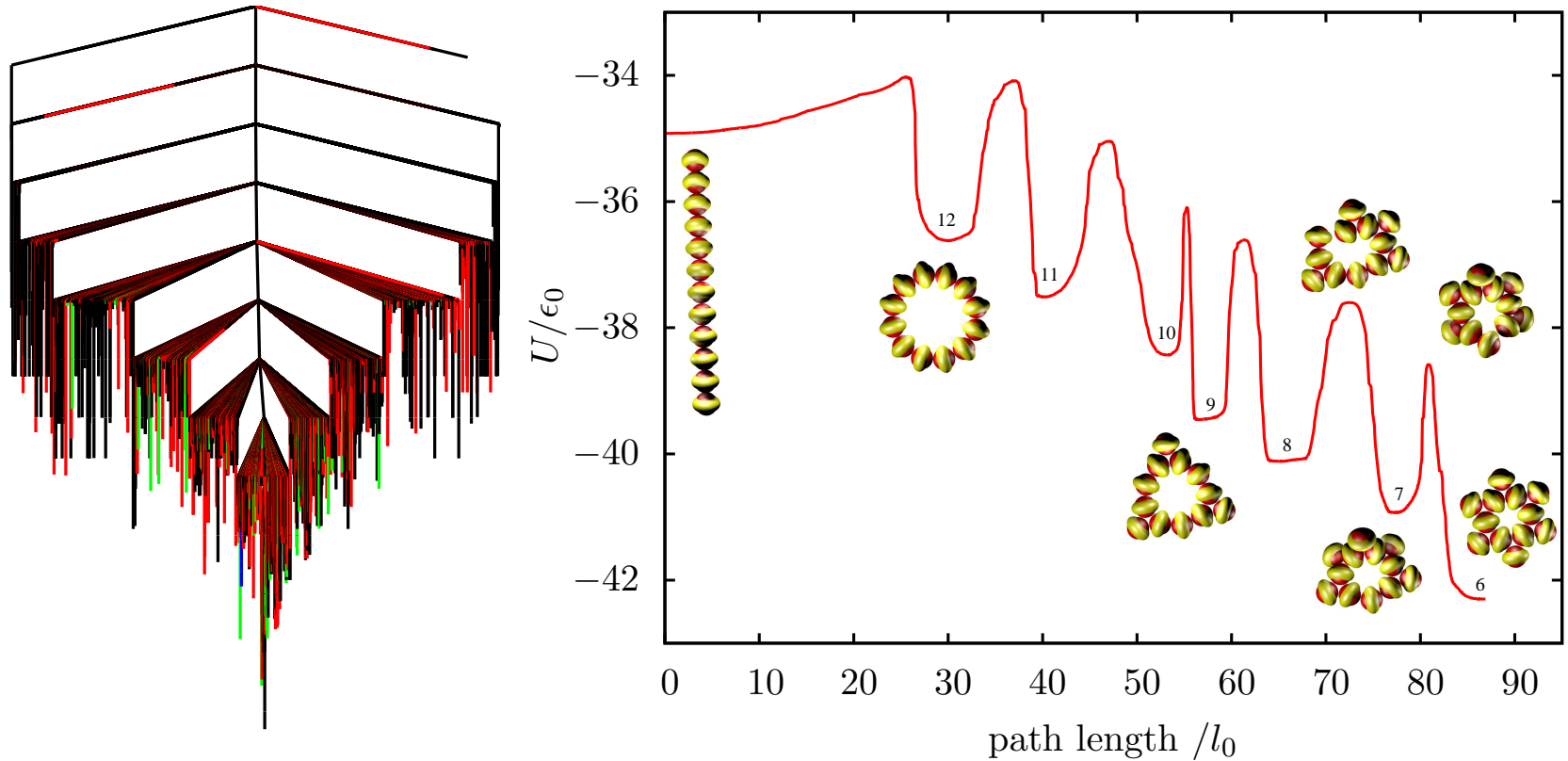



The simplest building blocks that support a Bernal spiral as the global minimum involve a **single** patch-antipatch pair **offset** by about 10° from linearity.

Left: Alternative views of a chiral Bernal spiral consisting of 18 particles.

Right: **compressed** spirals (30 particles, periodic boundaries) exhibit **super-coiling** or **breaks**, which resemble structures seen in **confocal microscopy**.

Kagome Structures (*Soft Matter*, **11**, 6663, 2015)



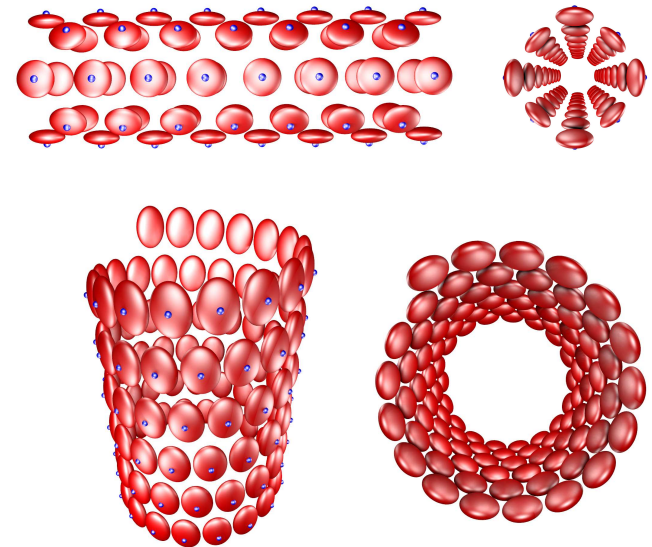
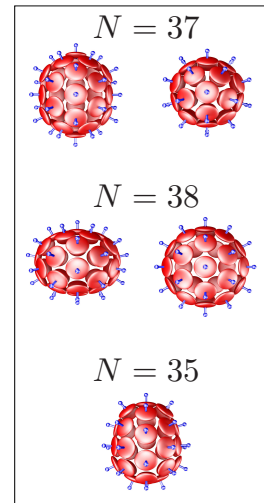
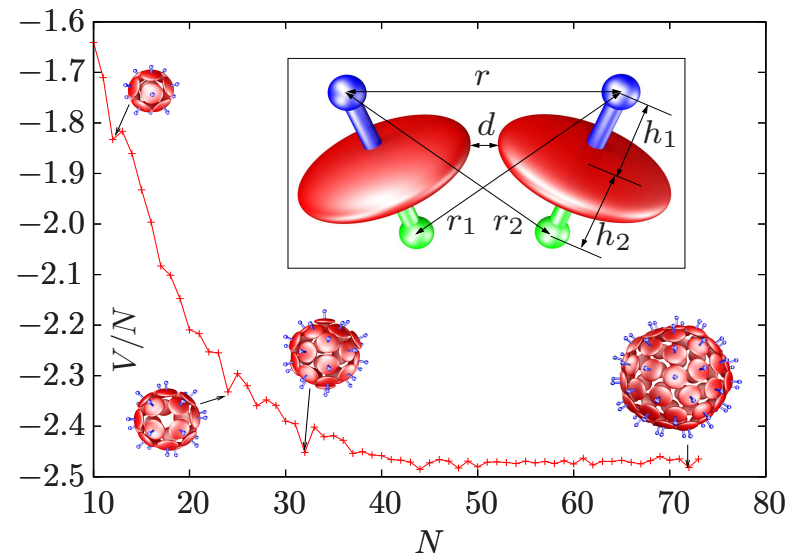
Energetically stabilised **Kagome** structures were designed using soft anisotropic triblock **Janus** particles.  This **unconstrained** model predicts that **sedimentation** effects **enhance** the stability.

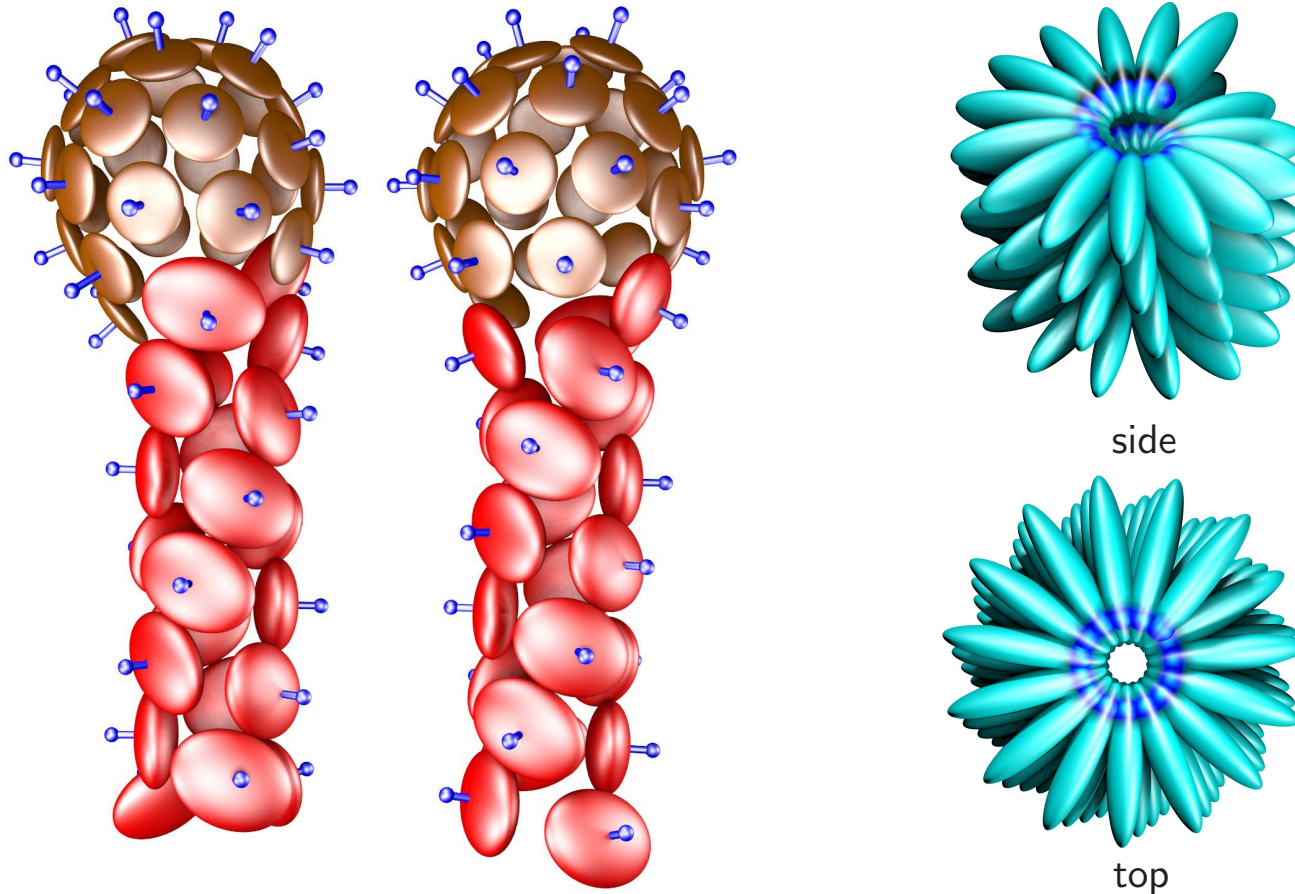
Rearrangements between competing structures are highly **cooperative**.

Emergent Behaviour from Simple Models (*ACS Nano*, 4, 219, 2010)

Adding two repulsive **axial** Lennard-Jones sites to an **ellipsoidal** core produces remarkably versatile building blocks. **Oblate** ellipsoids favour **shells**, while stronger repulsion for the longer semiaxis produces **tubes** and **spirals**.

Global minima for the **oblate** ellipsoids include **icosahedra** for $N = 12, 32$ and 72 ($T = 1, 3$ and 7), the **snub cube** observed for polyoma virus capsids at $N = 24$, and **conical**, **biaxial**, **prolate**, and **oblate** shells at other sizes.

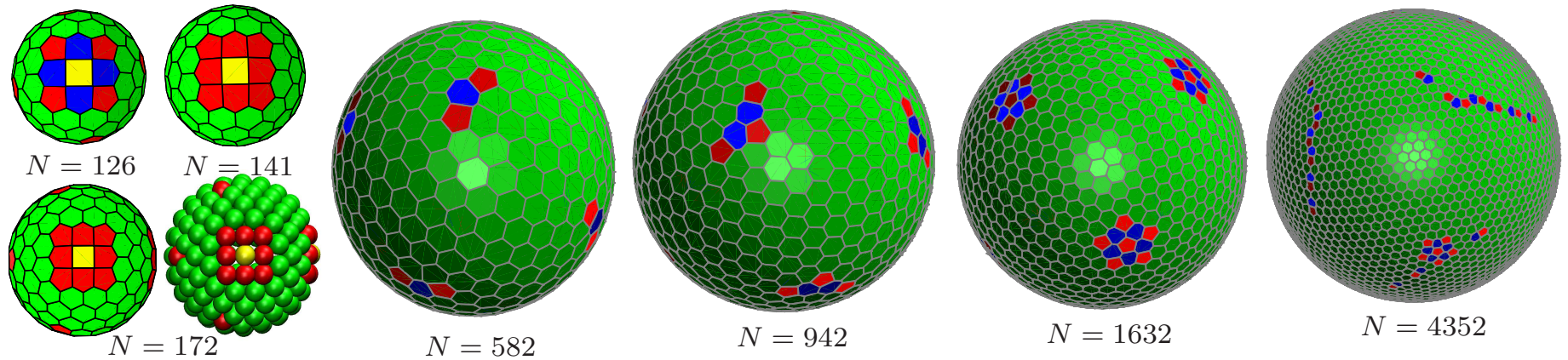




Mixing ellipsoidal building blocks that favour shells and tubes produces structures with distinct **head** and **tail** regions (left): the **Frankenphage**.

Particles with a Lennard-Jones site **buried** in the ellipsoid assemble into a **spiral** structure (right) with parameters similar to **tobacco mosaic virus**.

The Thomson Problem (*Phys. Rev. B*, **74**, 212101, 2006; **79**, 224115, 2009)



Long-ranged potential: $V = \sum_{i < j} 1/|\mathbf{r}_i - \mathbf{r}_j|$ with $|\mathbf{r}_i| = 1$. Twelve five-coordinate particles (**disclinations**) enable a spherical system to obey Euler's rule for the **disclination charge**.

Pentagon patches, extended dislocations (**scars**), **twinned** defects, **rosettes**, and embryonic **grain boundaries** occur in larger systems.

Structures provide models for **spherically constrained** systems: multielectron bubbles in superfluid helium, cell surface layers, 'colloidosomes', colloidal silica microspheres, superconducting films, lipid rafts deposited on vesicles.

Thermodynamic Properties from the Superposition Approach

In the **superposition** approach we write the global partition function as a sum over local minima. The weight for minimum α includes the number of distinguishable **permutation-inversion** isomers, N_{α}^{PI} , which cannot be **superimposed** by rigid body rotations when the atoms are **labelled**.

This factor is related to the **order** of the rigid molecule point group, o_{α} , as

$$N_{\alpha}^{\text{PI}} = \left(2 \prod_{\beta} N_{\beta}! \right) / o_{\alpha} \equiv P / o_{\alpha}. \quad (1)$$

$Z(T)$ is **decomposed** into contributions from each local minimum, $Z_{\alpha}(T)$, which are **identical** for all the corresponding permutation-inversion isomers:

$$Z(T) = \sum_{\alpha} N_{\alpha}^{\text{PI}} Z_{\alpha}(T) = P \sum_{\alpha=1}^{N^{\text{st}}} Z_{\alpha}(T) / o_{\alpha}. \quad (2)$$

Higher point group **symmetry** acts to **reduce** the thermodynamic weight.

Regrouping Stationary Point Databases

Lumping local minima together (recursively) if they are separated by **low** barriers or **fast** rates reduces the dimension of the kinetic transition network (*J. Chem. Phys.*, **123**, 234901, 2005; *J. Chem. Phys.*, **121**, 1080, 2004). It also provides a self-consistent definition of **products** and **reactants**.

The **occupation probability** and **free energy** of a group of minima, J are

$$p_J^{\text{eq}}(T) = \sum_{j \in J} p_j^{\text{eq}}(T) \quad \text{and} \quad F_J(T) = -kT \ln \sum_{j \in J} Z_j(T),$$

and the free energy of the **transition states** connecting J and L is then

$$F_{LJ}^{\dagger}(T) = -kT \ln \sum_{(lj)^{\dagger}} Z_{lj}^{\dagger}(T), \quad l \in L, \quad j \in J,$$

$$\text{with} \quad k_{LJ}^{\dagger}(T) = \sum_{(lj)^{\dagger}} \frac{p_j^{\text{eq}}(T)}{p_J^{\text{eq}}(T)} k_{lj}^{\dagger}(T) = \frac{kT}{h} \exp \left[-\frac{(F_{LJ}^{\dagger}(T) - F_J(T))}{kT} \right].$$